



State-of-the-Art Superconducting RF Technology for Accelerators

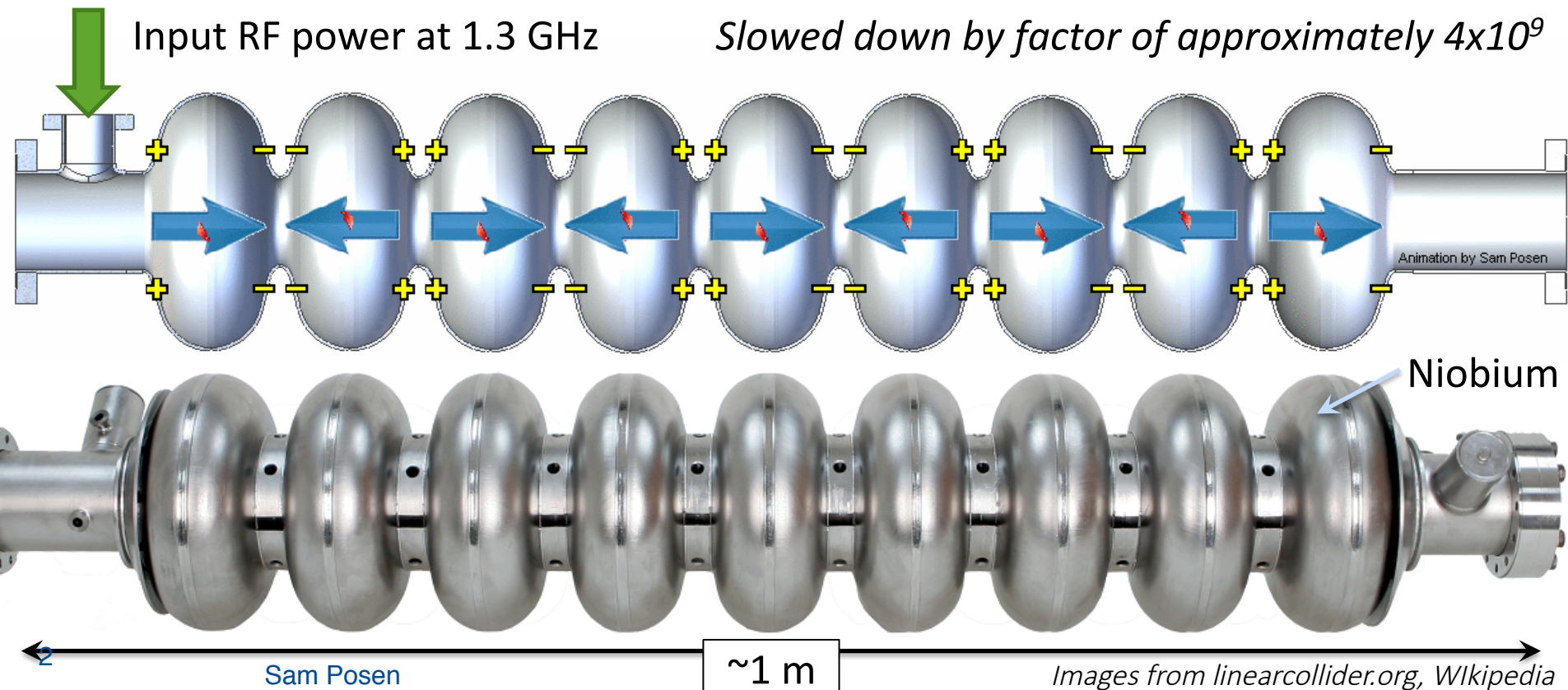
Sam Posen

Fermilab Colloquium

15 February 2017

Particle Acceleration via SRF Cavities

- Superconducting radiofrequency (SRF) cavities
- High quality EM resonators: Typical $Q_0 > 10^{10}$
- Over billions of cycles, large electric field generated
- Particle beam gains energy as it passes through



Particle Acceleration via SRF Cavities

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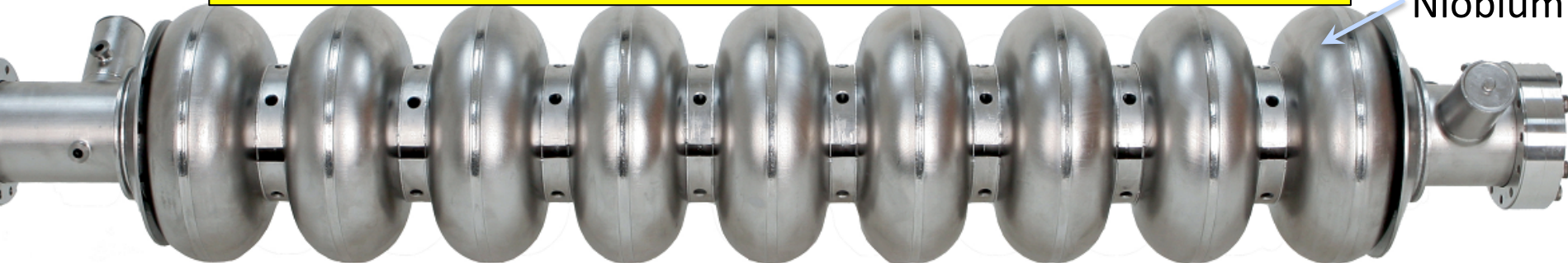
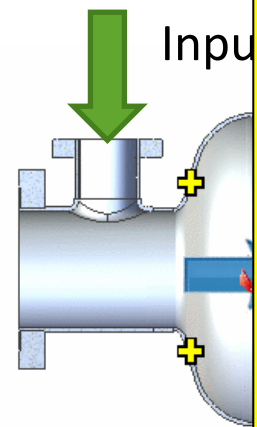


SRF: high current,
high energy, high
brightness beams

approximately 4×10^9

Animation by Sam Posen

Niobium

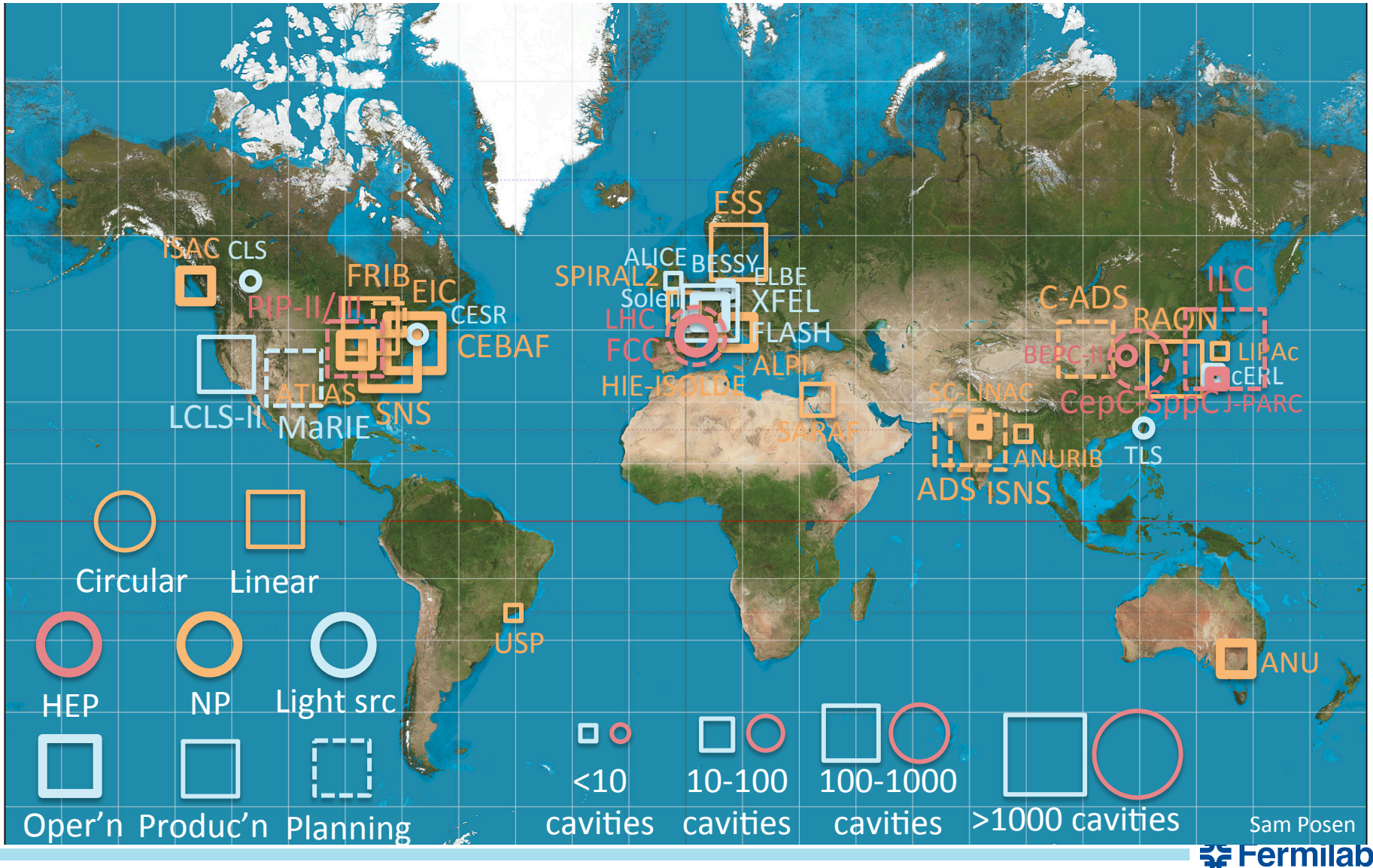


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~1 m

Images from linearcollider.org, Wikipedia

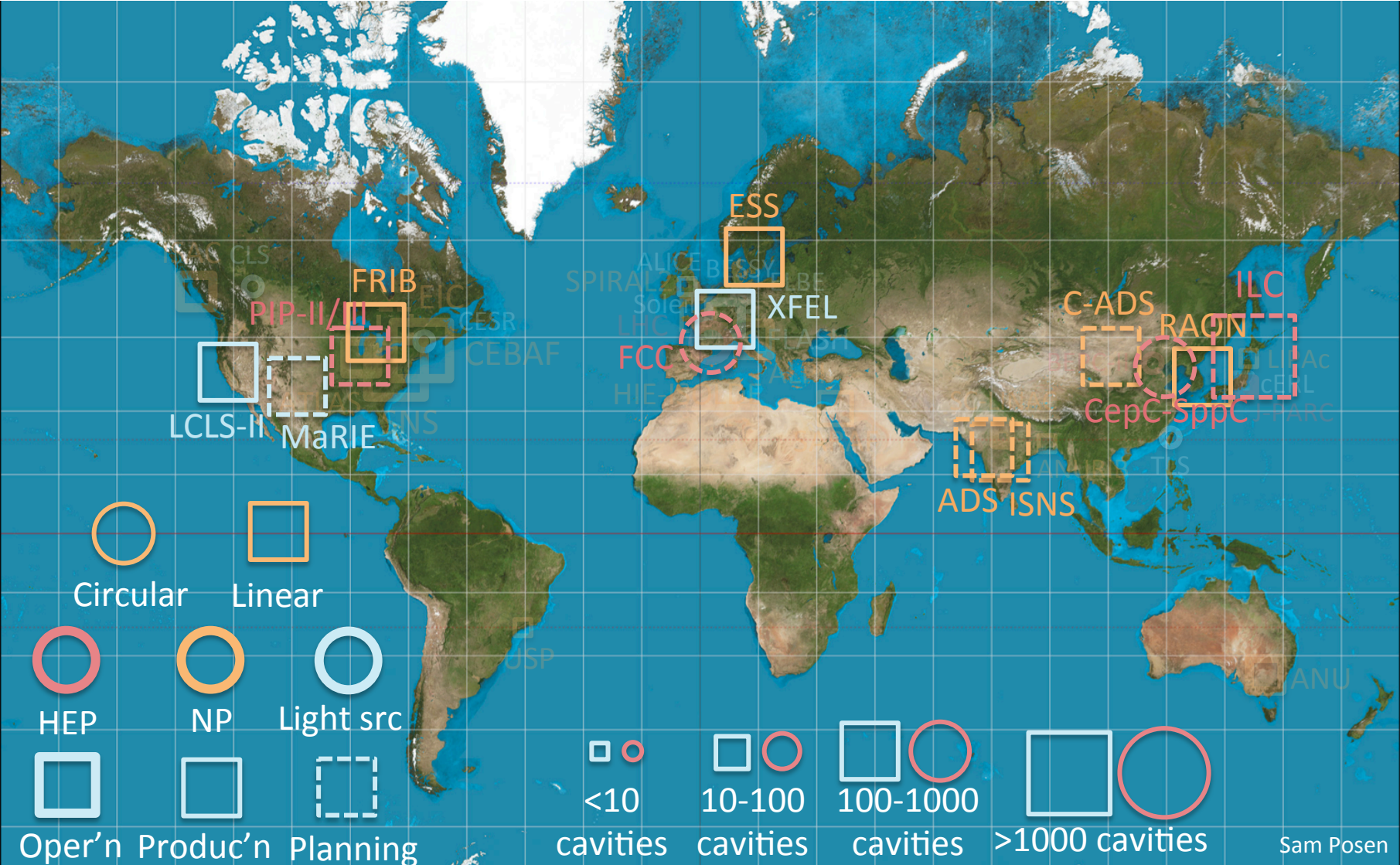
SRF Accelerators Around the World



Sam Posen



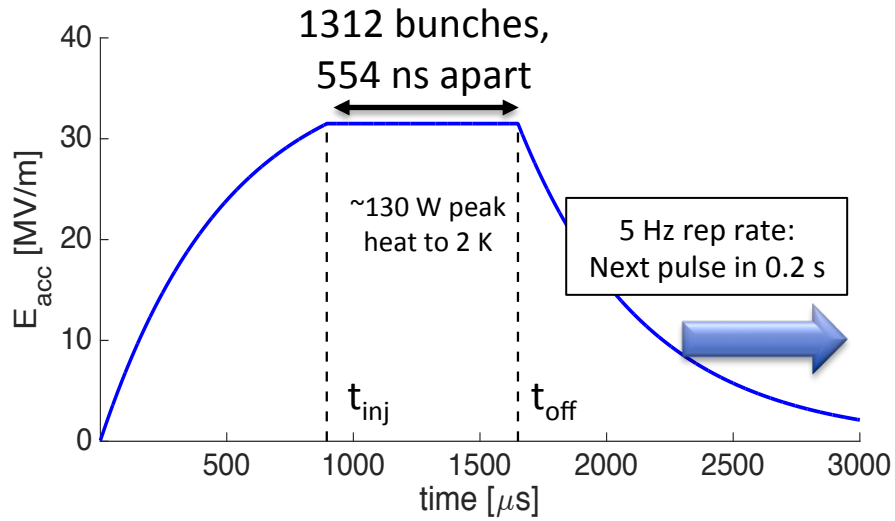
SRF Accelerators Around the World



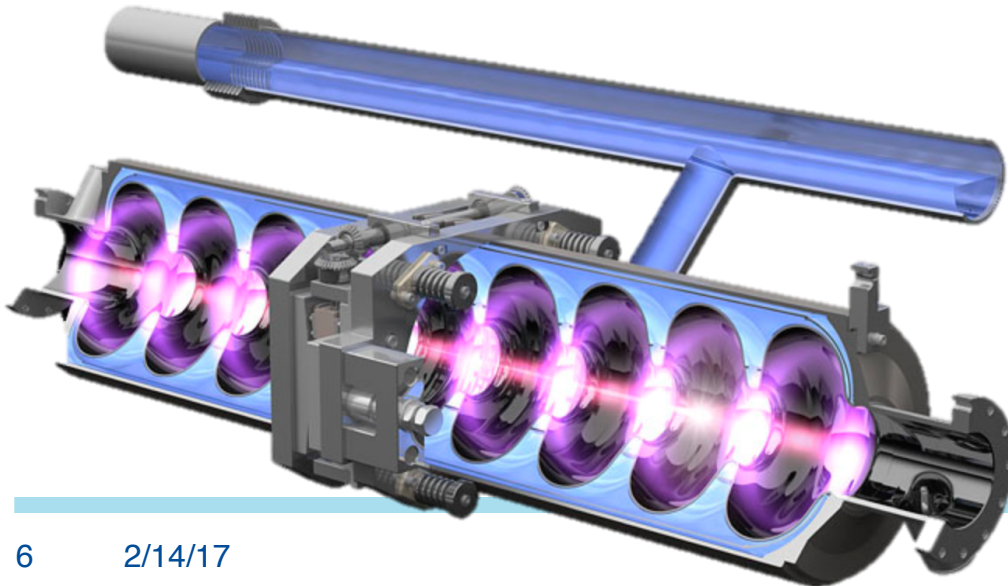
Sam Posen



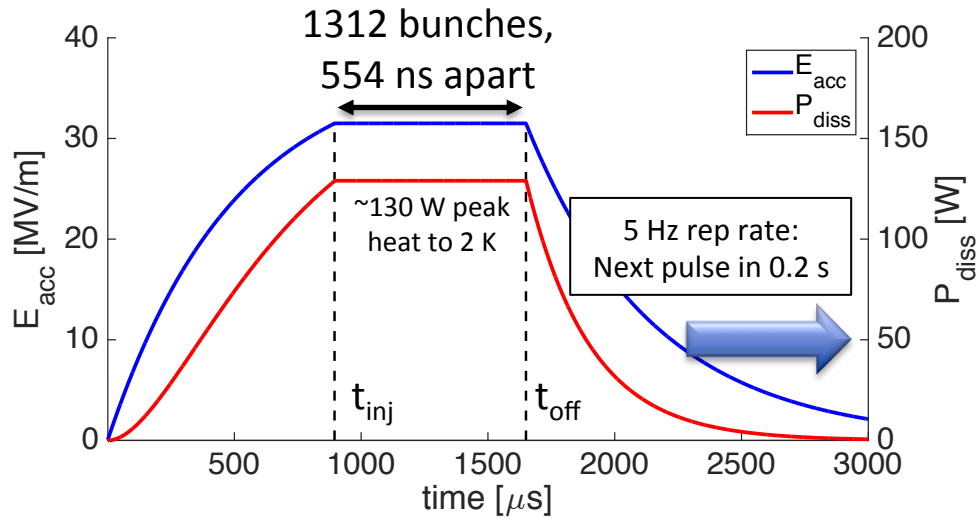
Duty Factor



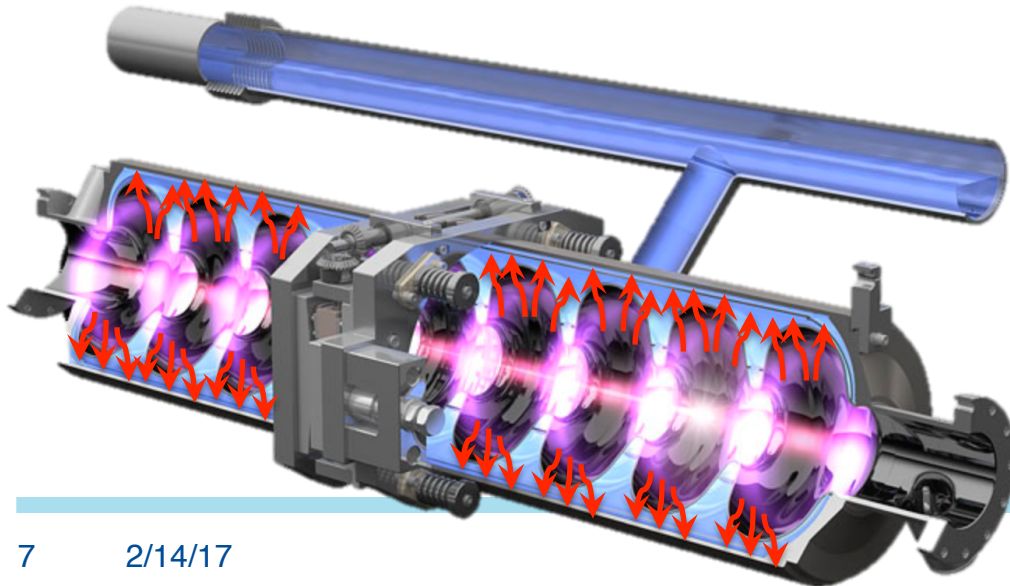
ILC (pulsed, <1% DF)
LCLS-II (CW, 100%DF)



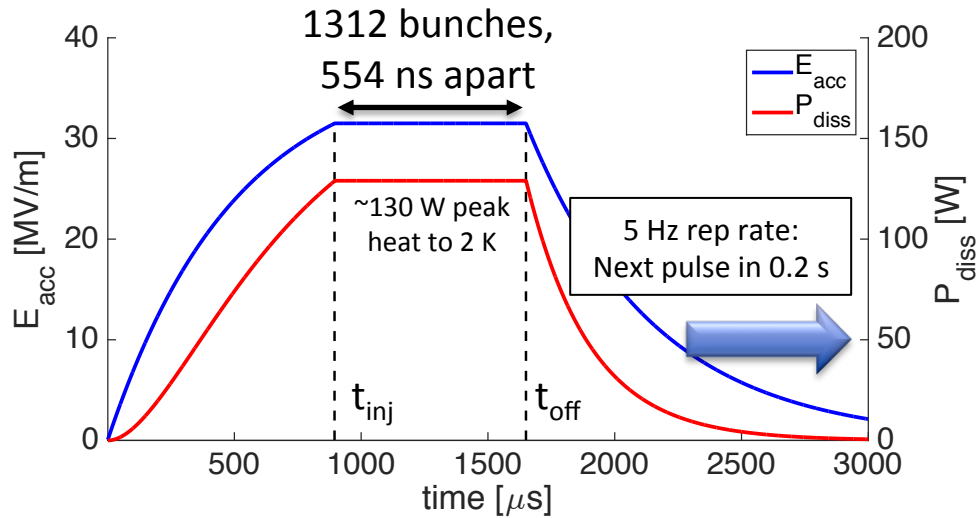
Duty Factor and Cryogenic Costs



ILC (pulsed, <1% DF) ~ 0.5 W average per cavity
 LCLS-II (CW, 100%DF) ~ 10 W average per cavity

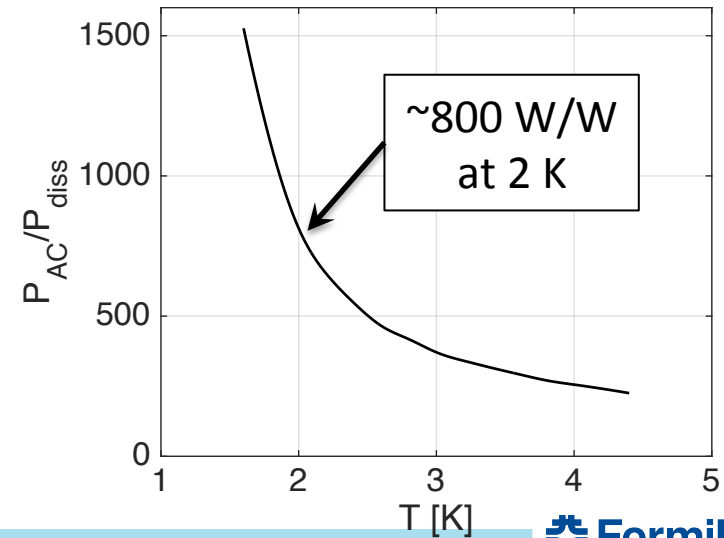
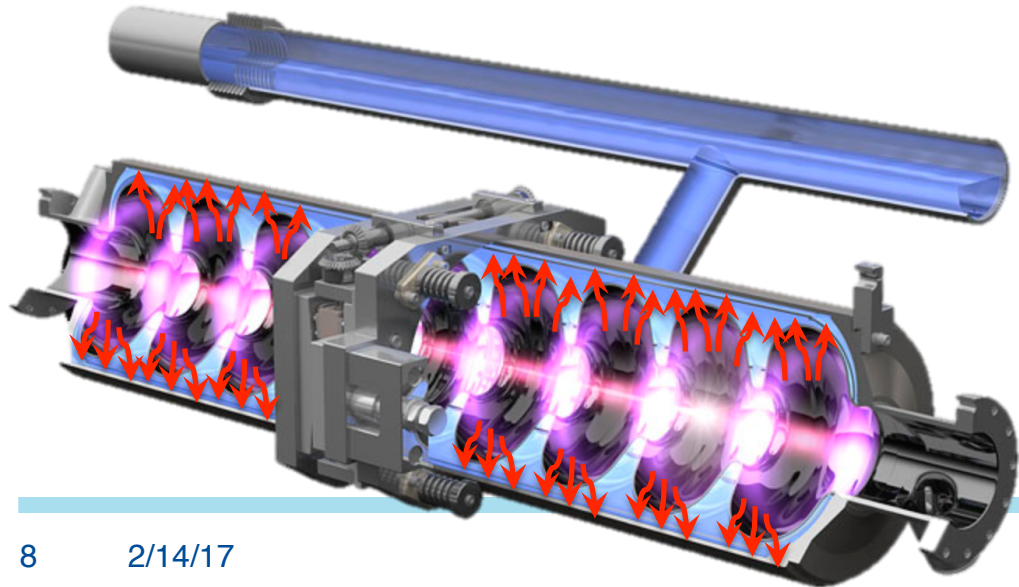


Duty Factor and Cryogenic Costs



ILC (pulsed, <1% DF) ~0.5 W average per cavity
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$$P_{diss} \sim E_{acc}/Q_0 \quad (\text{for fixed } E_{max})$$



$Q_0 \rightarrow$ Cryogenic Infrastructure, Operating Cost

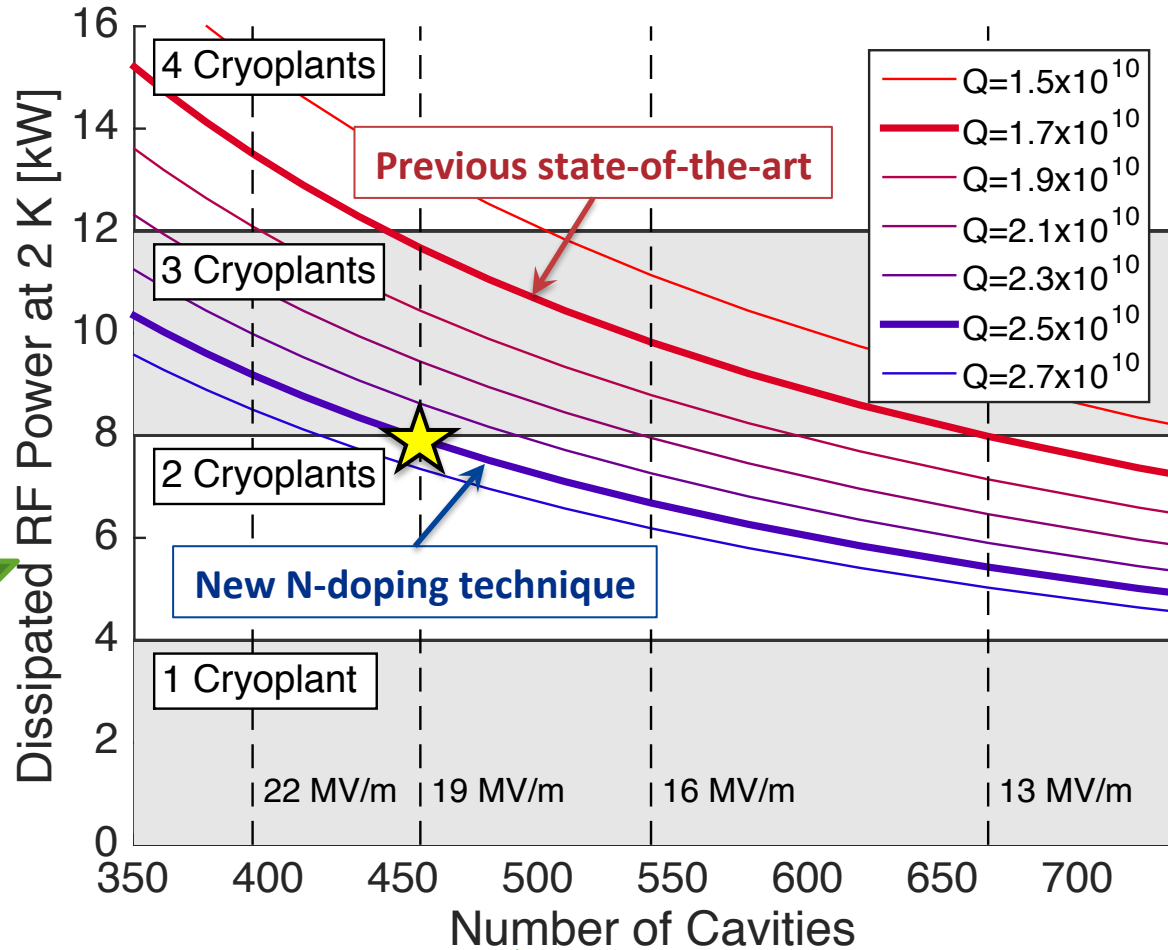
$$P_{diss} \sim E_{acc}/Q_0$$

(for fixed E_{max})

8 GeV CW SRF Linac



Lower cost

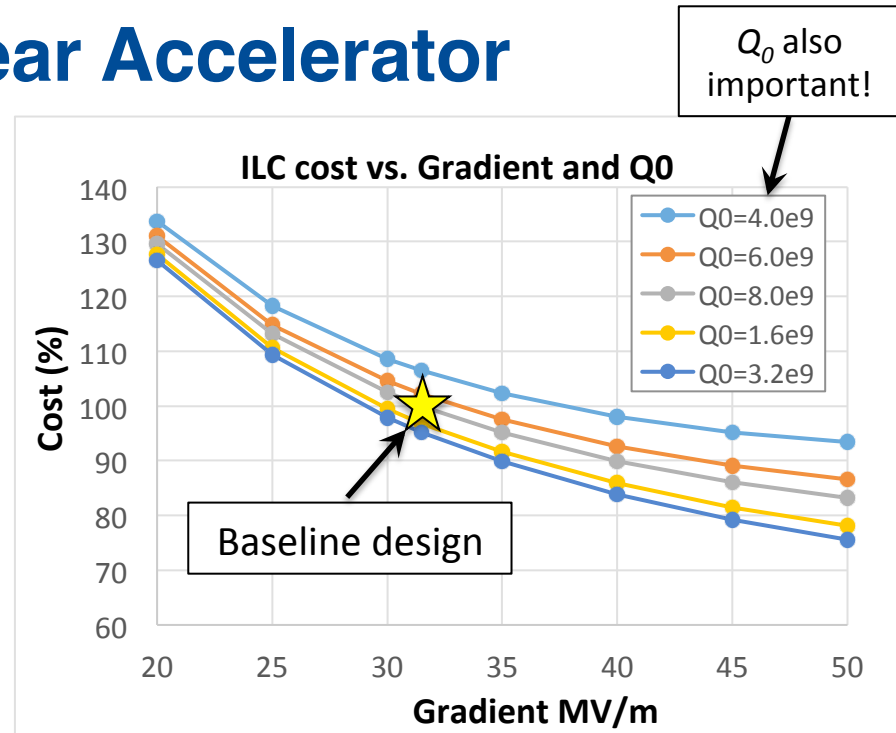
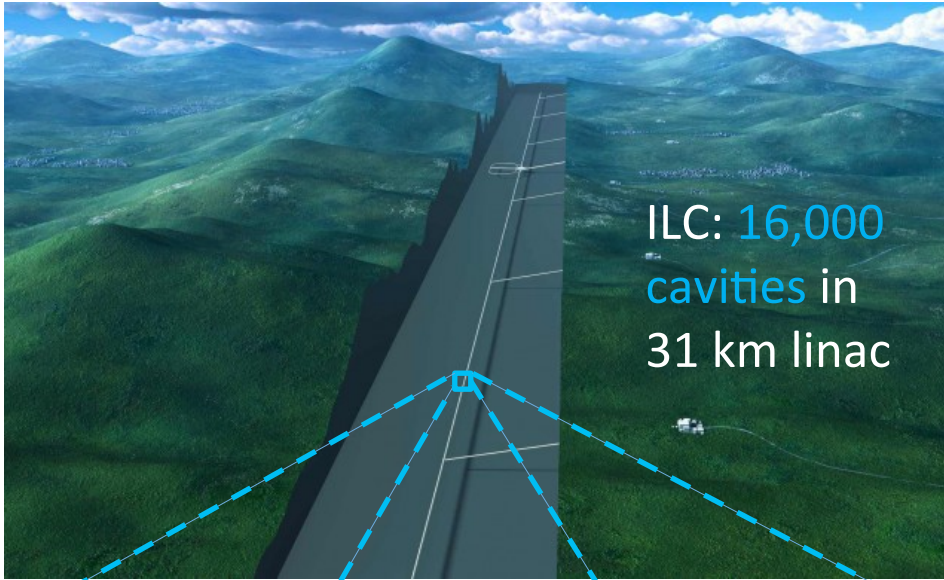


Lower cost



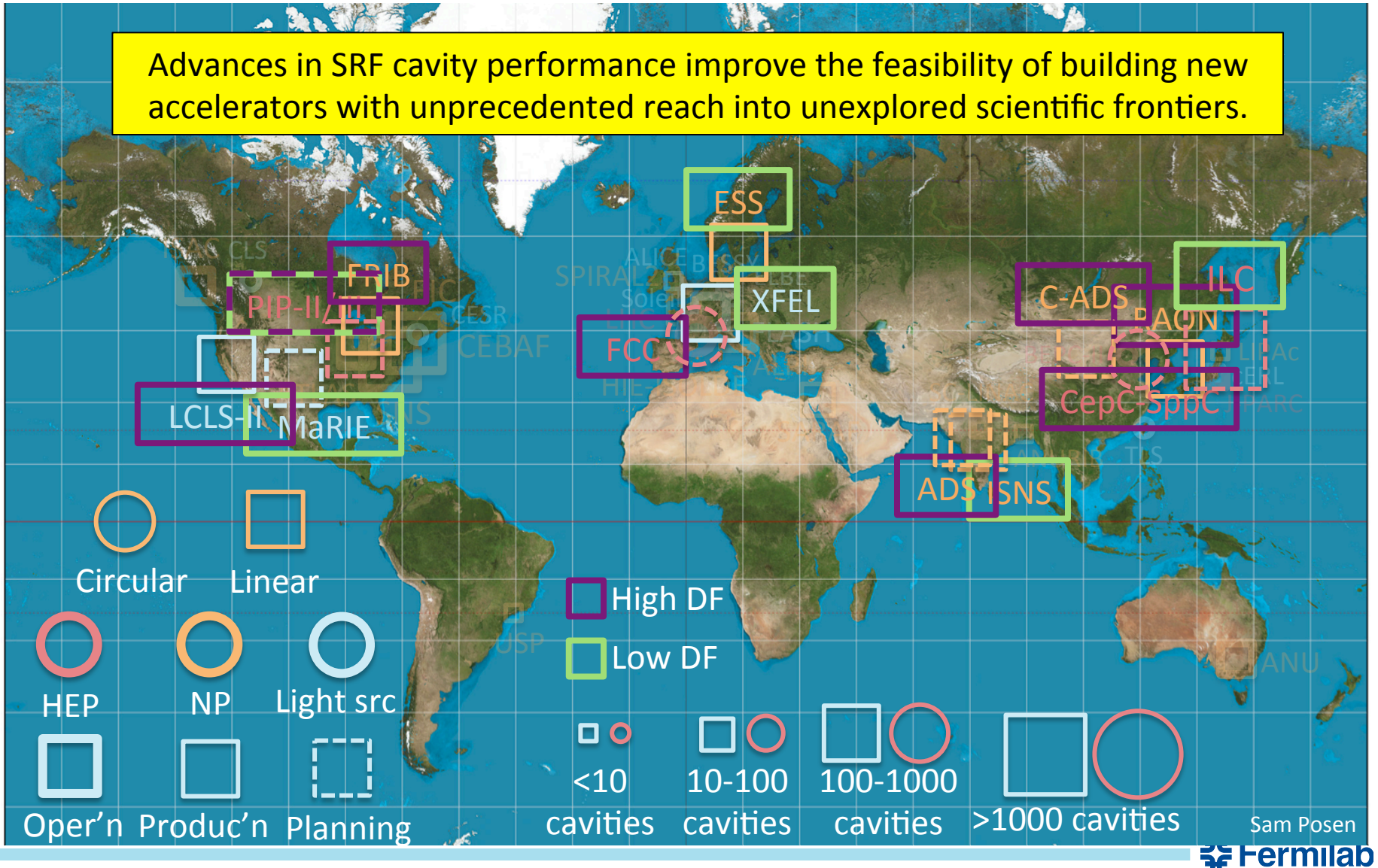
Above is a rough calculation for LCLS-II upgrade
(from me, not to be considered official numbers from project)

Gradient -> Length for Linear Accelerator



Motivation State-of-the-Art SRF Technology

Advances in SRF cavity performance improve the feasibility of building new accelerators with unprecedented reach into unexplored scientific frontiers.



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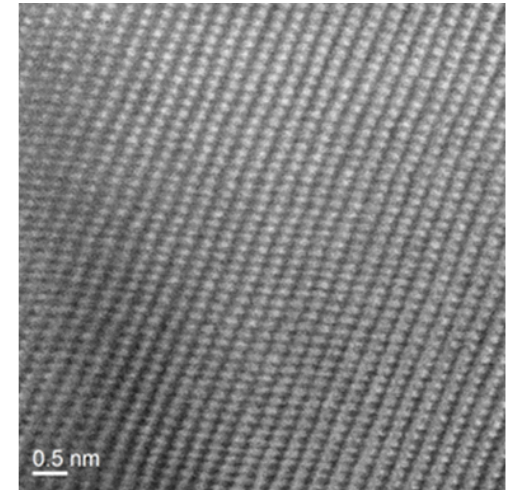


Outline

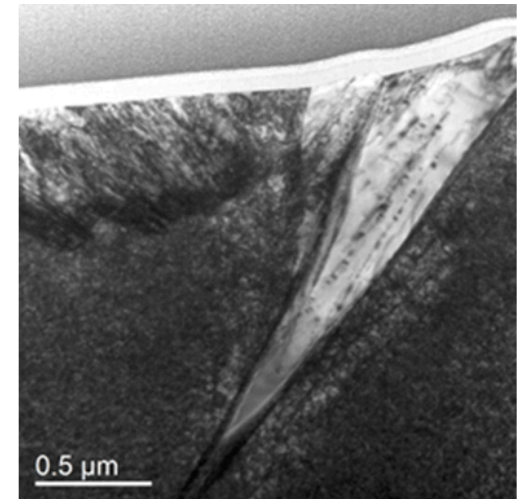
- High Q_0 (medium E_{acc})
 - Nitrogen doping
 - Flux expulsion
- High E_{acc} and high Q_0
 - Nitrogen infusion
 - Nb_3Sn
 - Plasma cleaning



ILC cryomodule with record average gradient for FAST facility



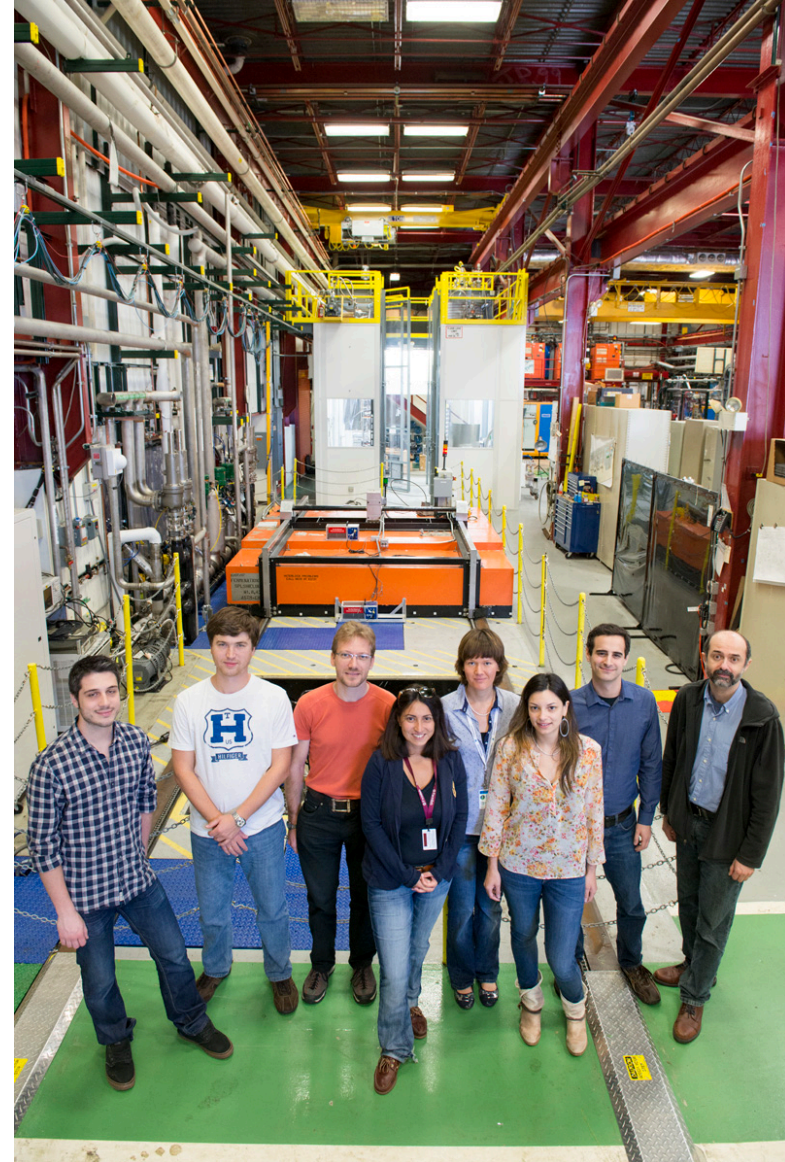
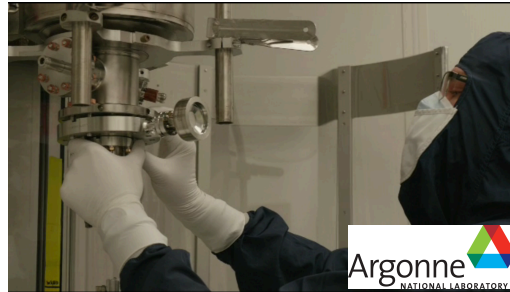
Aberration-corrected STEM image of the near-surface of Nb cavity - Y. Trenikhina



Niobium near-surface after the first step of nitrogen doping - Y. Trenikhina

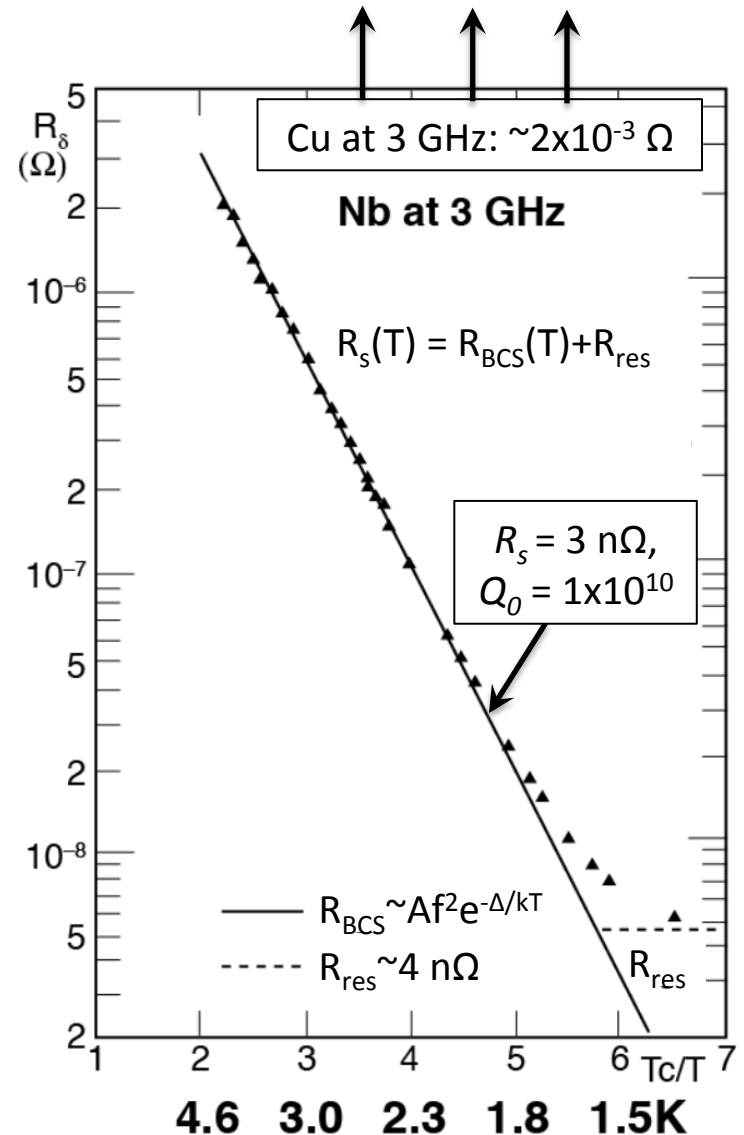
Team Effort

- Results shown here are due to many hardworking people
- Thanks to SRF department and Fermilab LCLS-II team for contributions



Note on Q_0 and BCS/Residual Surface Resistance

- Q_0 and R_s are related by a geometrical constant G : $Q_0 = G/R_s$
- They measure efficiency
- Heat dissipated in the walls of the cavity: $P_{diss} \sim R_s \sim Q_0^{-1}$
- R_s decreases exponentially with decreasing T/T_c but it saturates at low T : residual resistance R_{res}
- Generally we decompose R_s into temperature dependent $R_{BCS}(T)$ and temperature independent R_{res}
- Cavities often operate at ~ 2 K where both are significant



Treatment for Enhancement of RF Penetration Layer

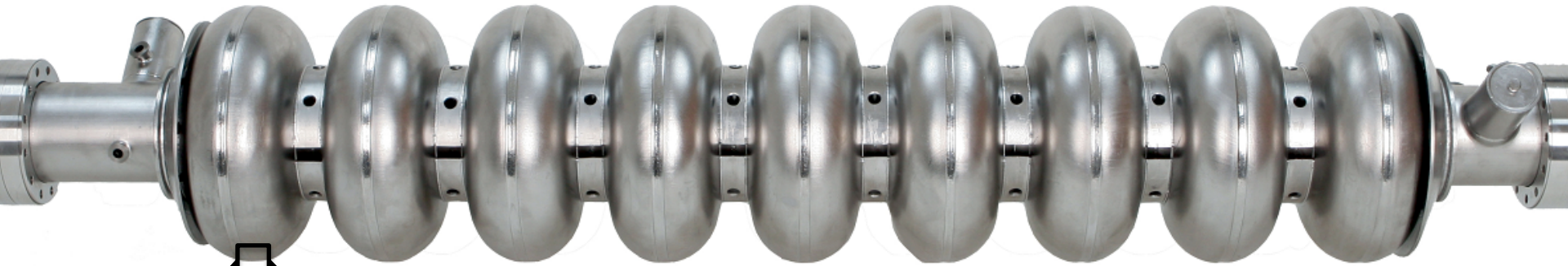
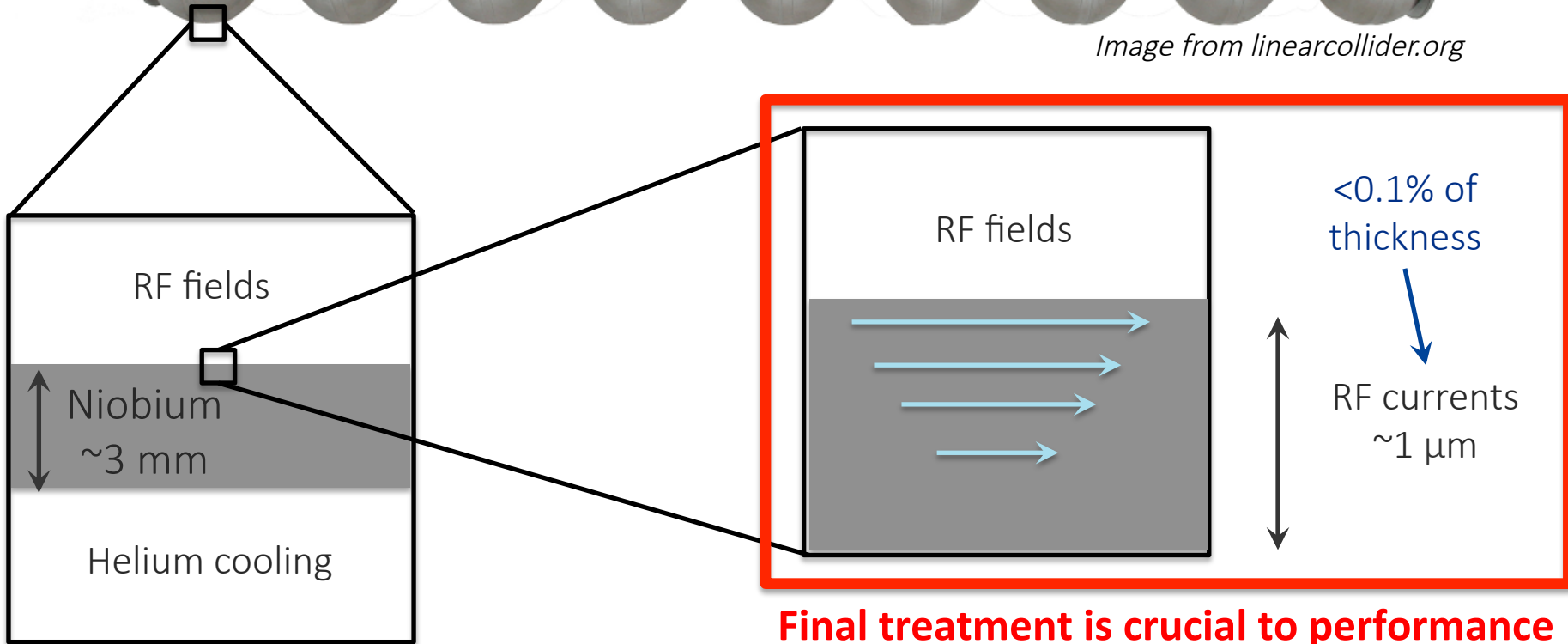
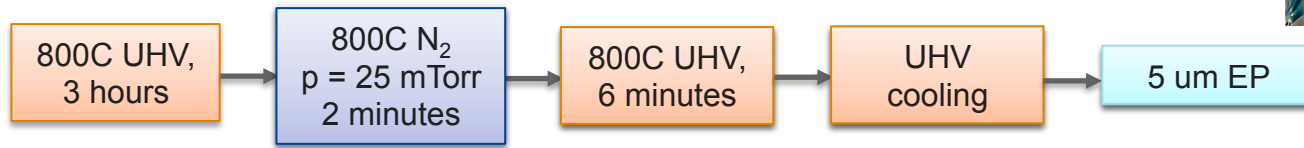
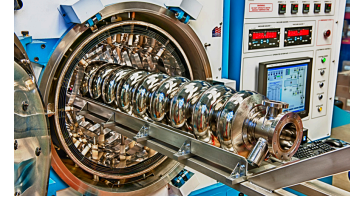


Image from linearcollider.org

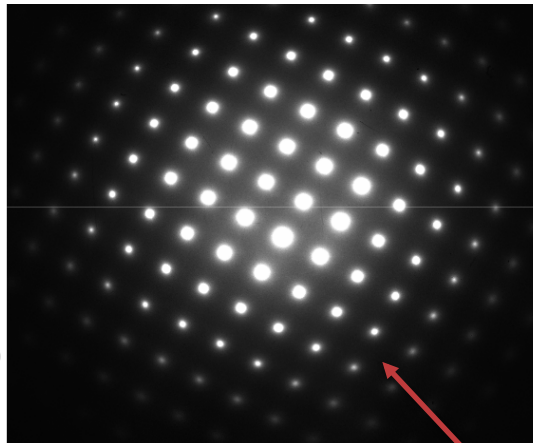


Final treatment is crucial to performance

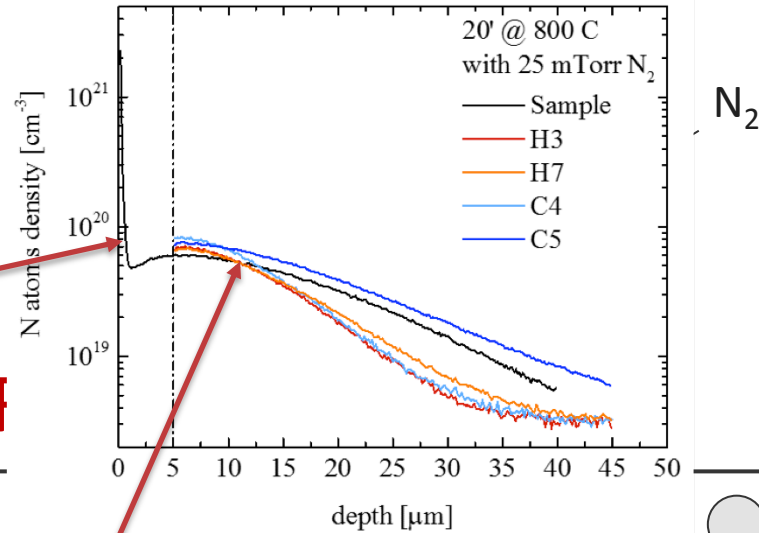
N-doping treatment



Y. Trenikhina et al, Proc. of SRF 2015



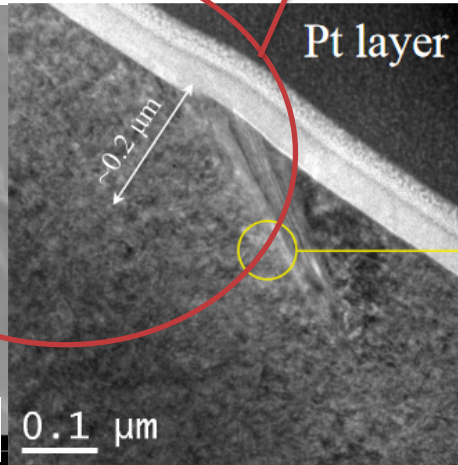
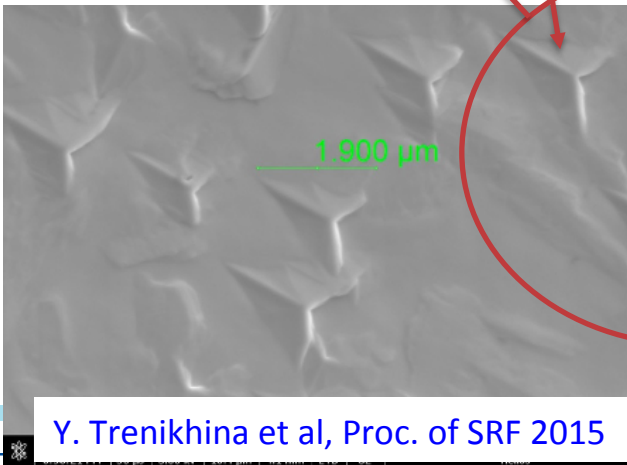
Final RI



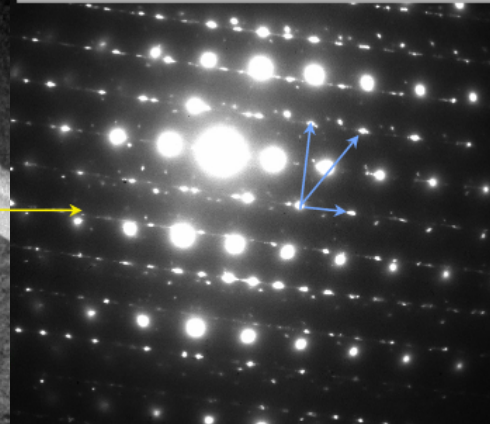
N₂

Nb_xN_y

N Interstitial

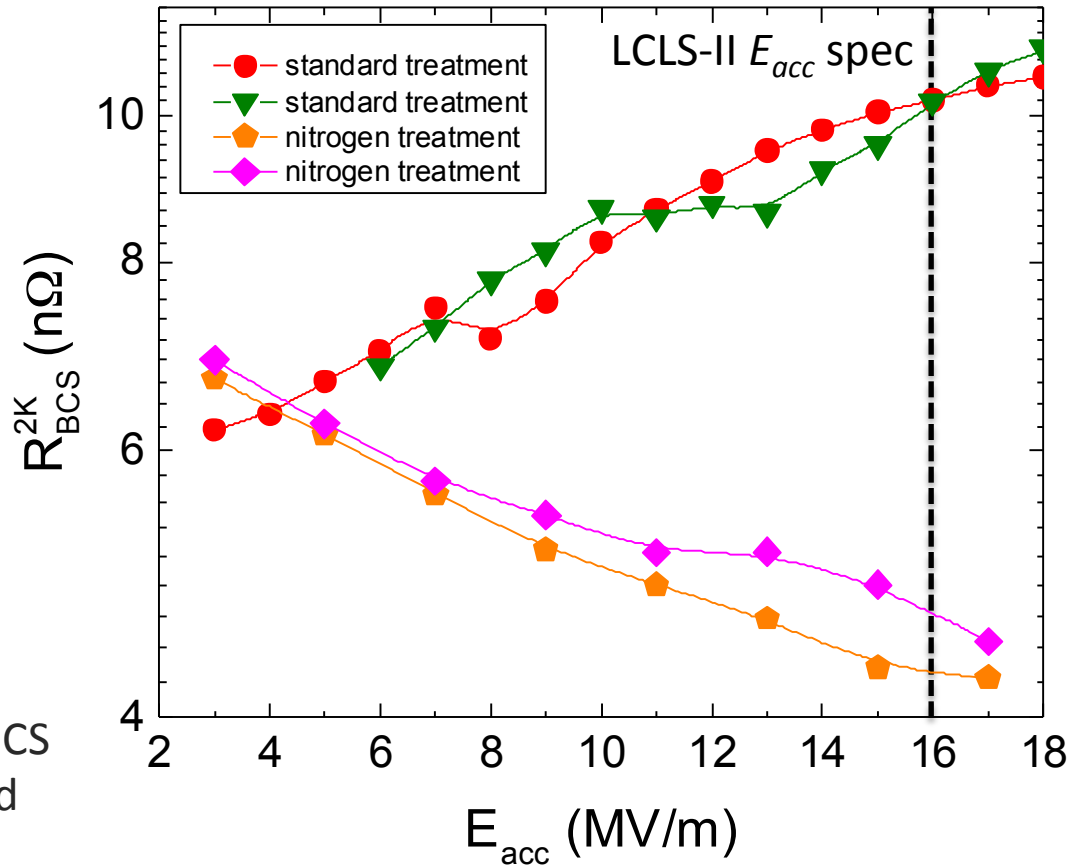
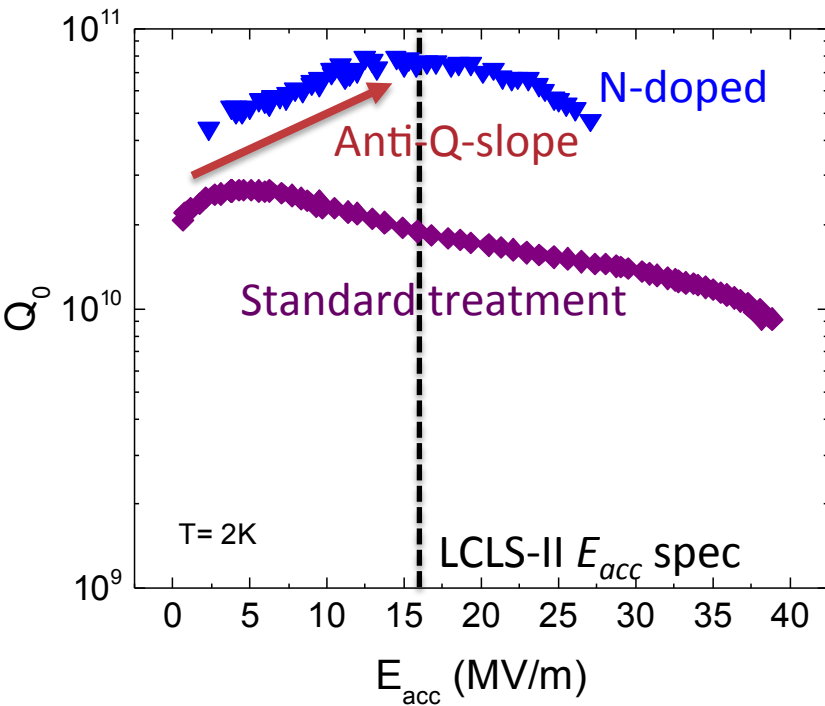


Nb [113]+Nb₂N [210]+?



N

Effect on Surface Resistance (and therefore $Q_0=G/R_s$)

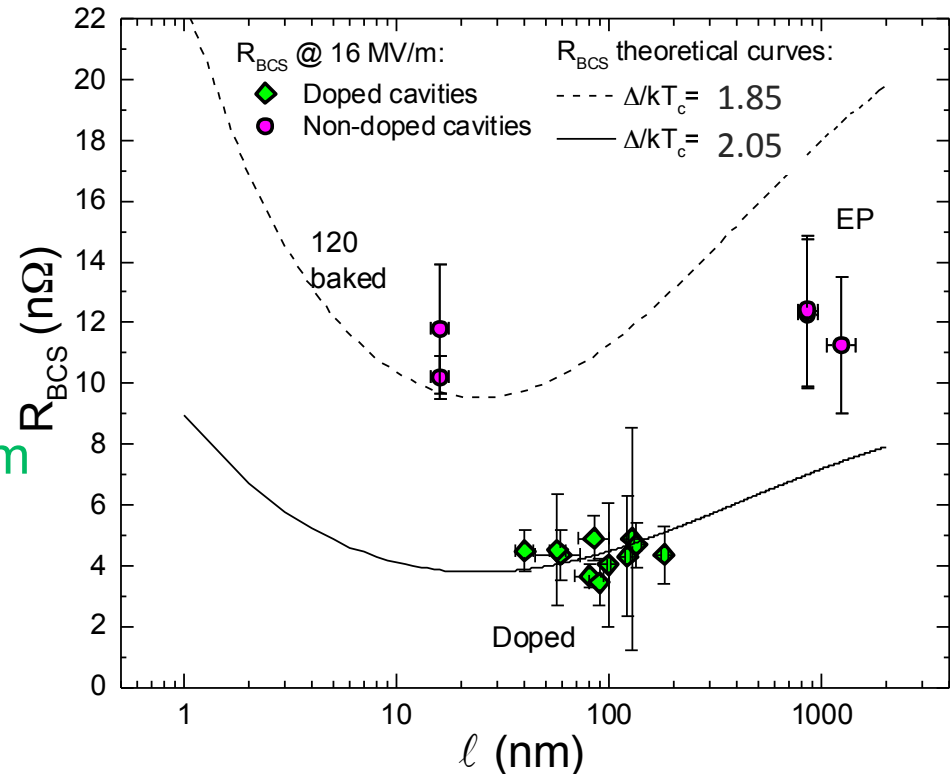
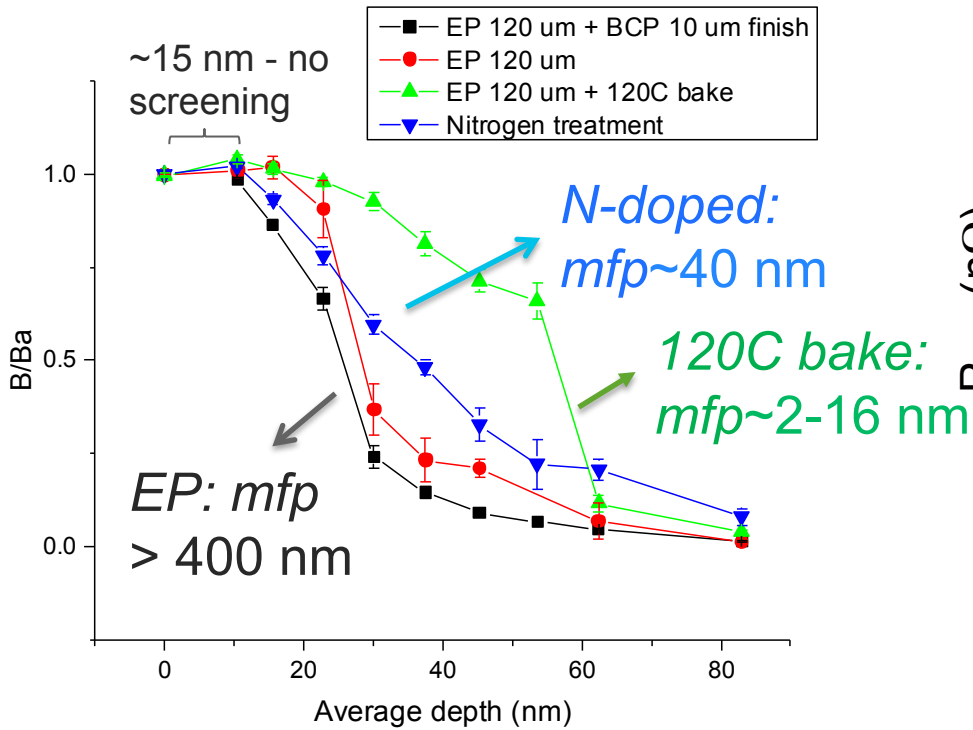


Anti-Q-slope emerges from the BCS surface resistance decreasing with field
 → **Unexpected, unprecedented**

- $>2x R_{BCS}$ improvement at 2 K, 16 MV/m
- Reduced maximum field OK for high duty factor applications

Origin of Improved Surface Resistance due to N-Doping

LE- μ SR measurements (Ba=25mT)



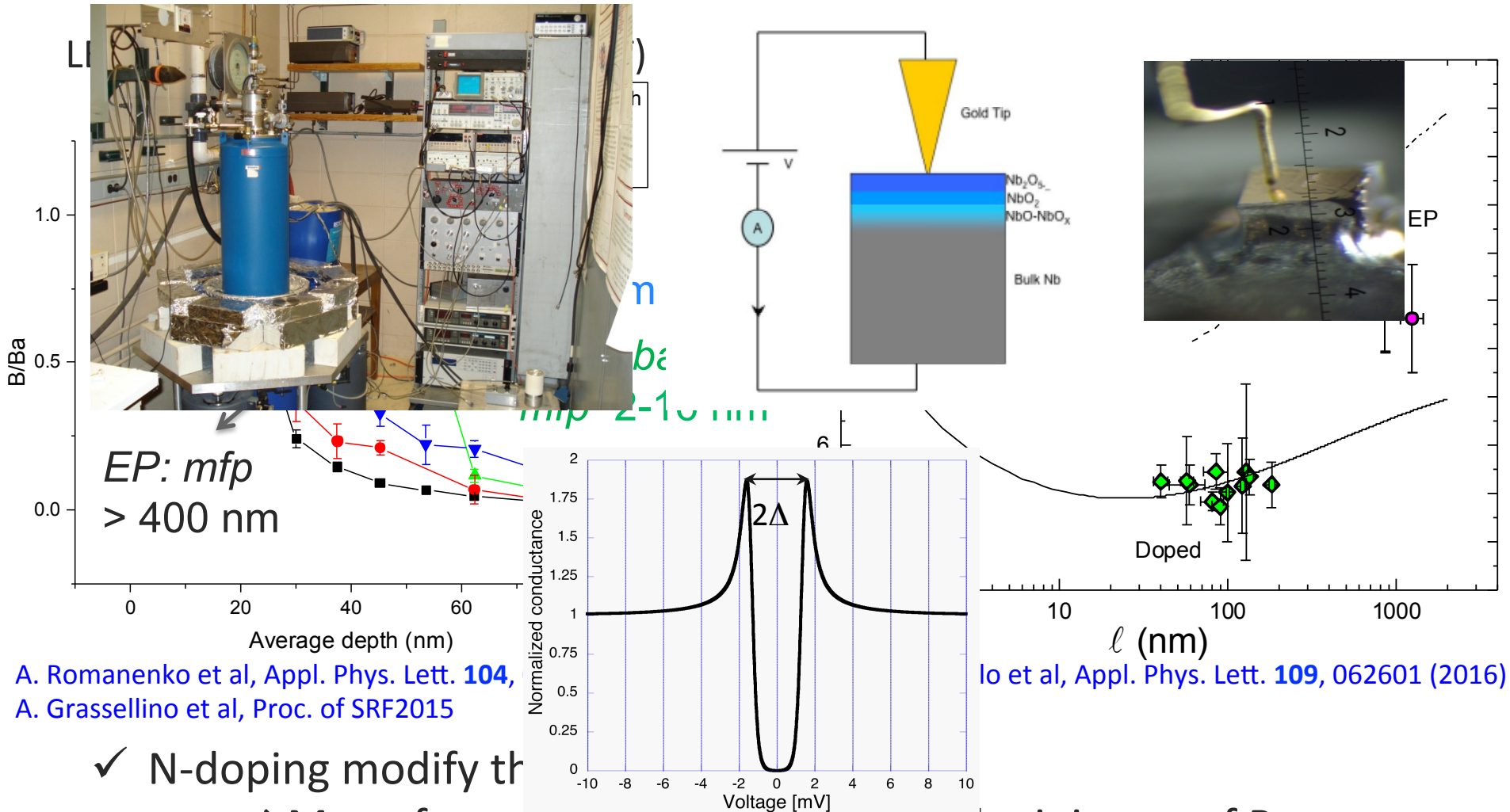
A. Romanenko et al, Appl. Phys. Lett. **104**, 072601 (2014)

M. Martinello et al, Appl. Phys. Lett. **109**, 062601 (2016)

A. Grassellino et al, Proc. of SRF2015

- ✓ N-doping modify the mean free path
 → Mean free path close to theoretical minimum of R_{BCS}
- ✓ N-doping seems to increase the reduced energy gap $\Delta/k_B T_c$

Origin of Improved Surface Resistance due to N-Doping

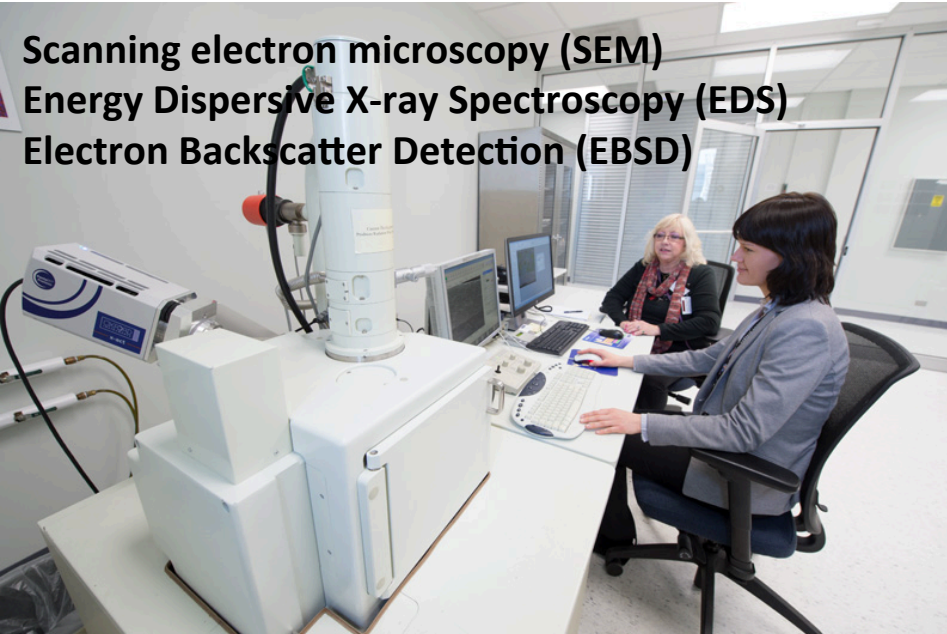
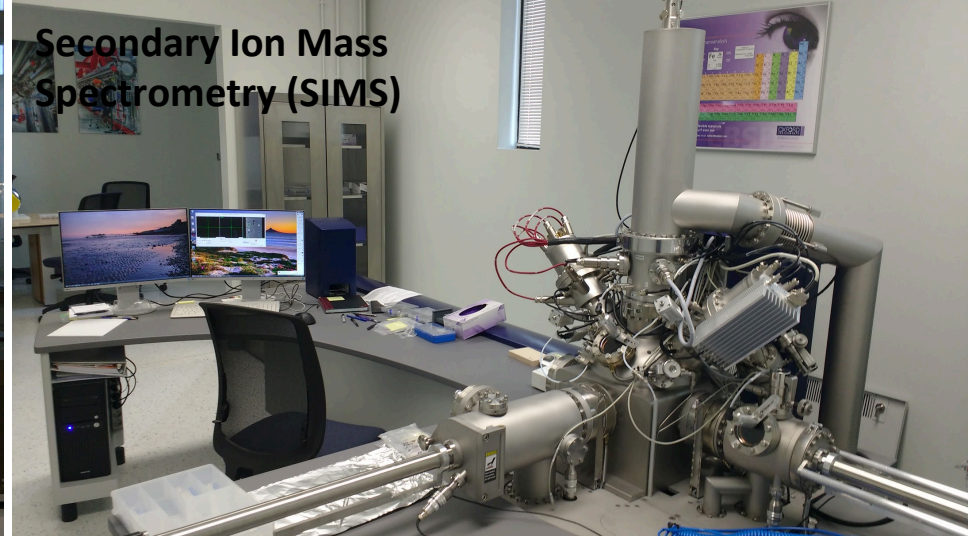


A. Romanenko et al, *Appl. Phys. Lett.* **104**,
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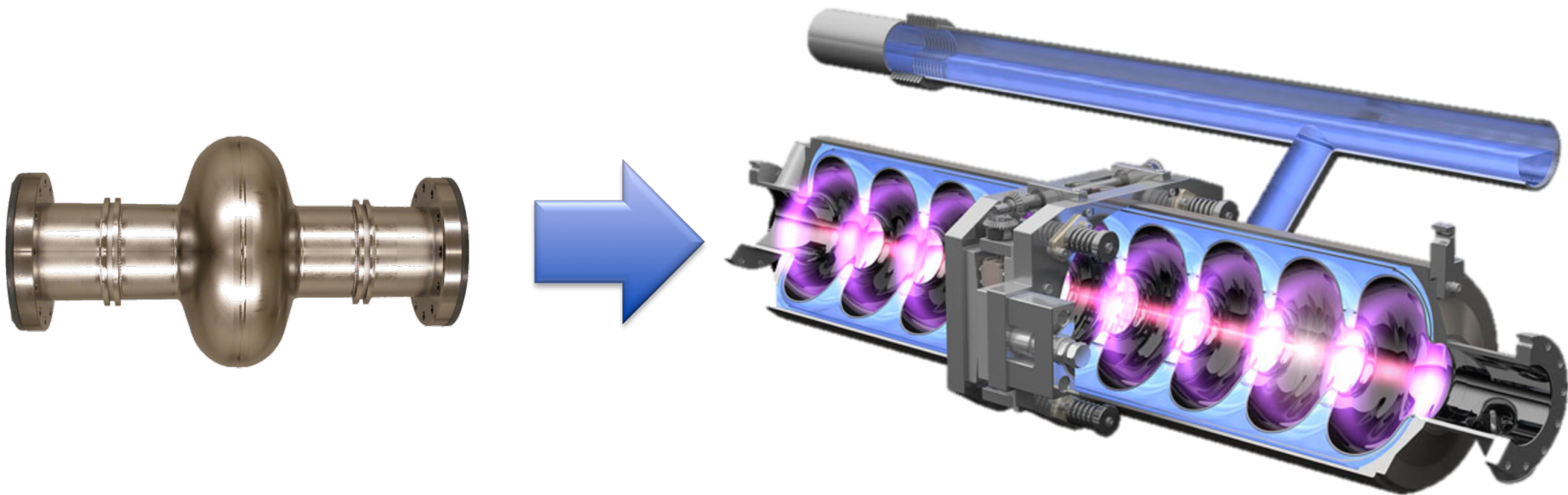
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Materials Science Lab

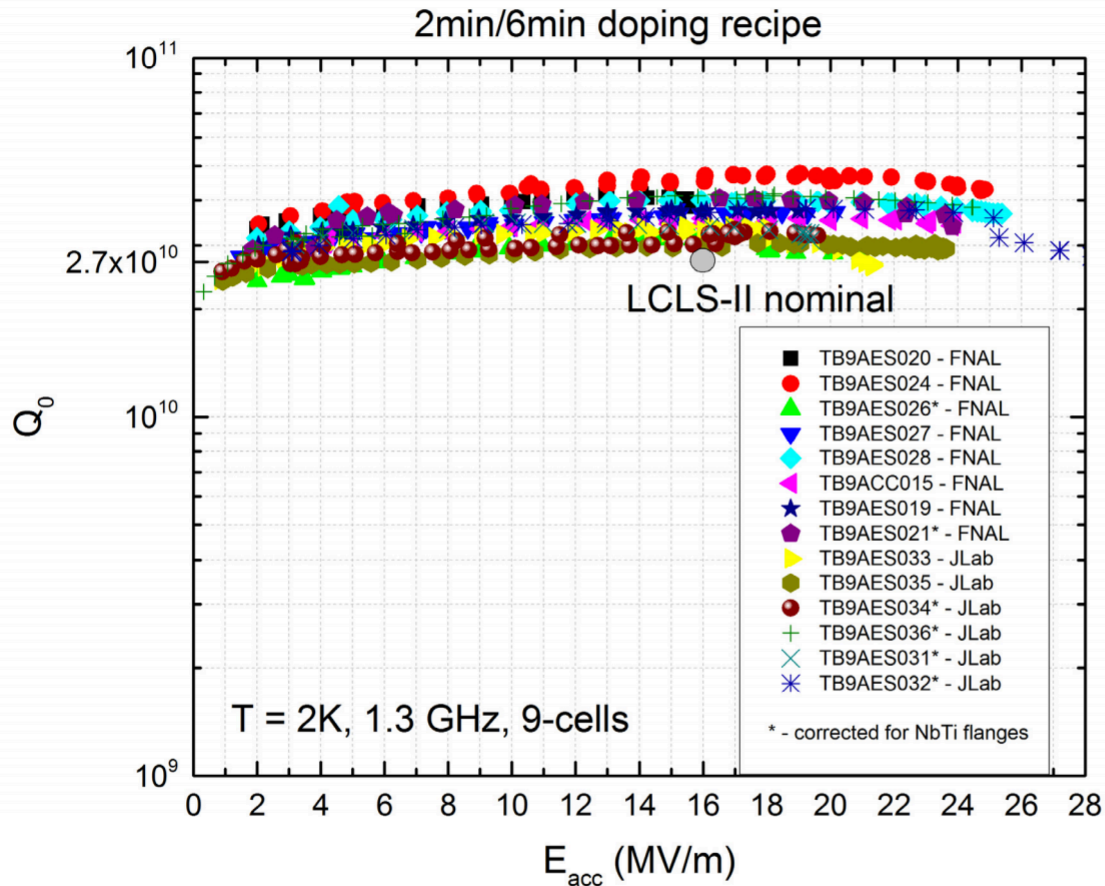


Scale-Up

- Initial studies on single cell R&D-style cavities
- Several milestones required to demonstrate technology maturity for accelerator applications



From single cell R&D to cryomodule ready technology: the two LCLS-II prototype cryomodules (FNAL and Jlab)



$$\langle Q \rangle = 3.6e10$$

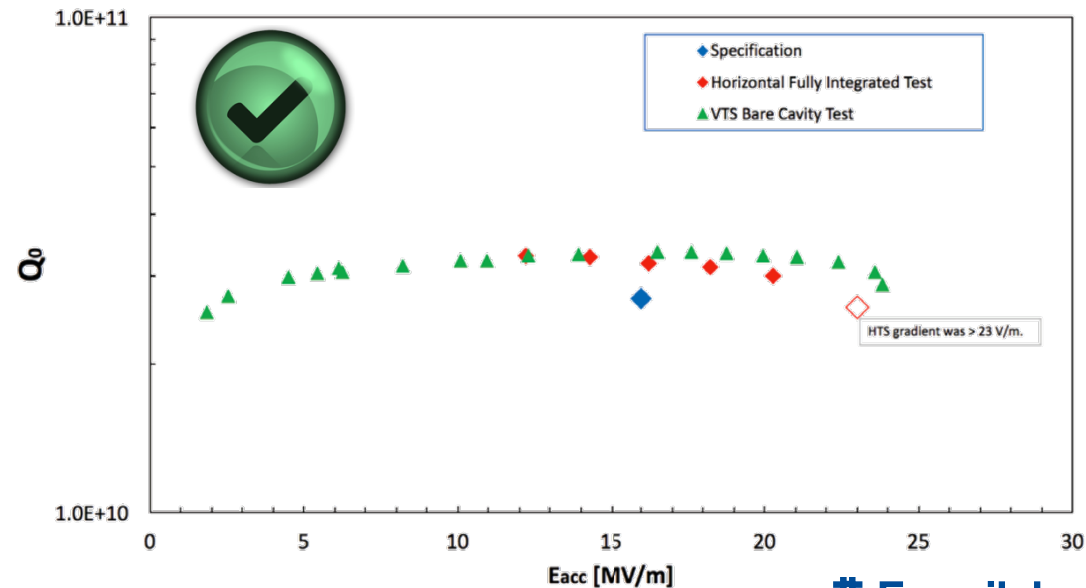
$$\langle E_{max} \rangle = 22.2 \text{ MV/m}$$

$$E_{max} \text{ median} = 22.8 \text{ MV/m}$$

It is the highest average Q ever demonstrated in vertical test for 1.3 GHz nine cells at 2K, 16 MV/m in the history of SRF (larger than a factor of two the state of the art)

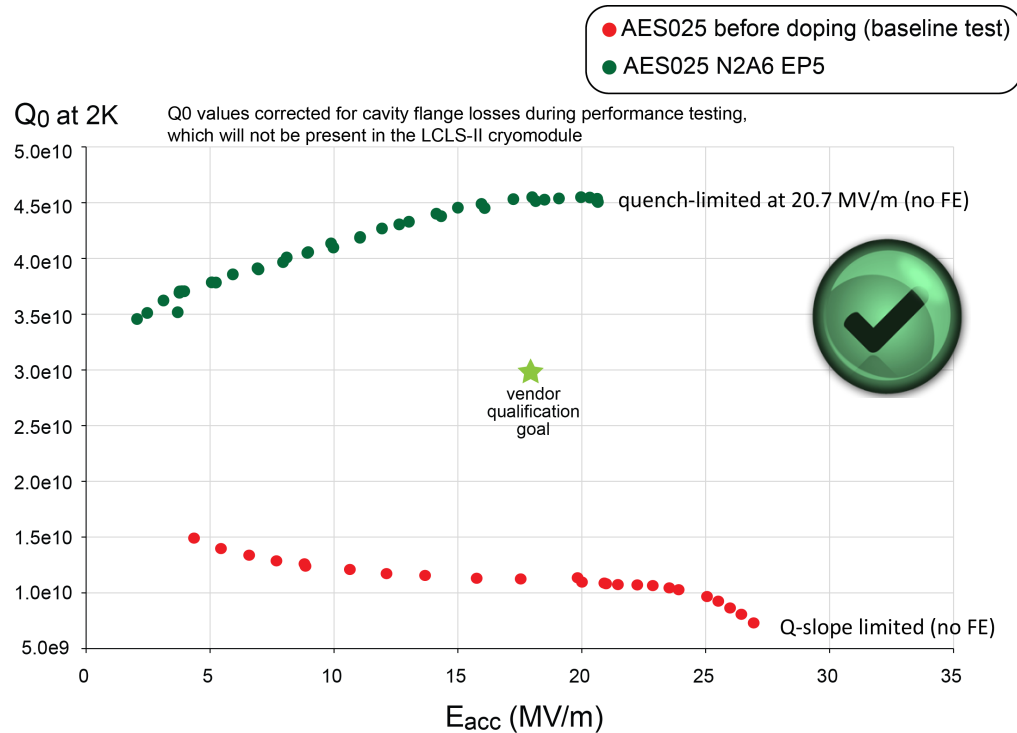
Integrated Testing

- Record $Q_0 > 3 \times 10^{10}$ at 16MV/m, 2 K for a LCLS-II dressed cavity in fully integrated horizontal test (with high power coupler, HOMs, tuner etc.)
- Important milestone to show new readiness for implementation in application



Technology Transfer

- SRF cavity vendors: from niobium material to N-doped cavities ready for qualification testing



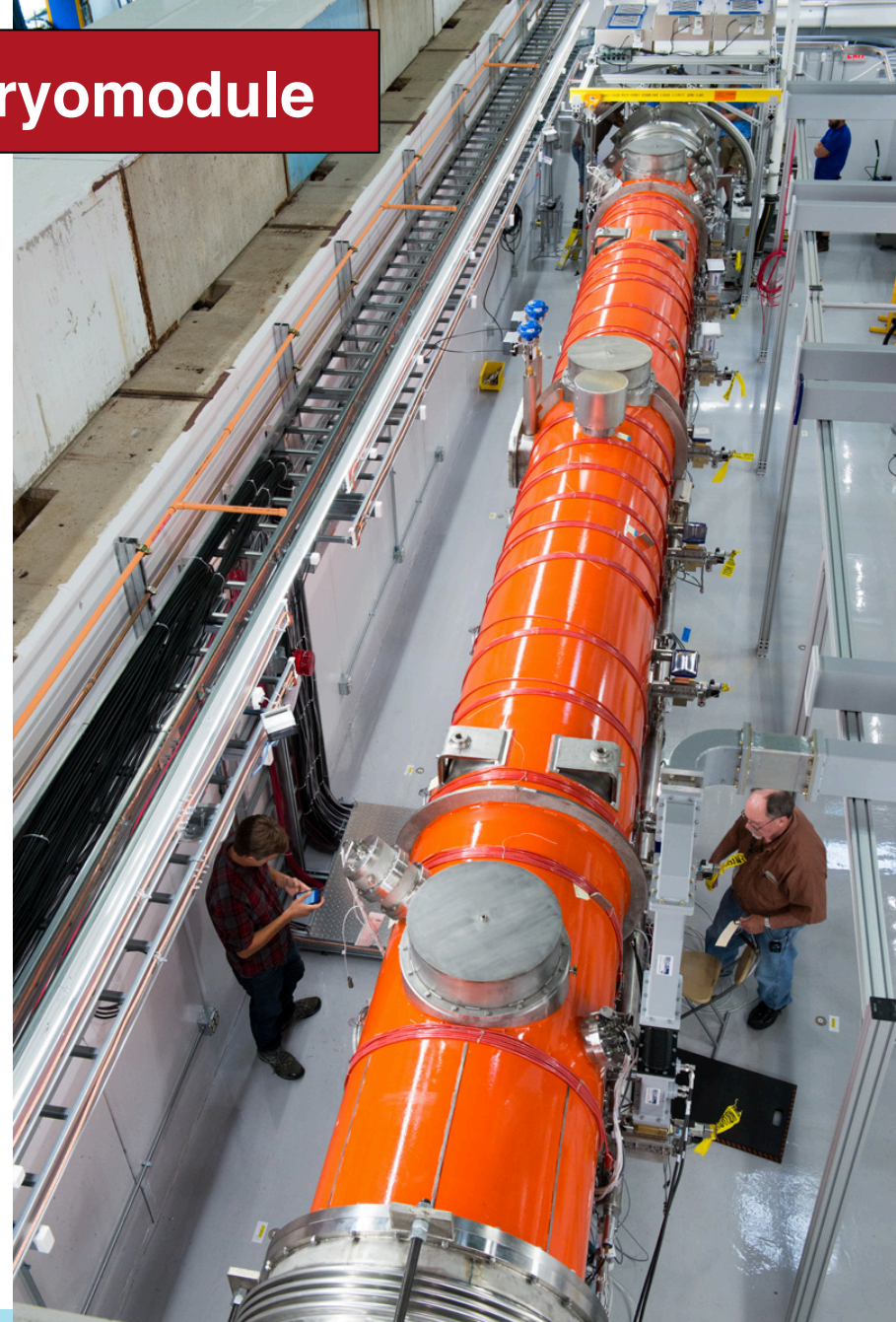
Fermilab Prototype LCLS-II Cryomodule

Cavity	Usable Gradient* [MV/m]	Q0 @16MV/m* 2K Fast Cool Down
TB9AES021	18.2	2.6E+10
TB9AES019	18.8	3.1E+10
TB9AES026	19.8	3.6E+10
TB9AES024	20.5	3.1E+10
TB9AES028	14.2	2.6E+10
TB9AES016	16.9	3.3E+10
TB9AES022	19.4	3.3E+10
TB9AES027	17.5	2.3E+10
Average	18.2	3.0E+10
Total Voltage	148.1 MV	

Spec:
133 MV



Spec:
 2.7×10^{10}

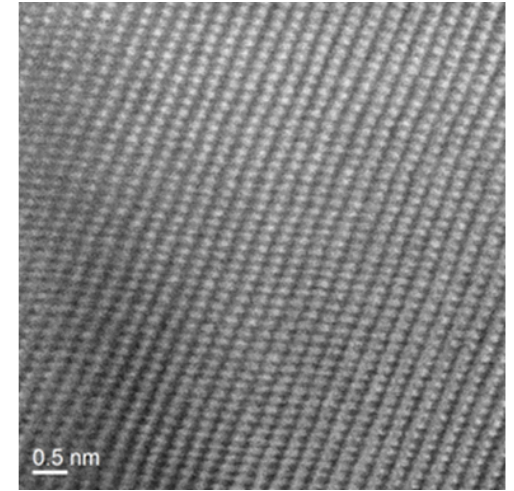


Outline

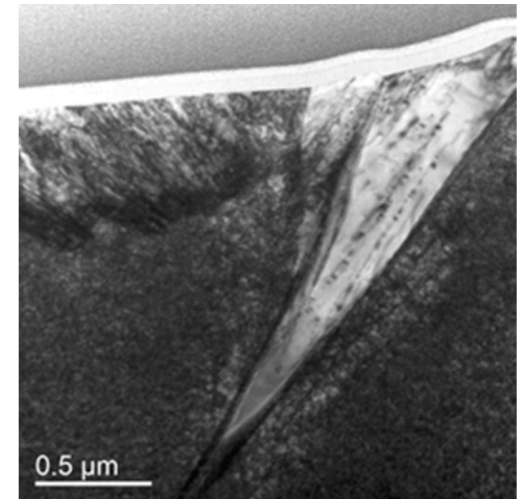
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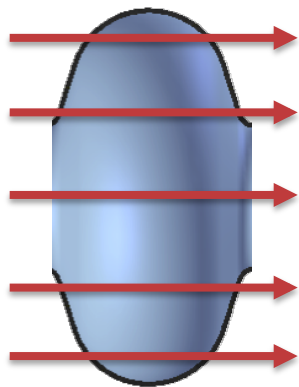
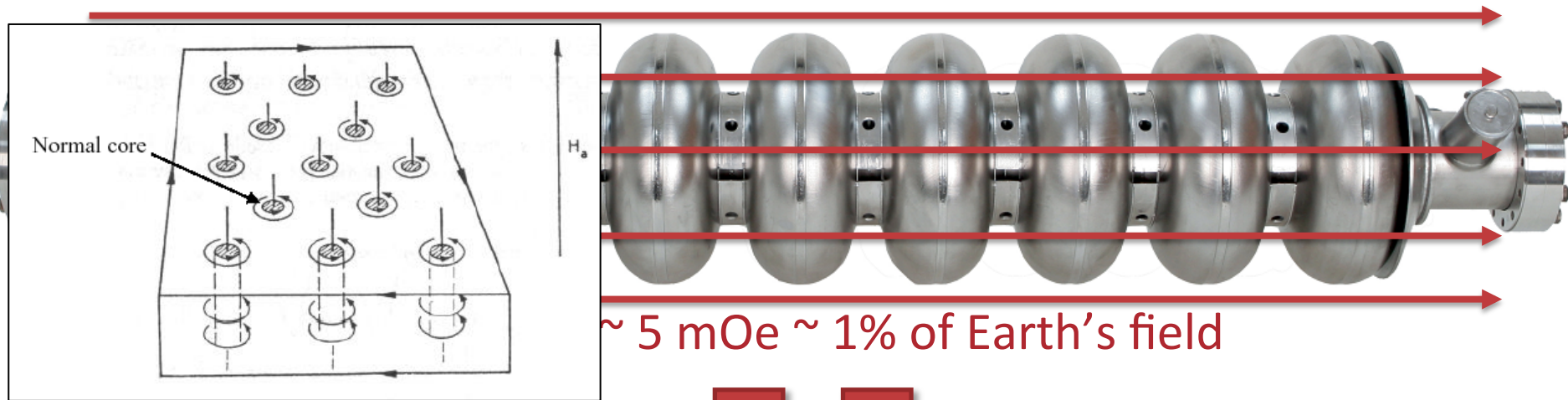


Aberration-corrected STEM image of the near-surface of Nb cavity - Y. Trenikhina

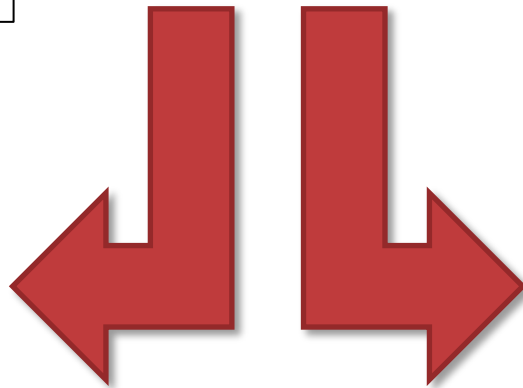


Niobium near-surface after the first step of nitrogen doping - Y. Trenikhina

Cooldown through Critical Temperature

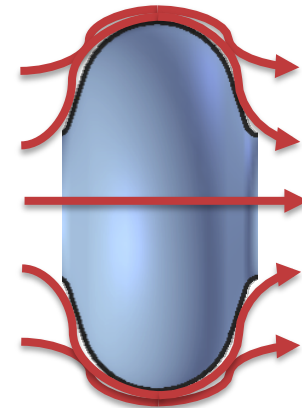


Magnetic Flux Trapping



Trapped flux increases R_{res}

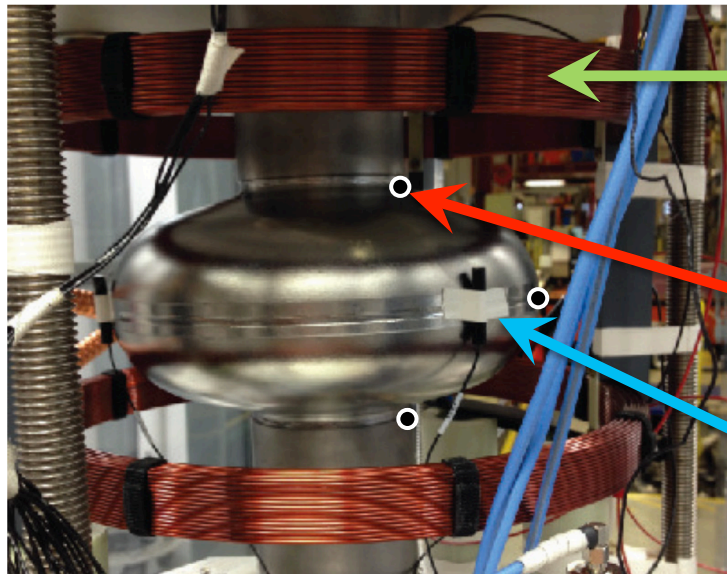
Closer to intrinsic R_{res}



Magnetic Flux Expulsion

Measuring Flux Expulsion

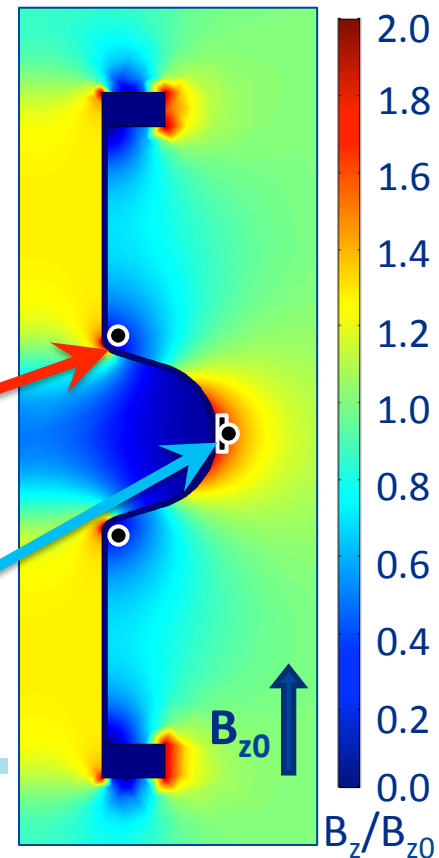
- An axial magnetic field is applied during cooldown. Fluxgate magnetometers at the equator measured the magnetic field before B_{NC} and after B_{SC} superconducting transition.
 - Complete trapping: $B_{SC}/B_{NC} = 1$
 - Complete expulsion: $B_{SC}/B_{NC} \sim 1.7$



External field coils

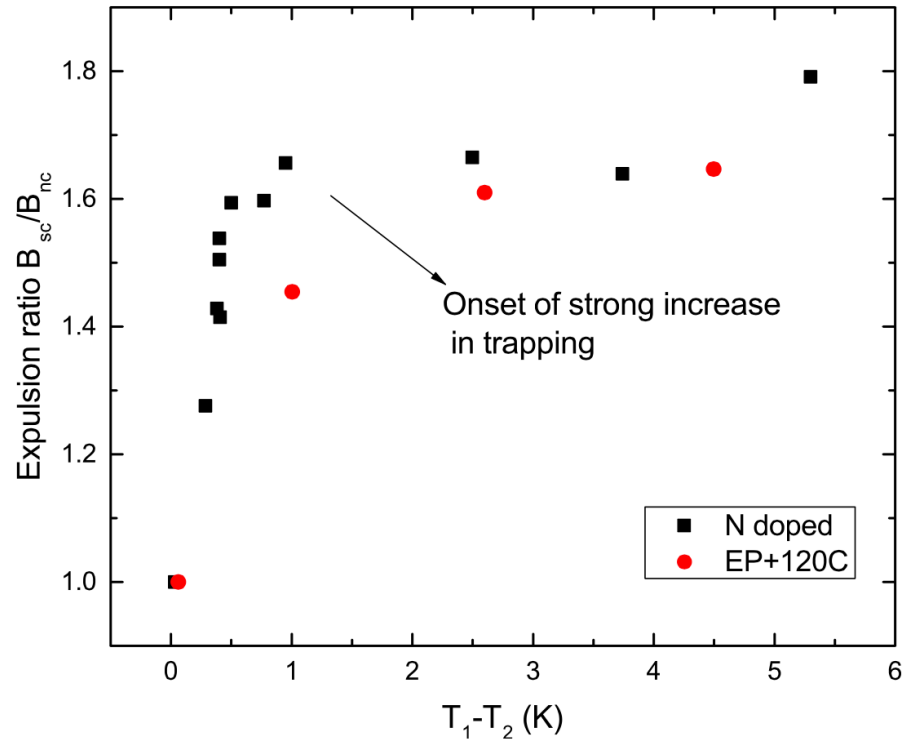
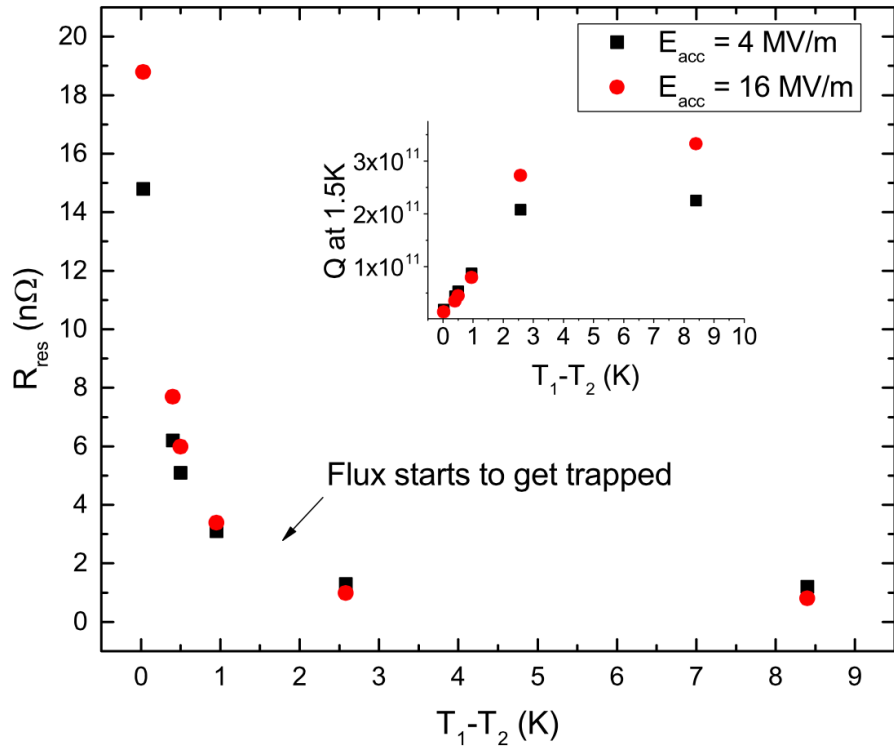
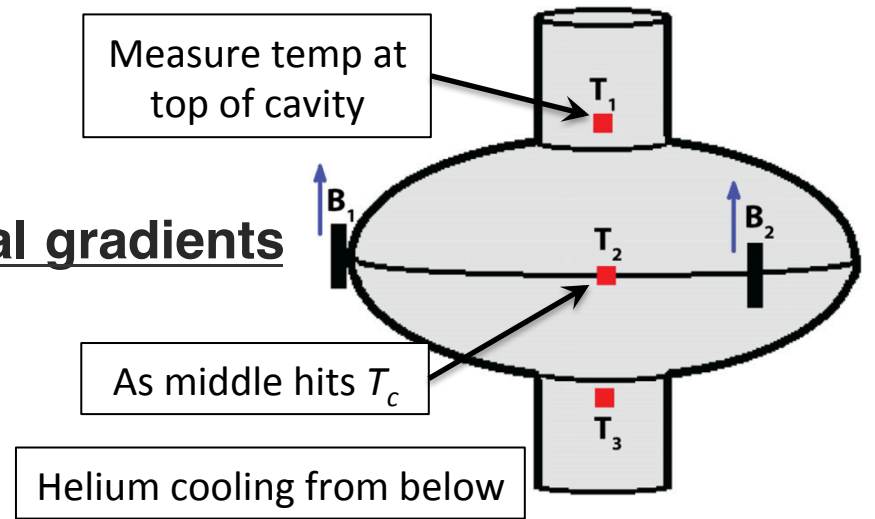
Temperature sensor

Fluxgate magnetometer



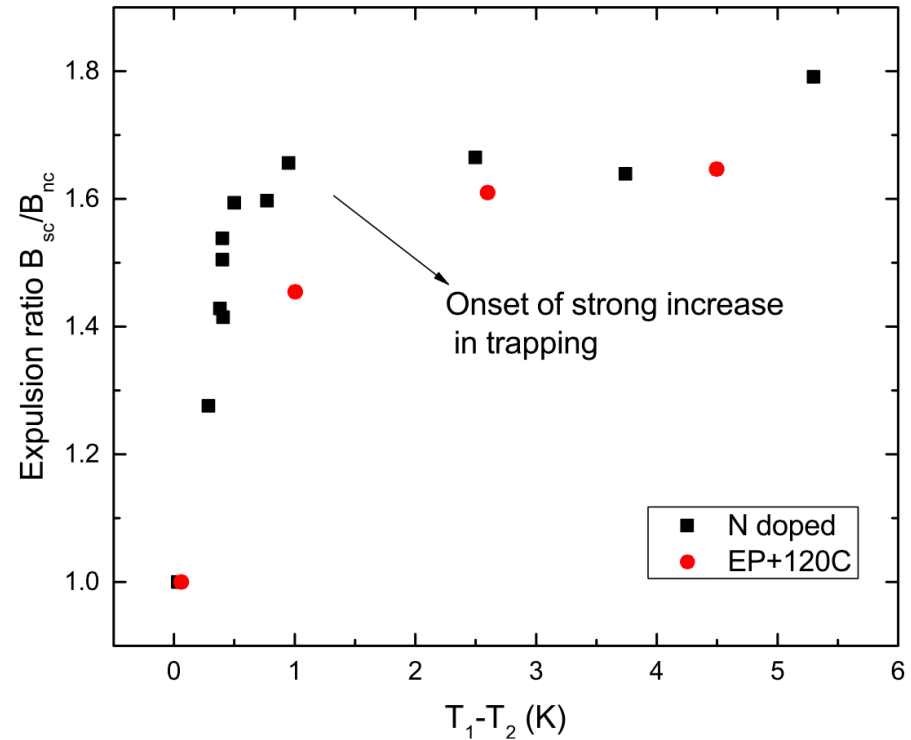
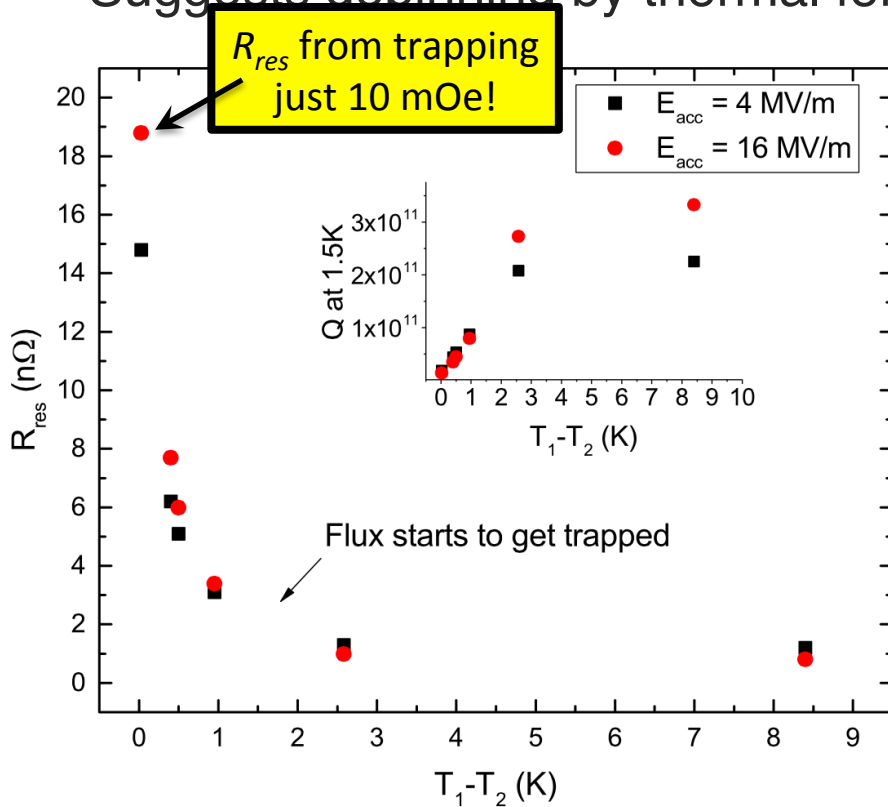
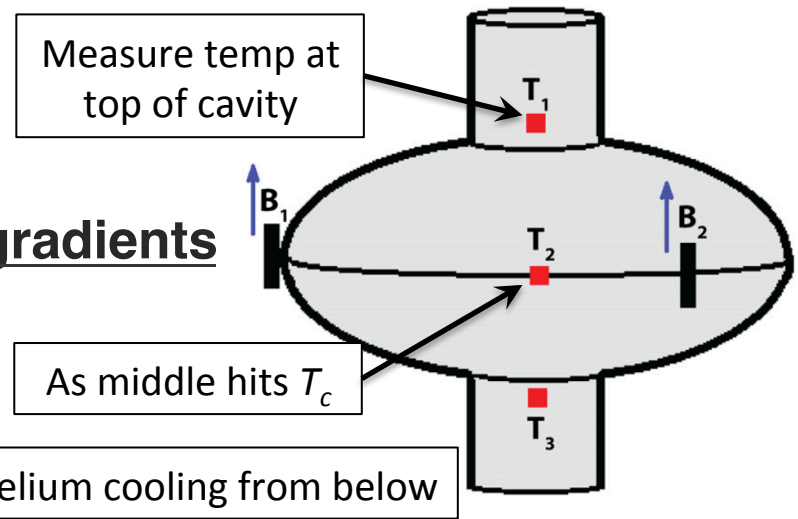
Thermal Gradients

- **Fast cool-down** lead to large thermal gradients which promote efficient flux expulsion
- **Slow cool-down** → poor flux expulsion
- Suggests depinning by thermal forces



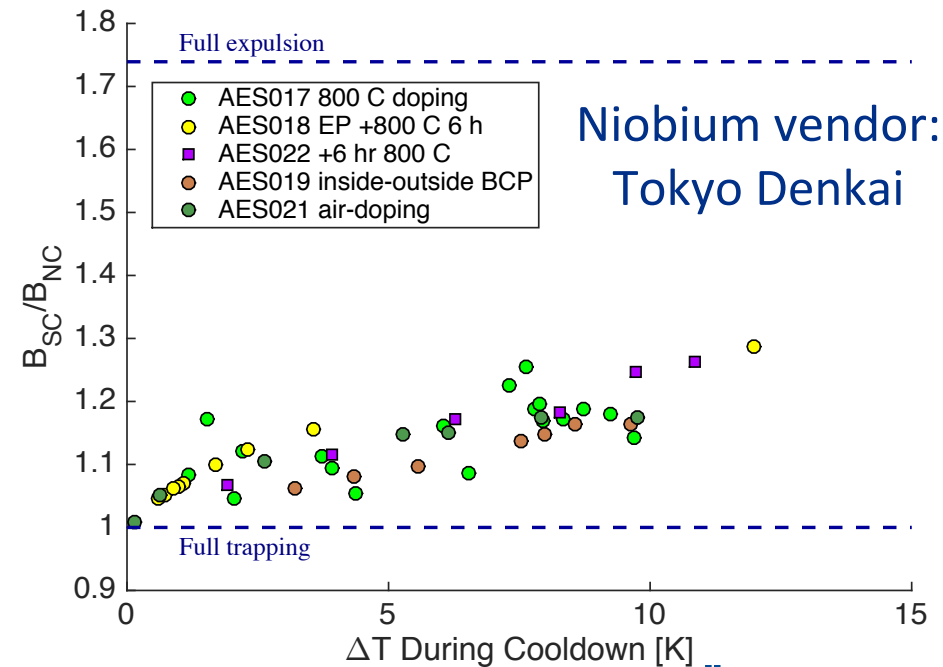
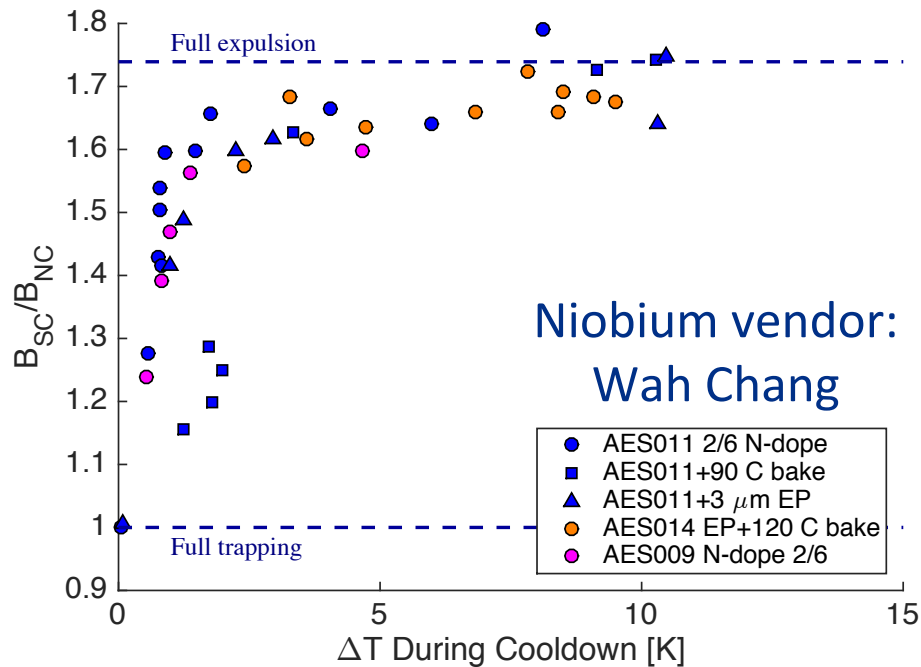
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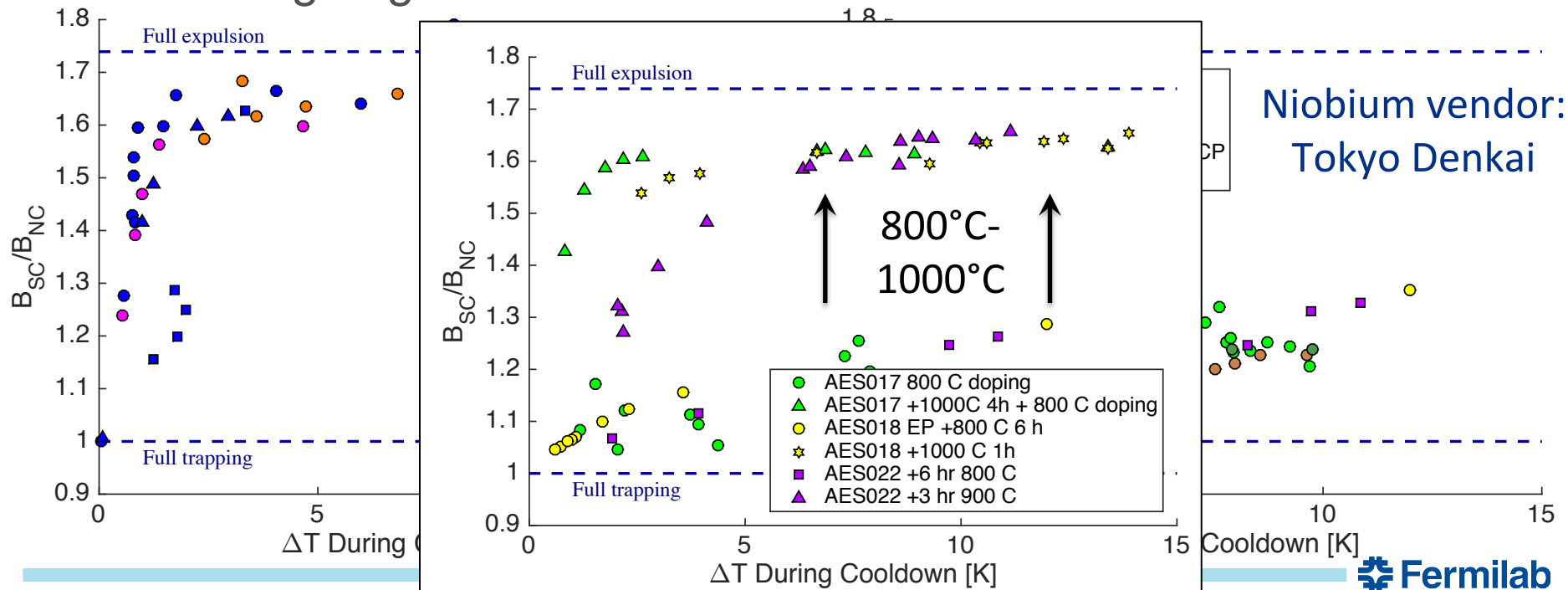
Variability in Material and Modifying Expulsion Behavior

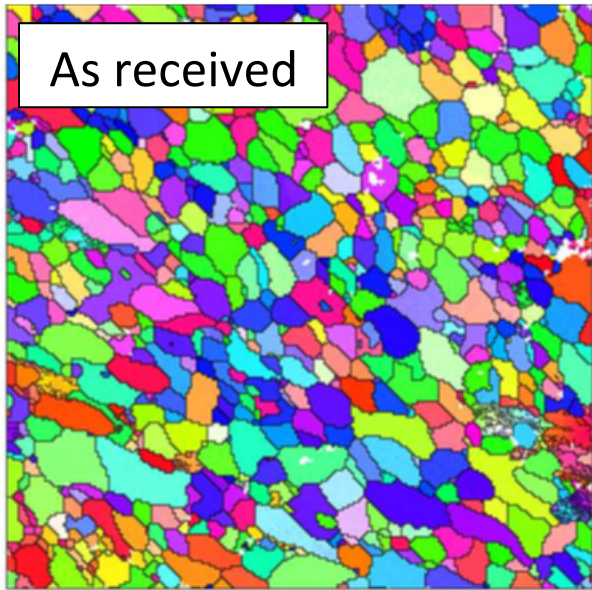
- Seems to be a great deal of variability in as-received material
- Variability from batches even within a single vendor



Variability in Material and Modifying Expulsion Behavior

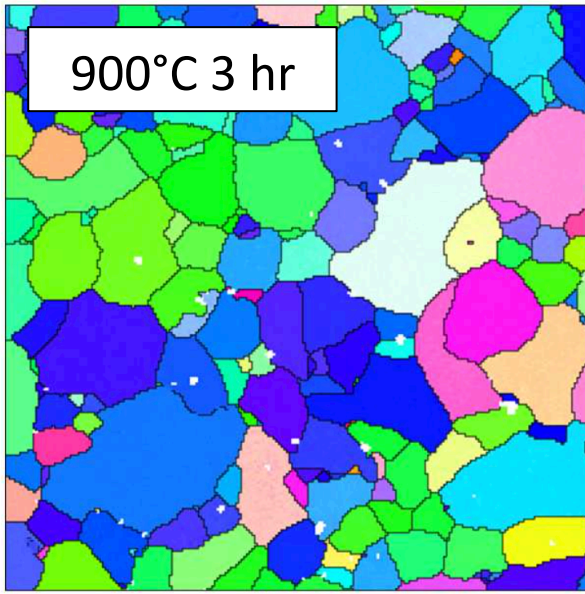
- Seems to be a great deal of variability in as-received material
- Variability from batches even within a single vendor
- High temp vacuum furnace treatment *improves* expulsion
- Surface treatments do not affect expulsion → bulk property
 - Pinning at grain boundaries? Dislocations?





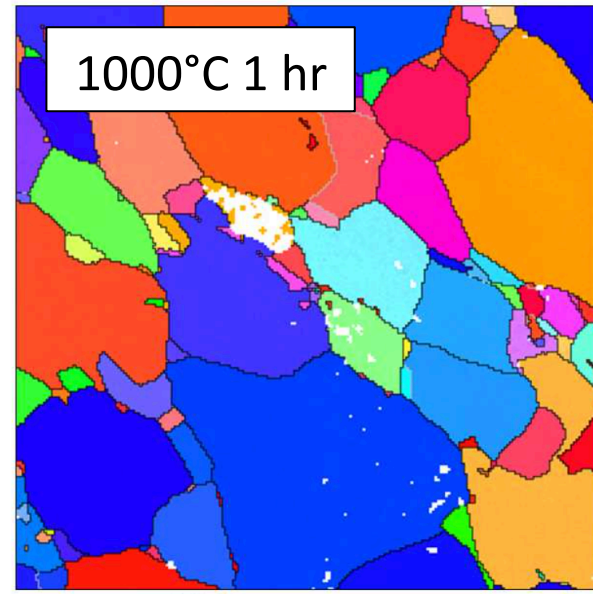
As received

200 μm



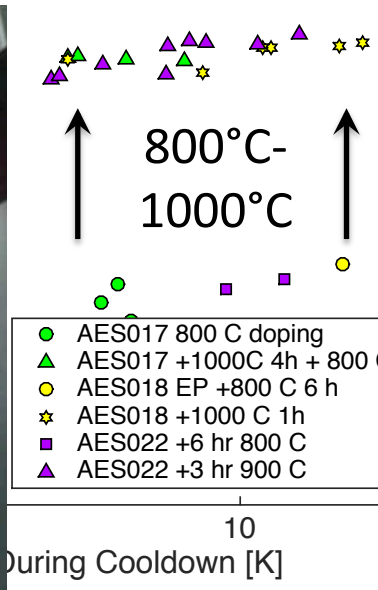
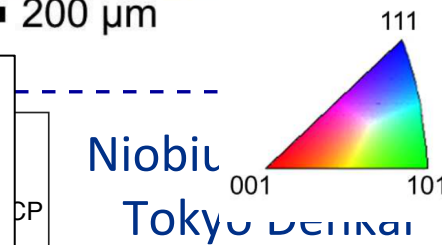
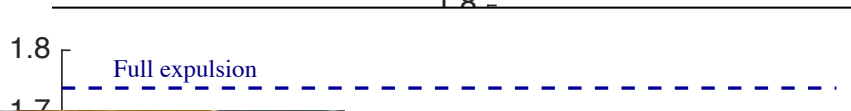
900°C 3 hr

200 μm



1000°C 1 hr

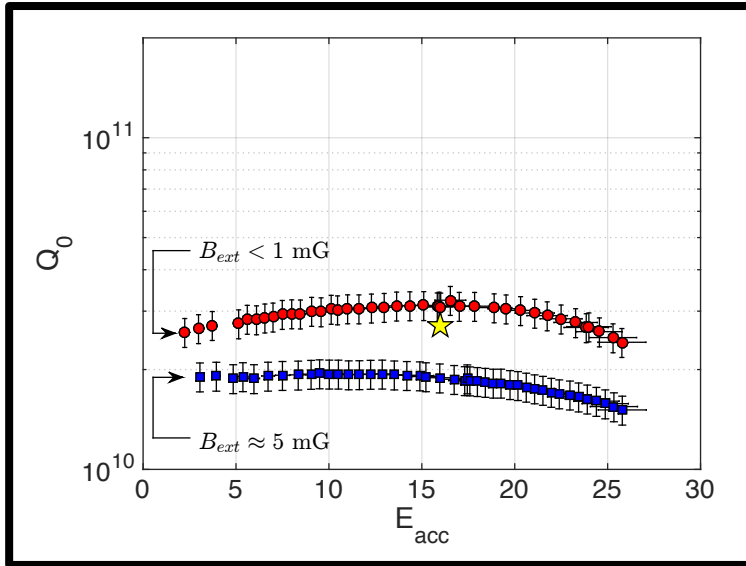
200 μm



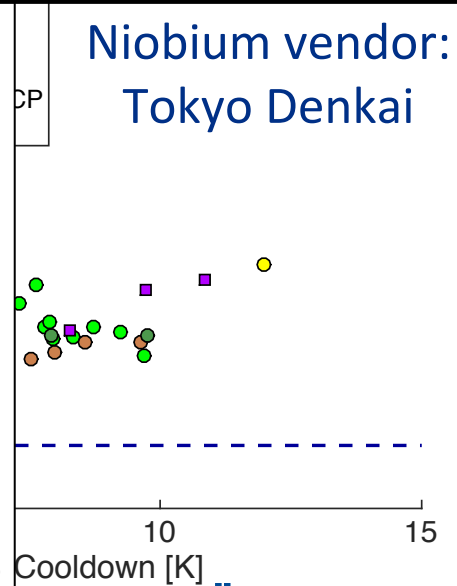
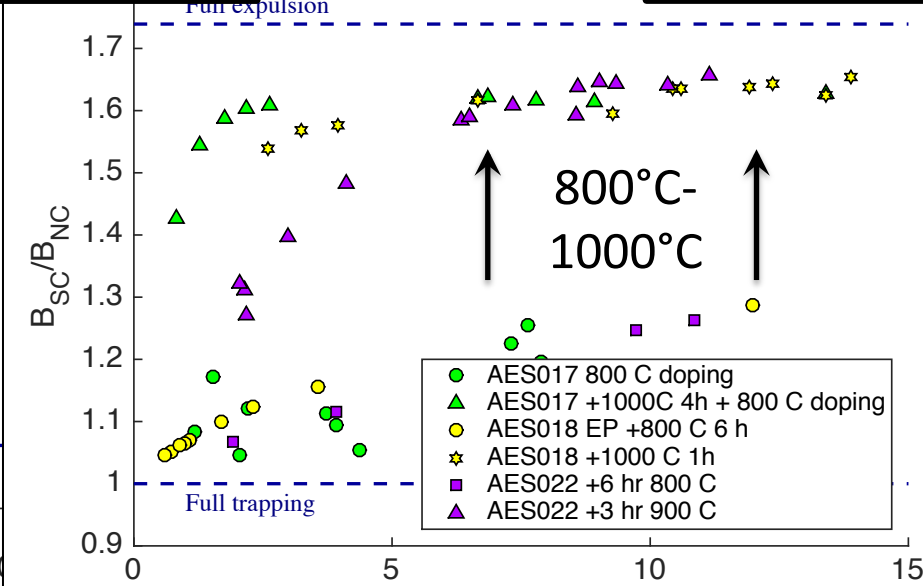
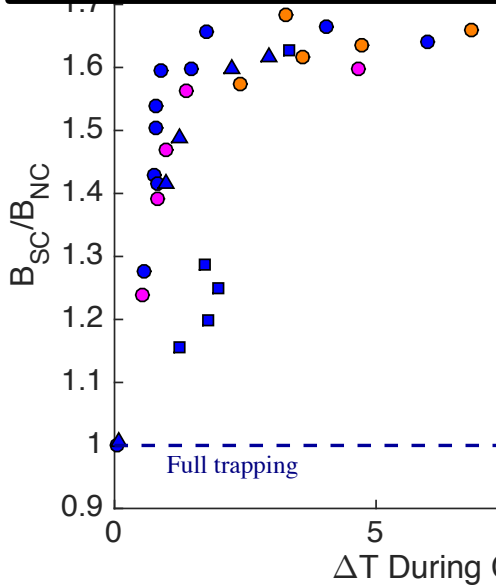
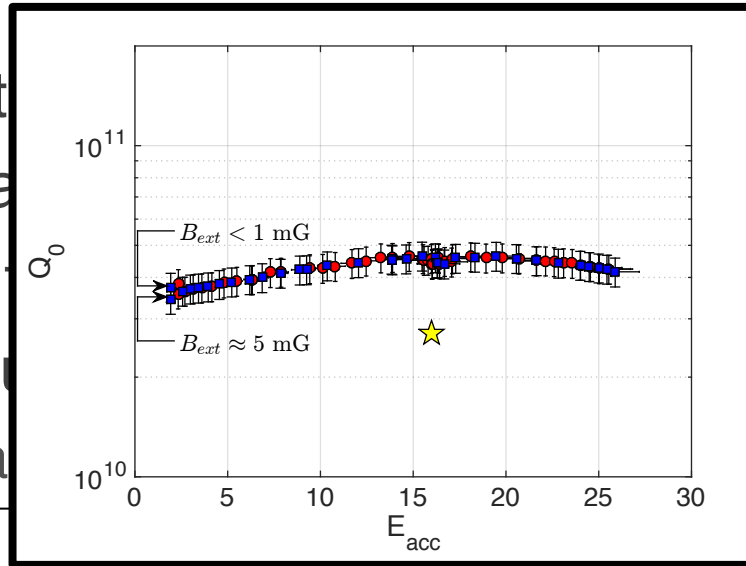
- AES017 800 C doping
- ▲ AES017 +1000C 4h + 800 C
- AES018 EP +800 C 6 h
- ★ AES018 +1000 C 1 h
- AES022 +6 hr 800 C
- ▲ AES022 +3 hr 900 C



Variability in Material and Modifying Expulsion Behavior

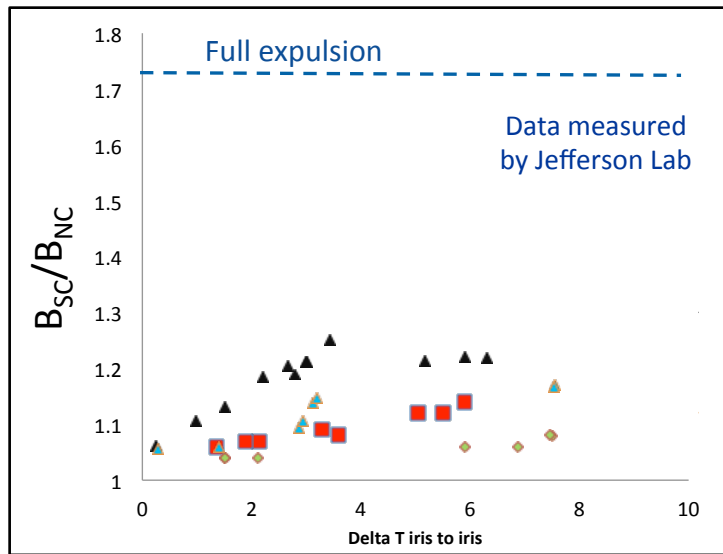


deal of variability
 surface treatment
 900°C
 treatment

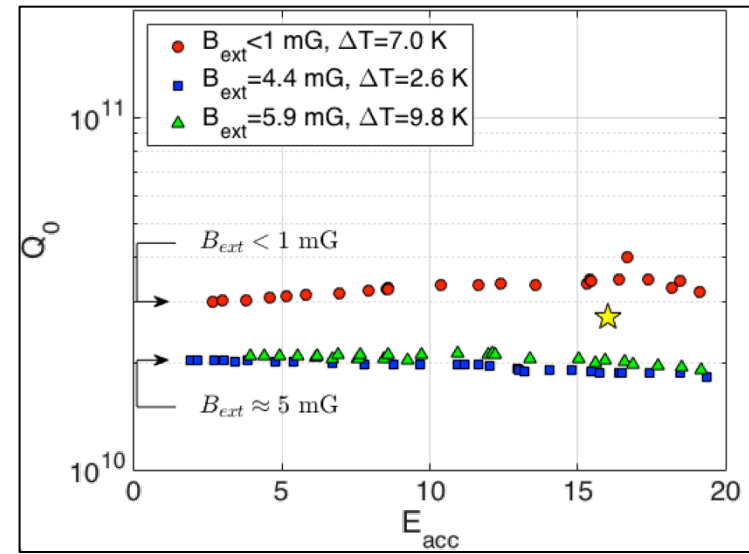


LCLS-II - Preproduction

As-received niobium material for LCLS-II production: very poor expulsion

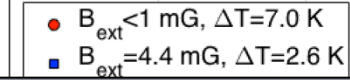


Cooling in 5 mG applied field
(spec for background field in module)



LCLS-II - Preproduction

Cooling in 5 mG applied field
(spec for background field in module)



As-rec
niobi
mate
for LC
produc
very p
expul



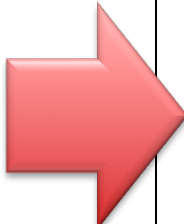
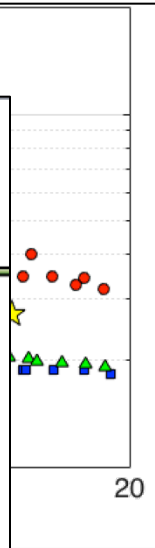
2.7 Cryomodules

D. Stout, MSU / Subcommittee 7

OFFICE OF SCIENCE

Recommendations

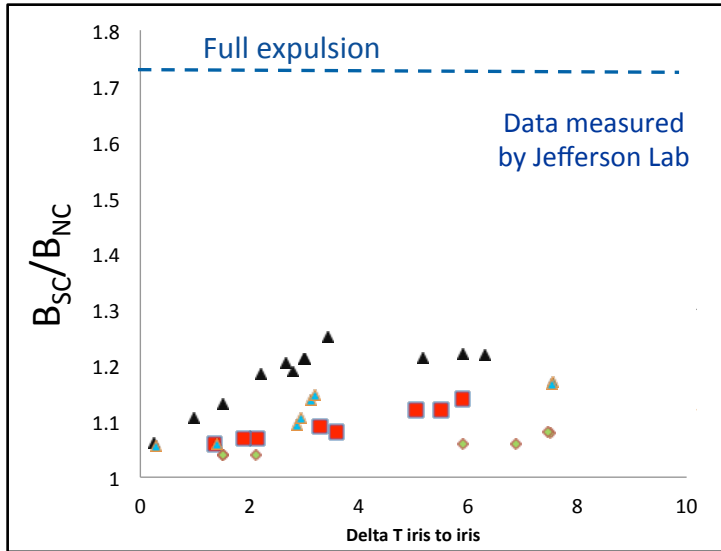
1. The Project is ready to proceed to CD-2/3
2. Finalize cavity and cryomodule minimum acceptance criteria based on the current project baseline – by 3/2016
3. Conduct a supply chain risk assessment of critical cryomodule assembly components to identify items needing second sources or other mitigations – by 3/2016
4. Develop a cure to improve the flux expulsion of the procured niobium material and implement before cavity production.
5. Conduct an independent peer review of the detailed assembly methods for connecting cryomodules – prior to first connection in 2017



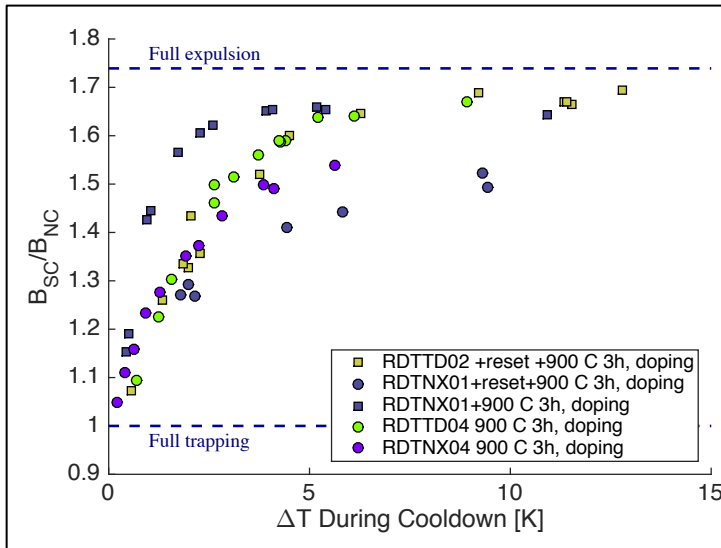
LCLS-II CD2/3 Review Closeout, Dec 2015

LCLS-II - Preproduction

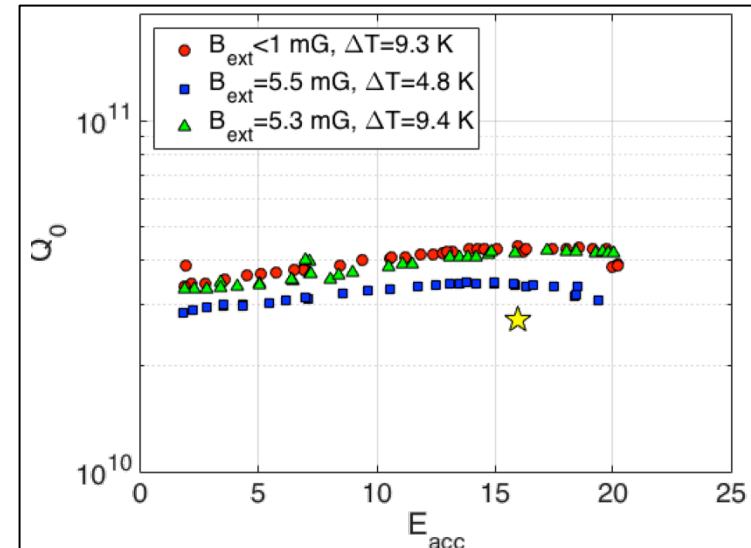
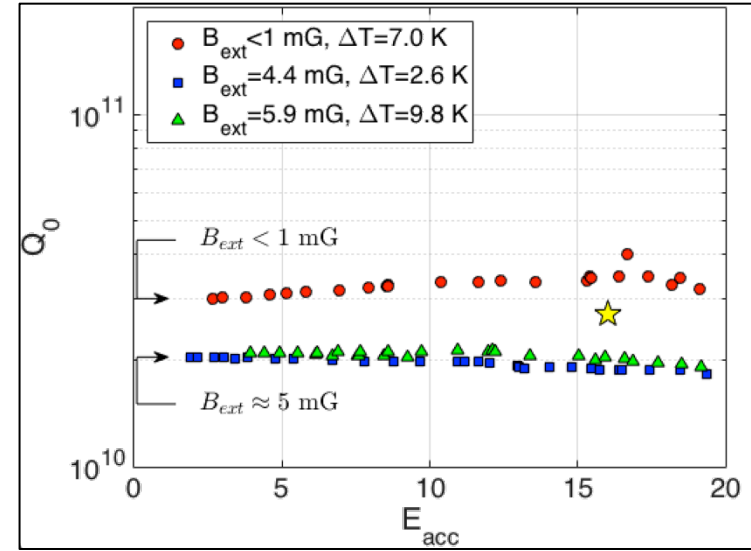
As-received niobium material for LCLS-II production: very poor expulsion



After 900°C treatment: much improved

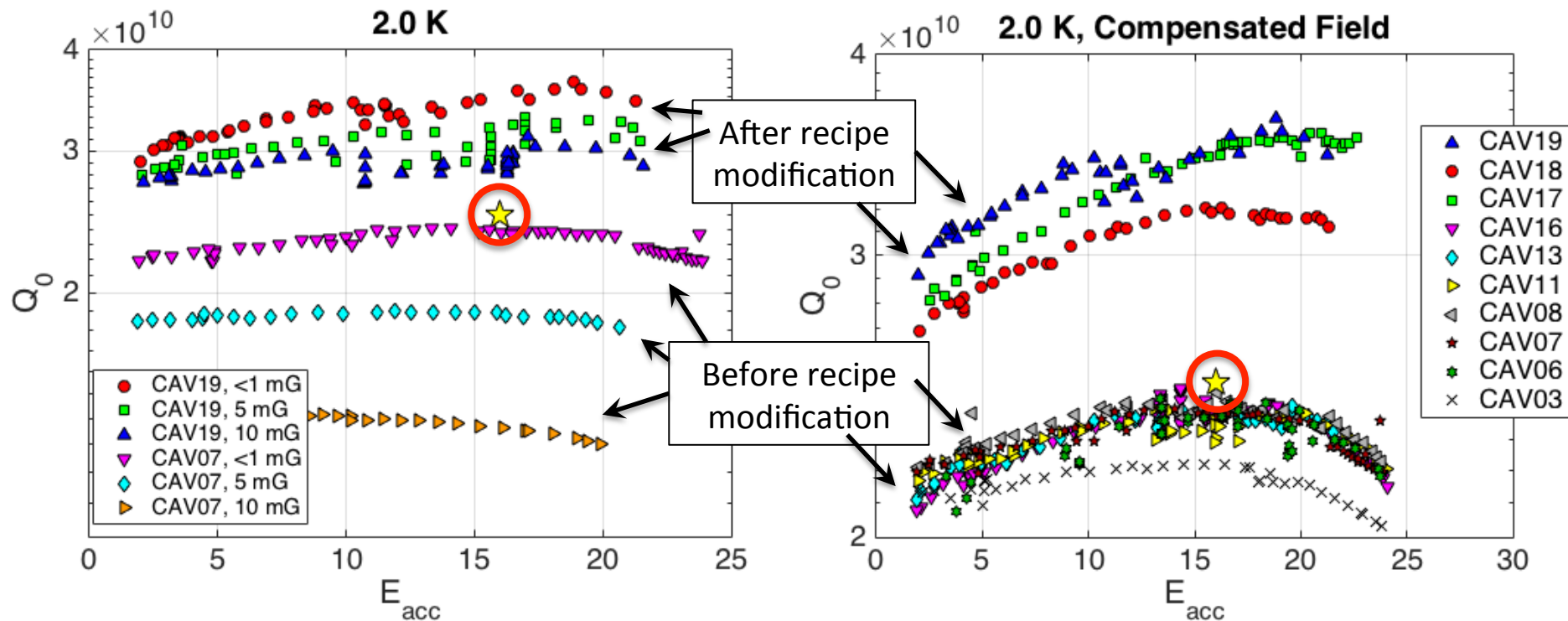


Cooling in 5 mG applied field
(spec for background field in module)



LCLS-II - Production

- After seeing results of first batch of 16 cavities, planned for modifications for next batch
- Modified recipe – 900 C now standard
- Still some stubborn material, but now qualifying great cavities!

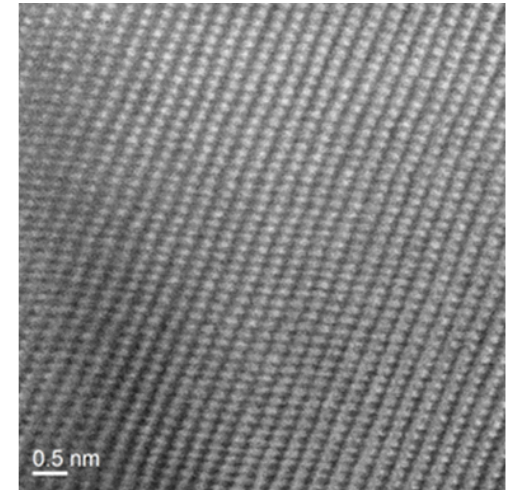


Outline

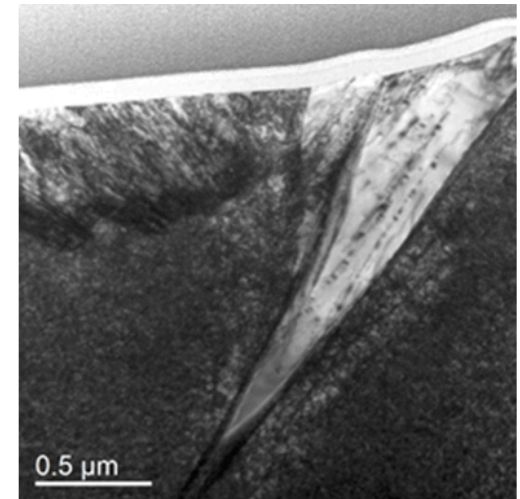
- High Q_0 (medium E_{acc})
 - Nitrogen doping
 - Flux expulsion
- High E_{acc} and high Q_0
 - Nitrogen infusion
 - Nb_3Sn
 - Plasma cleaning



ILC cryomodule with world record gradient for FAST facility



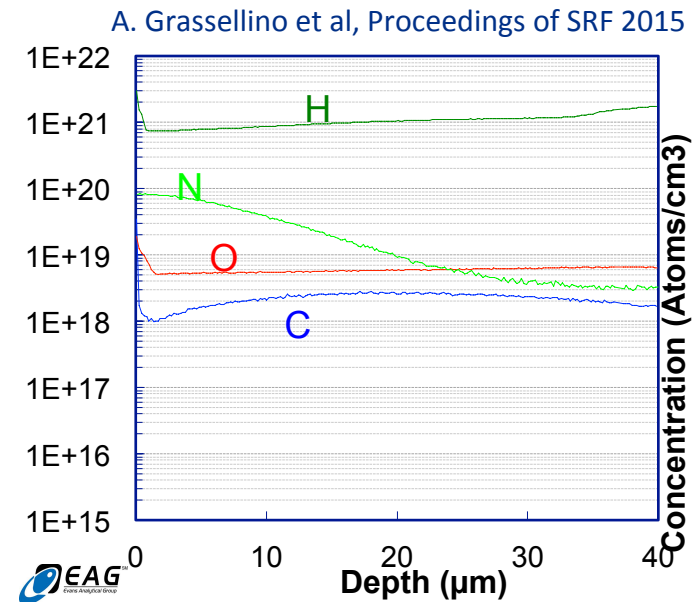
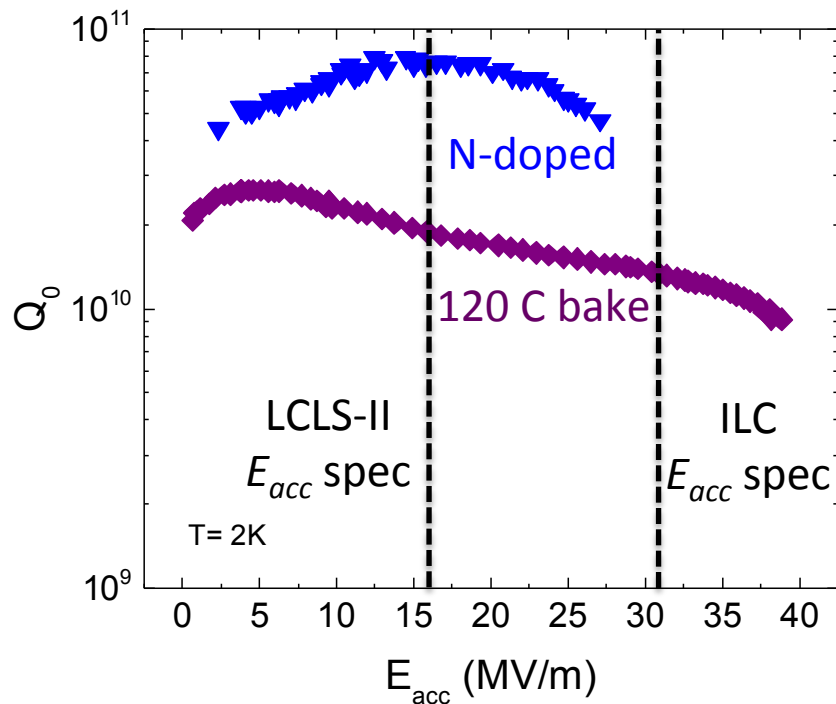
Aberration-corrected STEM image of the near-surface of Nb cavity - Y. Trenikhina



Niobium near-surface after the first step of nitrogen doping - Y. Trenikhina

Motivation behind experiments

- **N Doping** at $T > 800\text{C}$ proven to manipulate mean free path, but constantly throughout several microns, **giving high Q**
- **120C bake** known to manipulate mean free path at very near surface on clean bulk, and **produce the highest gradients**



N doping, nitrogen throughout several microns

Motivation behind experiments

- Therefore, we decided to study how to better “**engineer**” a dirty layer on top a clean bulk, using low T nitrogen treatments → aim to **create few to several nanometers of nitrogen enriched layer** on top of clean EP bulk, to attempt to bring together the benefit of the Q and gradient
- Nitrogen enriched nanometric layer to be created in the furnace post 800C treatment – when no oxide is present at the moment of injection of nitrogen at low T
- Studies aim also at fundamental understanding of HFQS and 120C cure of high field Q slope

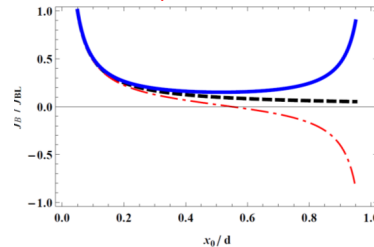
Kubo and Checchin models on bi-layer potentially increasing achievable accelerating gradients

- This idea is further strengthened by Checchin (FNAL) and Kubo (KEK) models on bi-layer structure (eg dirty N doped layer on clean Nb) –claim that can enhance the achievable accelerating gradient
- Ideal Depth of this layer ? Can this trick help push beyond the 200 mT or achieve 200 mT with higher yield? We are investigating this empirically via low T N infusion (different T and durations)

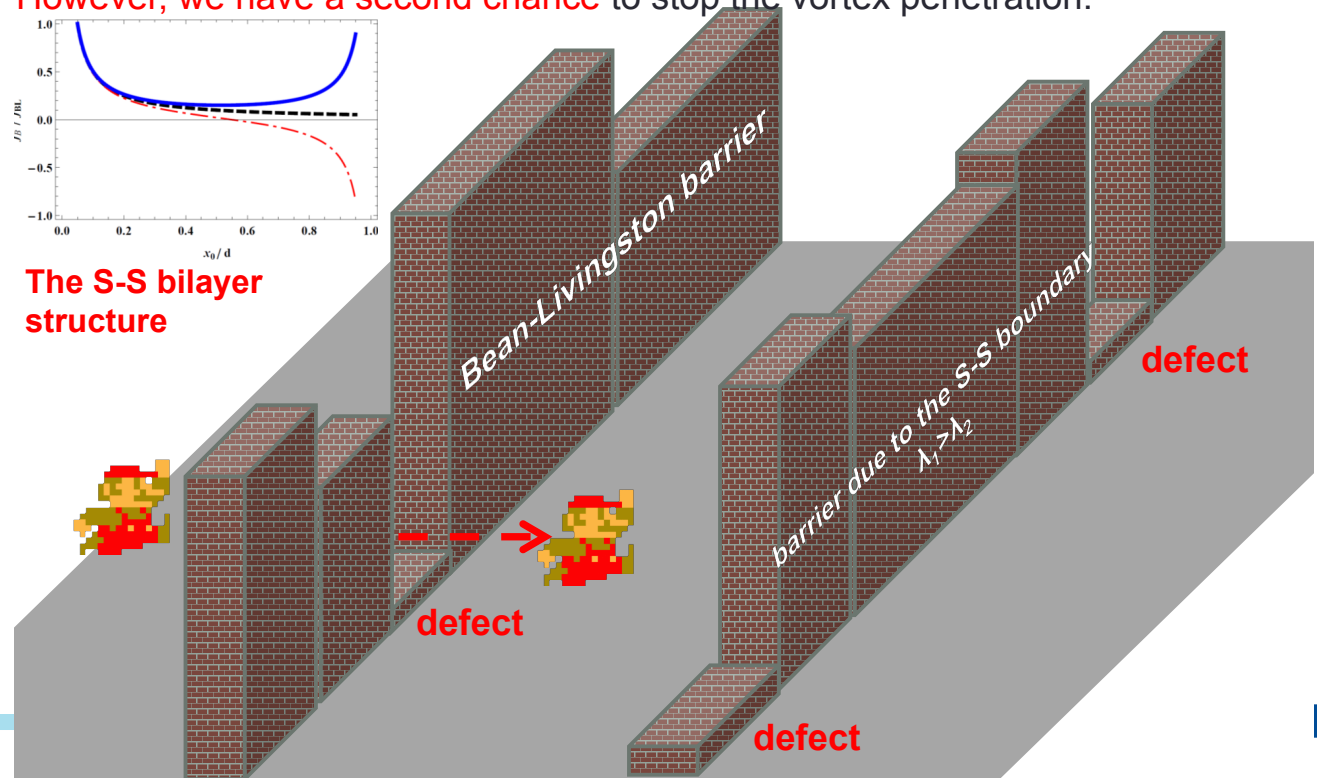
TTC@Saclay

40

In addition to the BL barrier, we have the second barrier due to the S-S boundary. **The second barrier is also imperfect:** easily weakened by defects. **However, we have a second chance** to stop the vortex penetration.



The S-S bilayer structure

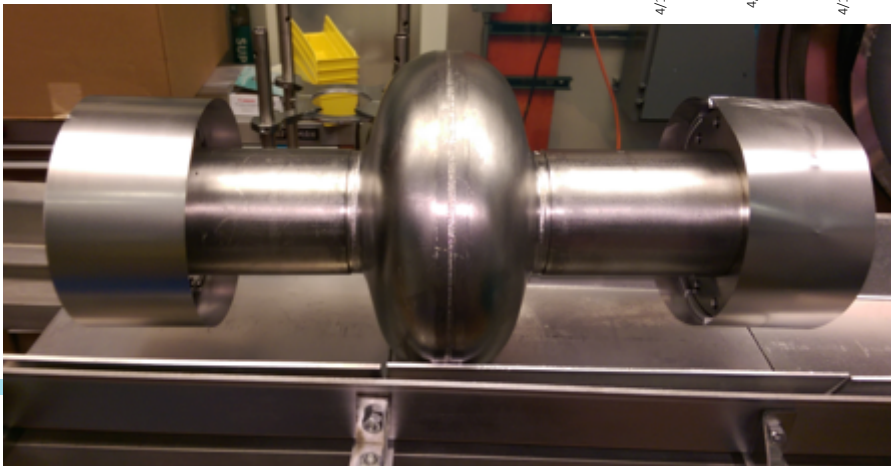
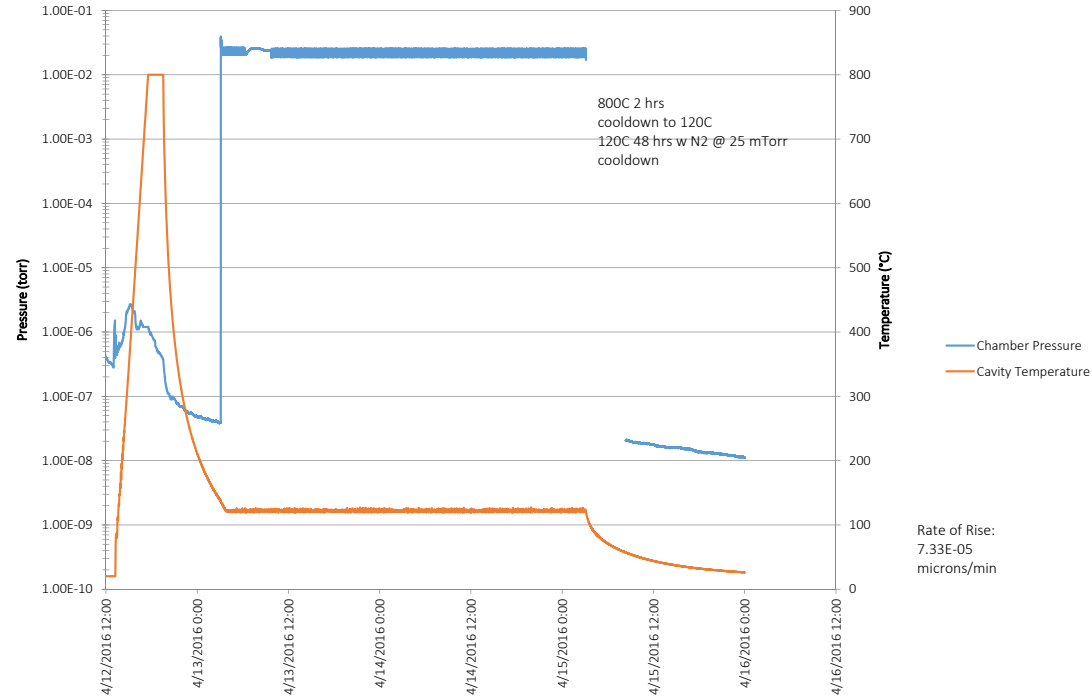


T. Kubo, TTC Meeting 2016

Example of new surface processing sequence

- Bulk electro-polishing
- High T furnace with caps to avoid furnace contamination:
 - 800C 3 hours HV
 - 120C 48 hours with N2 (25 mTorr)
- NO chemistry post furnace
- HPR, VT assembly

TE1PAV007 - with caps - Process
12 April 2016 - IB4 Furnace

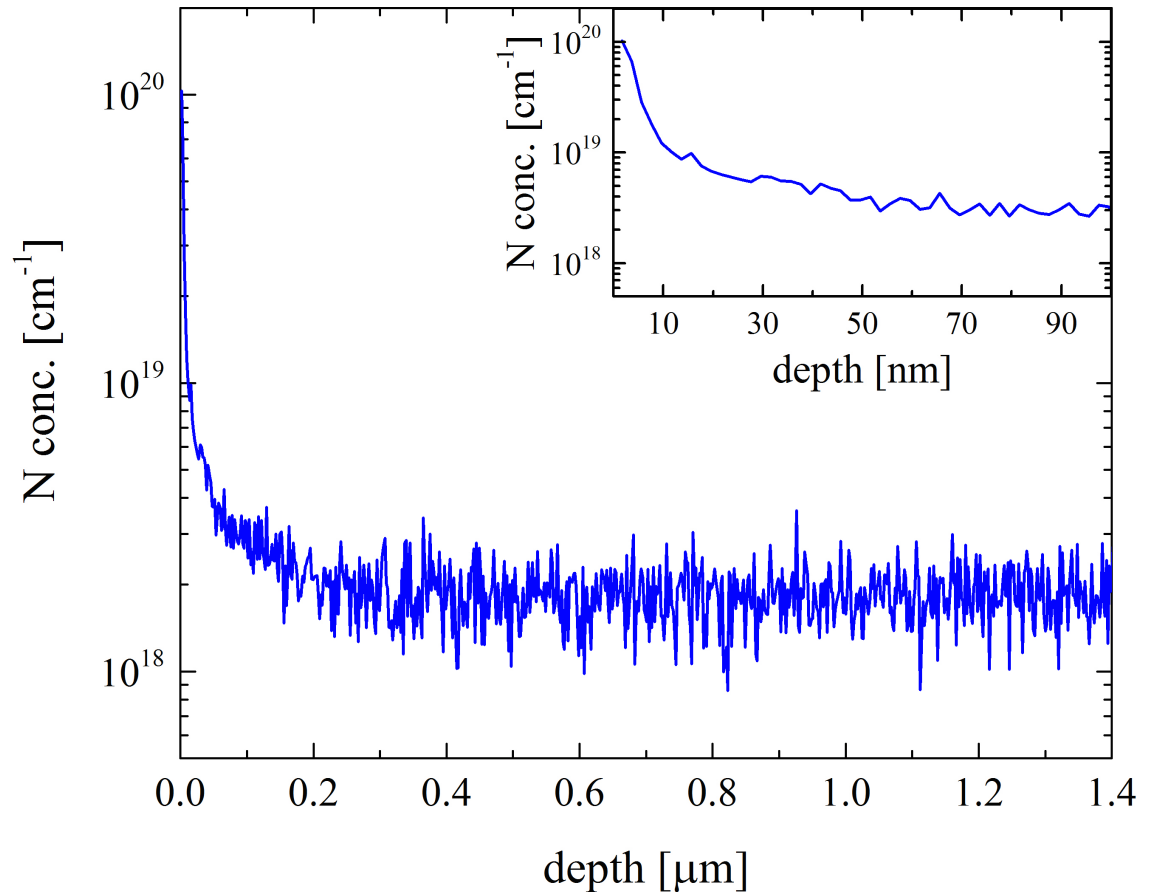


Protective caps and foils are BCP'd prior to every furnace cycle and assembled in clean room, prior to transporting cavity to furnace area

[A. Grassellino et al, arXiv:1305.2182](https://arxiv.org/abs/1305.2182)

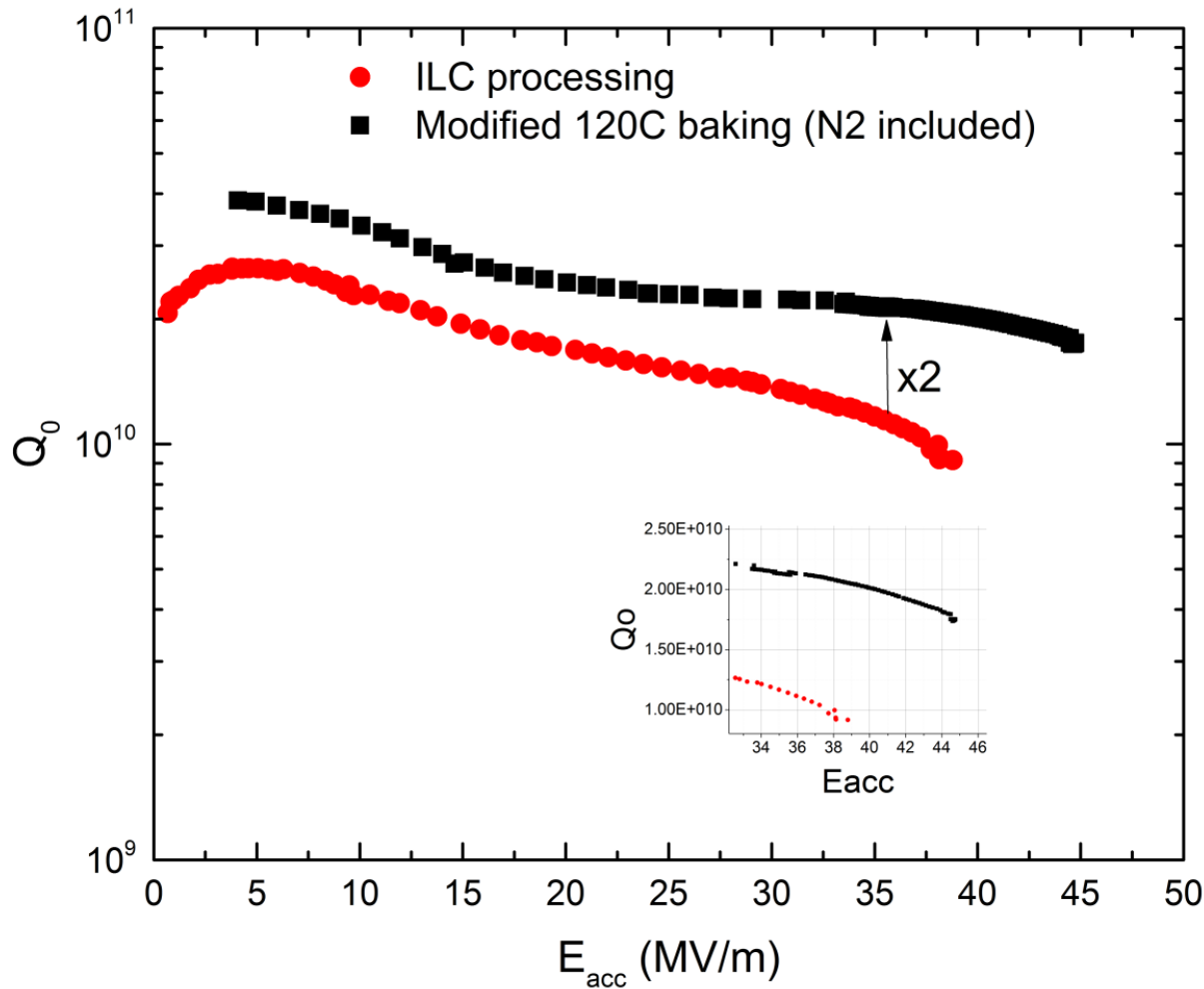
To what depth does nitrogen diffuse?

- First SIMS measurements on samples baked with cavities in furnace with N at low T show ~ few-tens nm potential nitrogen enriched layer, depending on T used



Results comparison :

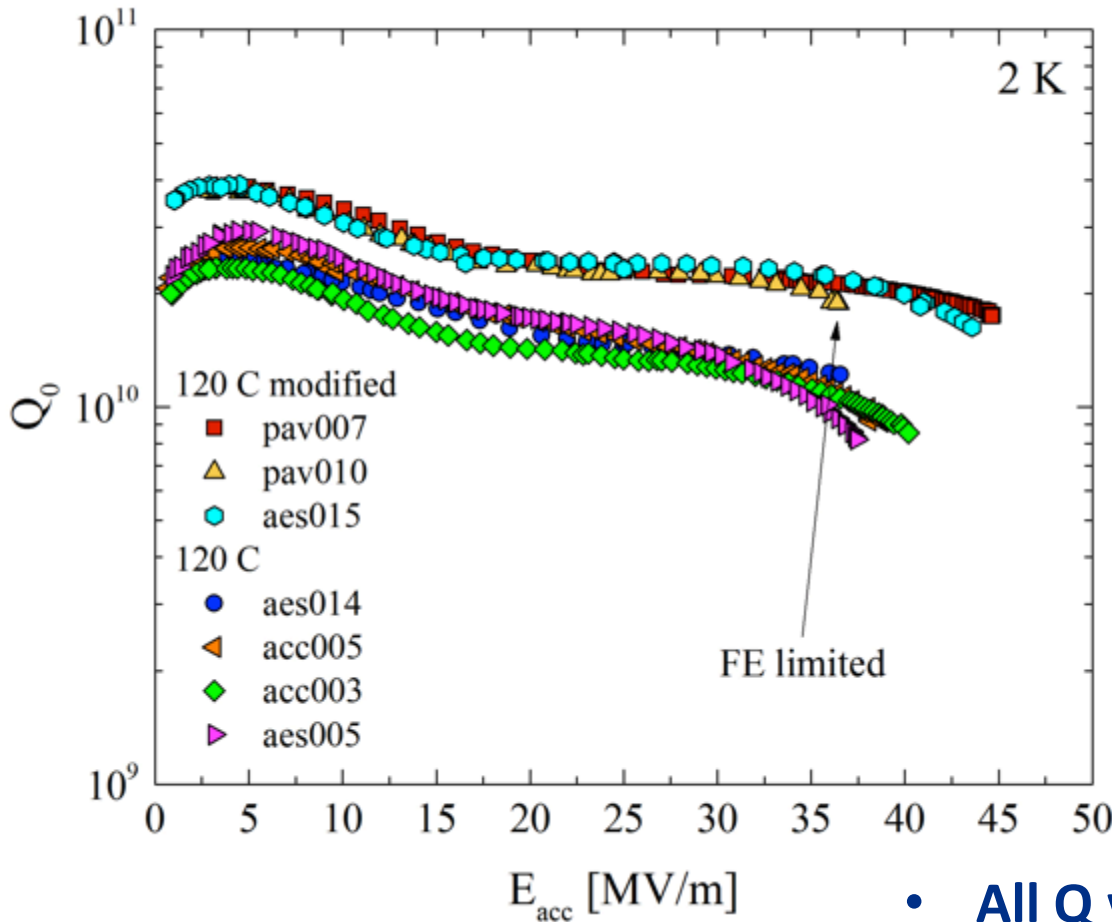
“standard” 120C bake vs “N infused” 120C bake



Increase in Q factor of two, increase in gradient $\sim 15\%$

- Same cavity, sequentially processed, no EP in between
- Achieved:
45.6 MV/m
 \rightarrow 194 mT
With $Q \sim 2e10!$
- Q at ~ 35 MV/m
 $\sim 2.3e10$
- **All Q vs E curves shown are for 1.3 GHz single cells, $T=2K$**

Reproducibility: repeatedly highest Q ever measured $>2e10$ at very high gradients > 40 MV/m!

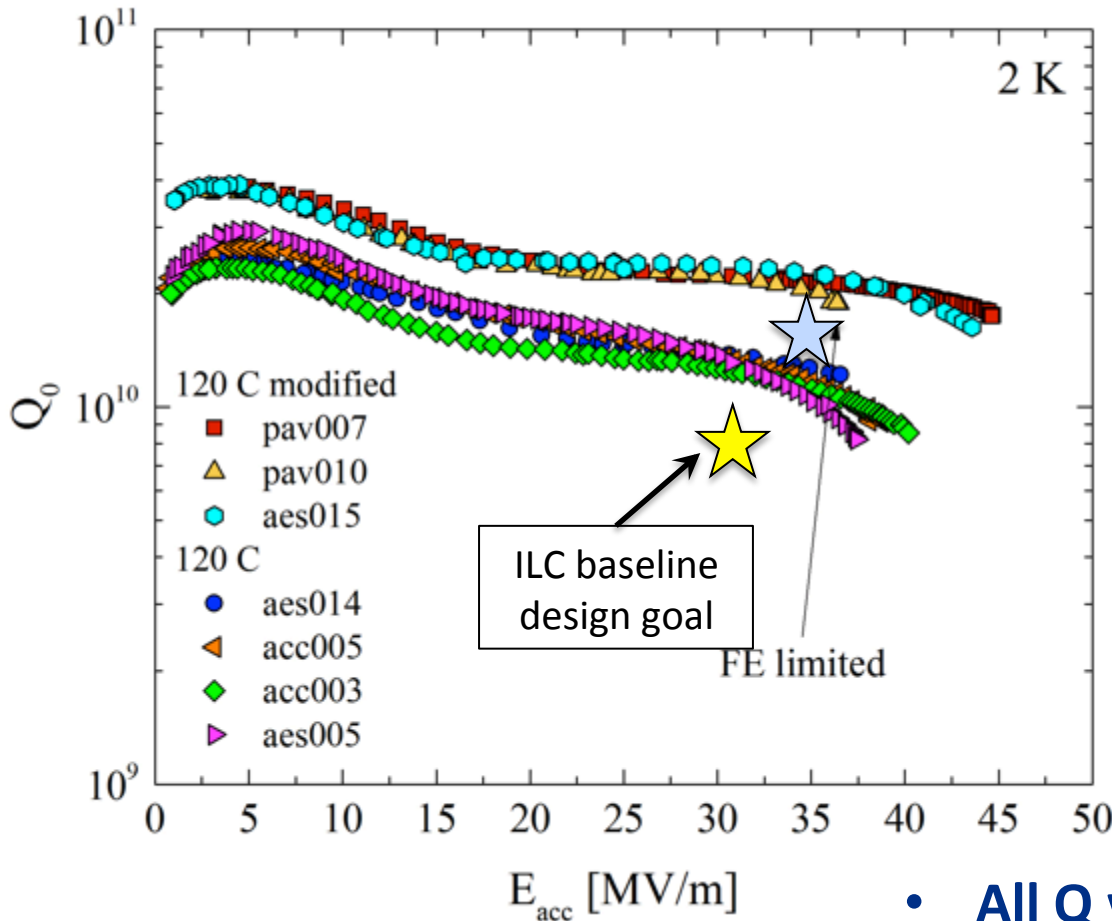


- So far three out of 4 cavities processed with this regime have reached 45 MV/m with high Q

- Performed slow cooldown in 10mG and extracted very low sensitivity to B on order of 0.3 nOhm/mG -> very robust for Q preservation

- All Q vs E curves shown are for 1.3 GHz single cells, T=2K

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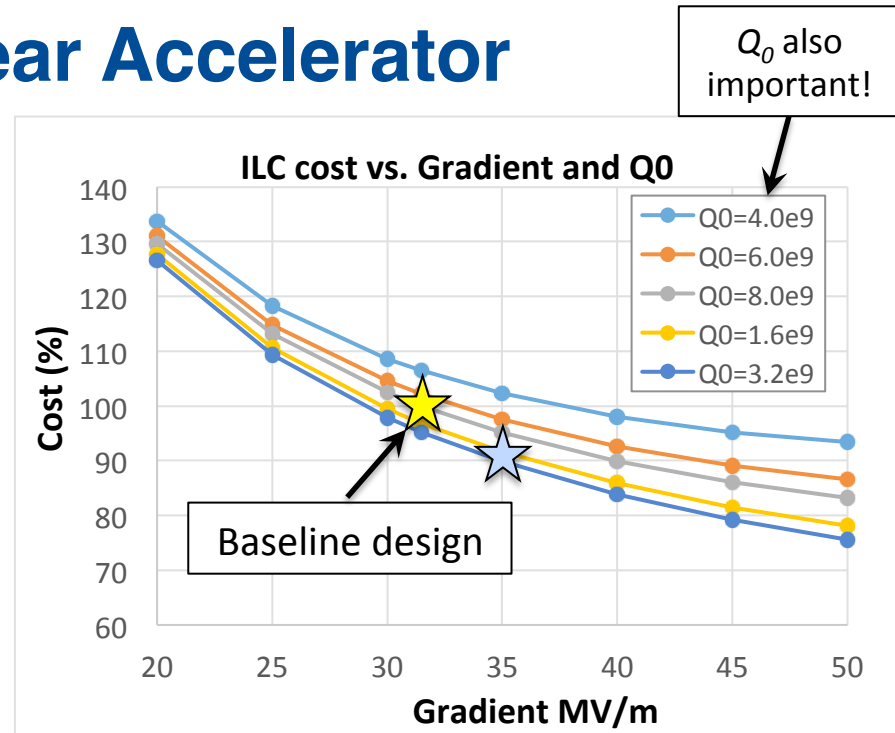
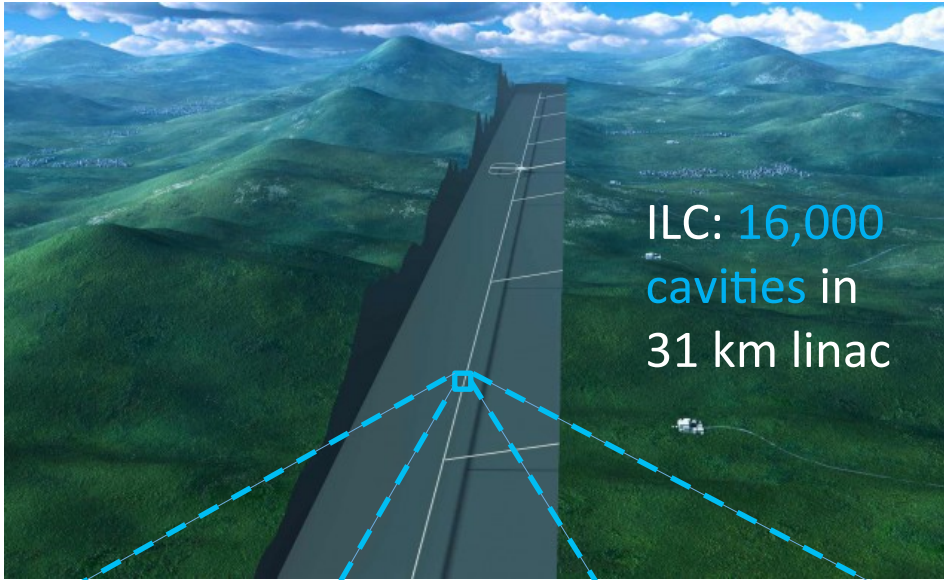


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Gradient -> Length for Linear Accelerator

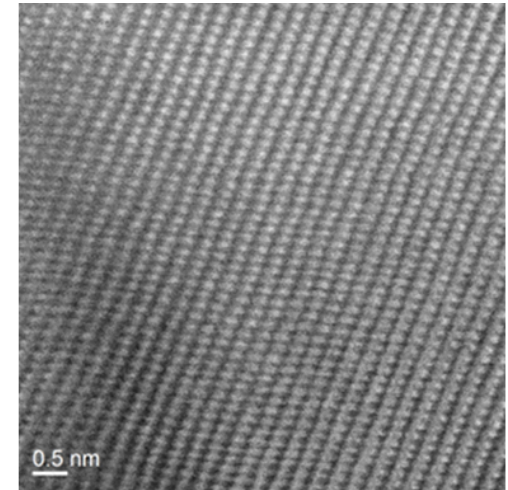


Outline

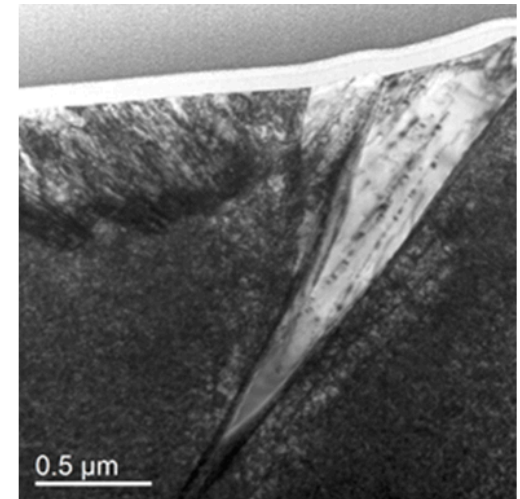
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ILC cryomodule with record average gradient for FAST facility

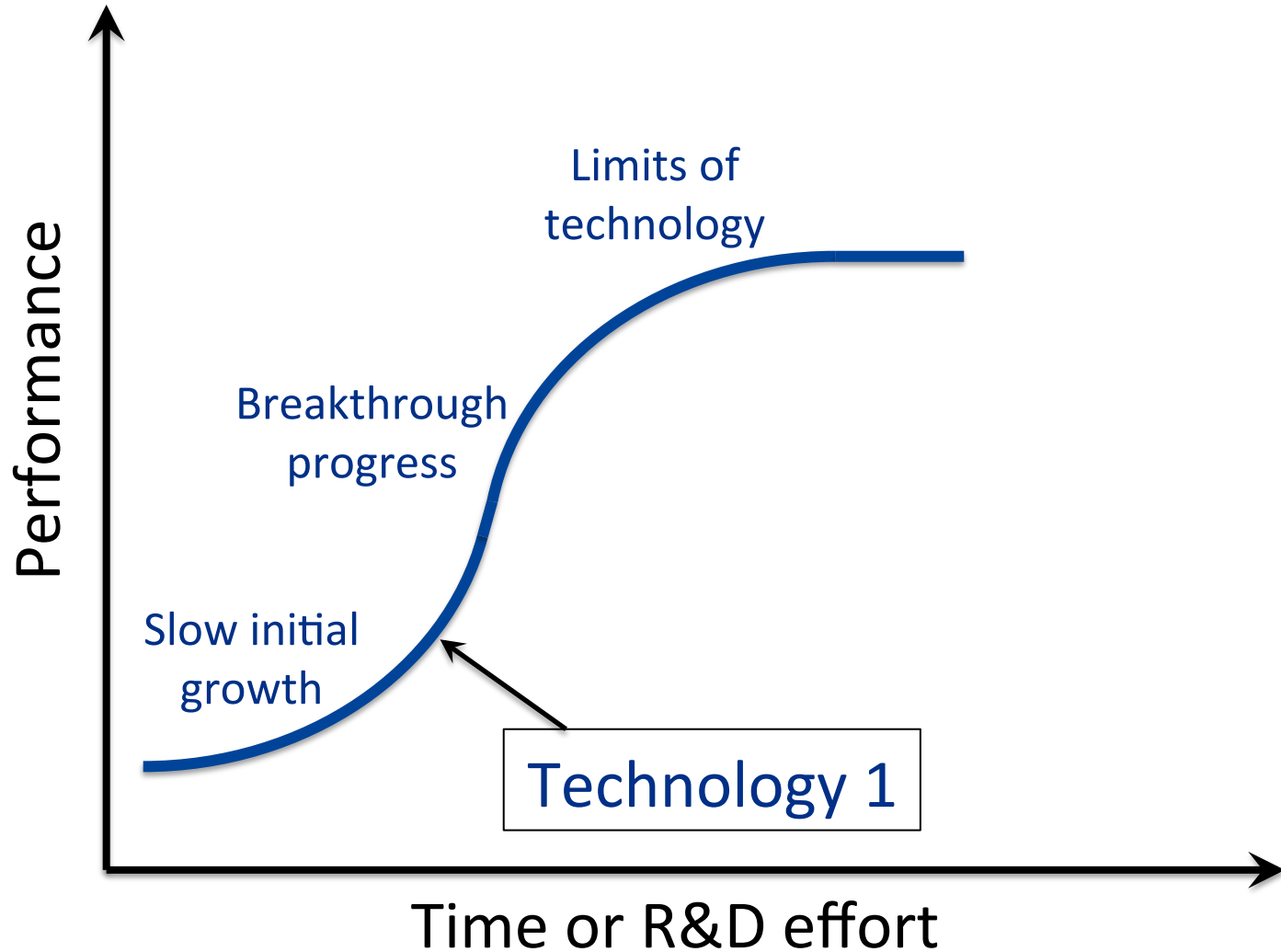


Aberration-corrected STEM image of the near-surface of Nb cavity - Y. Trenikhina

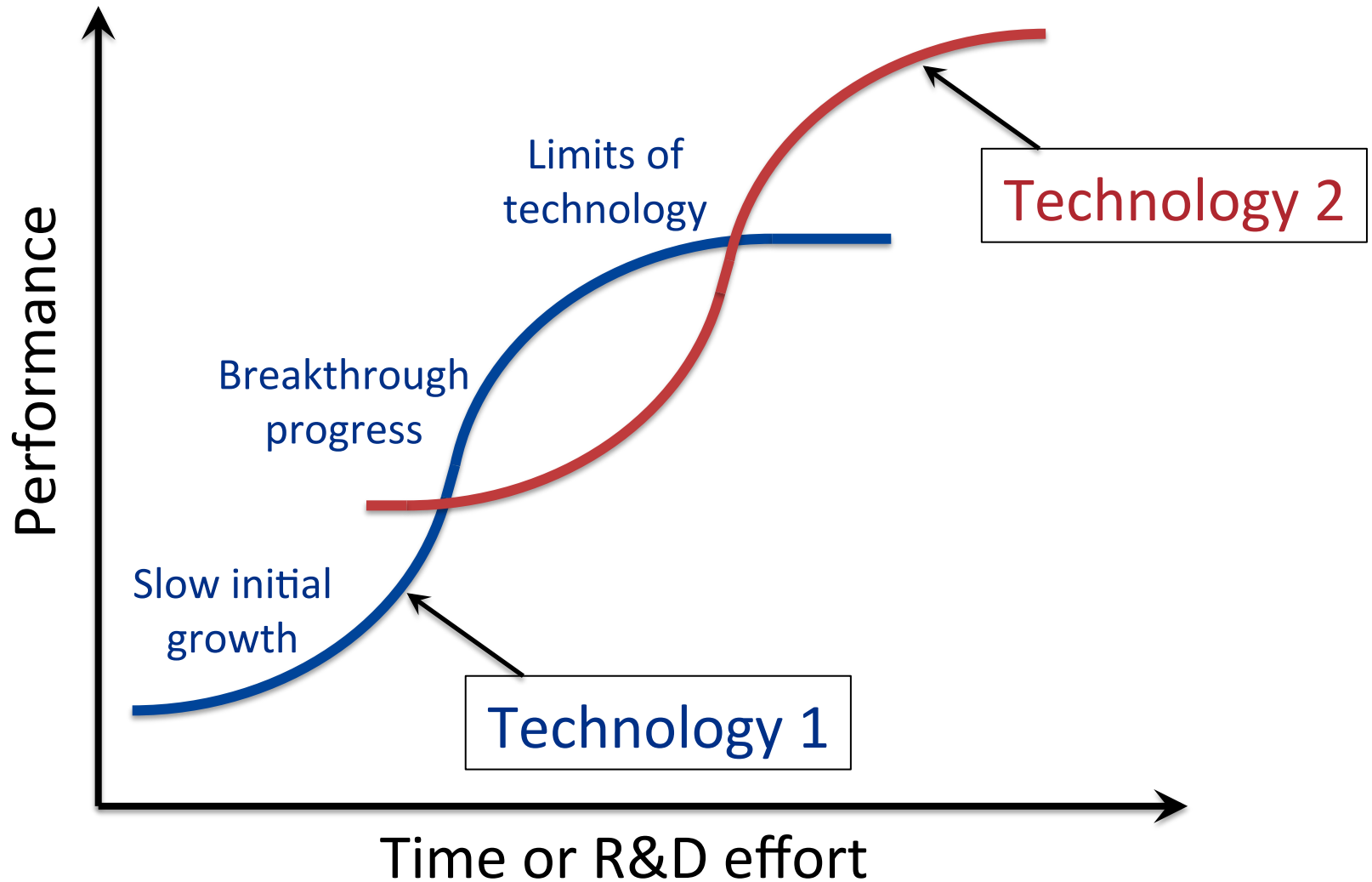


Niobium near-surface after the first step of nitrogen doping - Y. Trenikhina

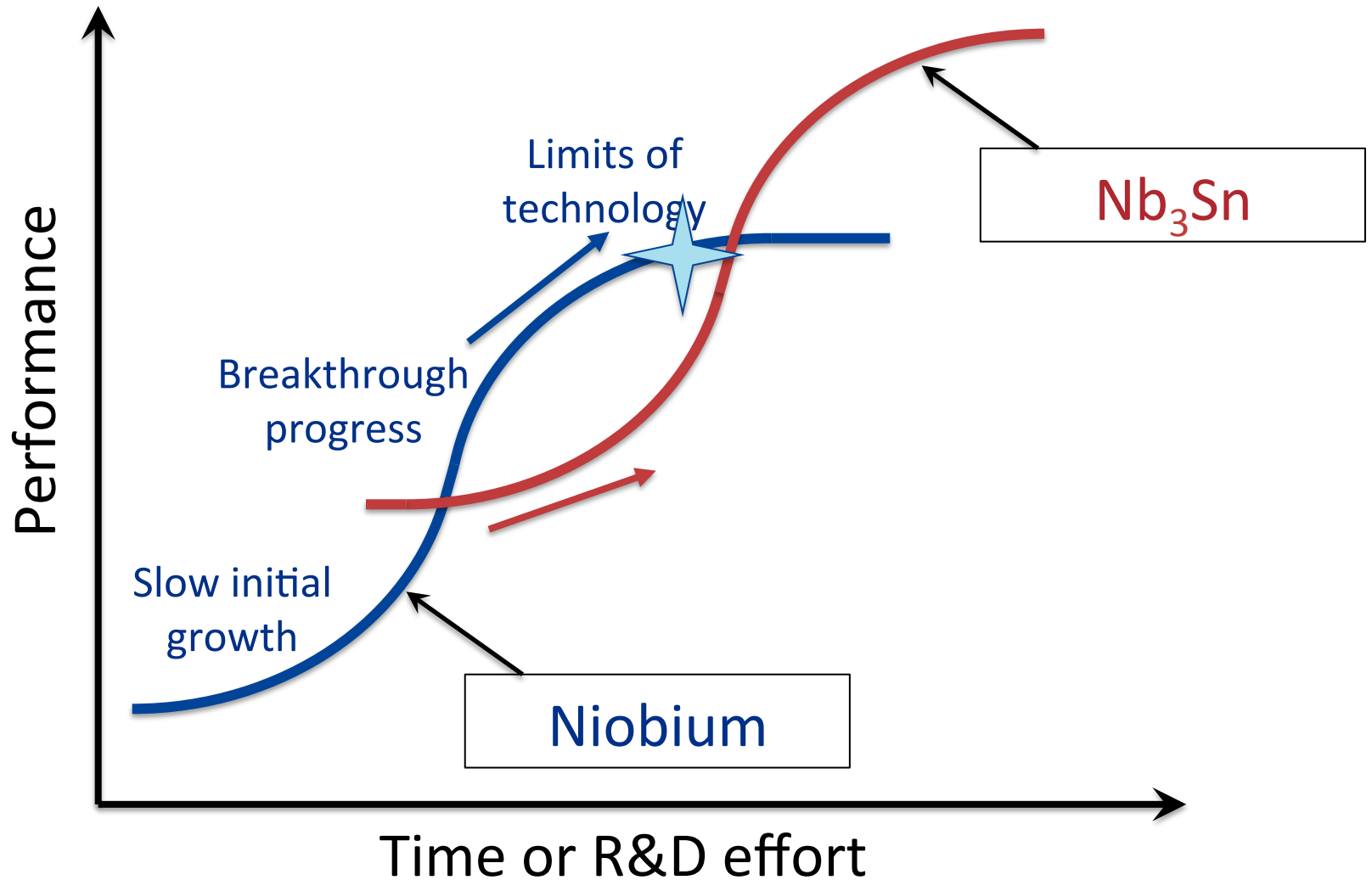
'S-shaped technology curve'



'S-shaped technology curve'



'S-shaped technology curve'

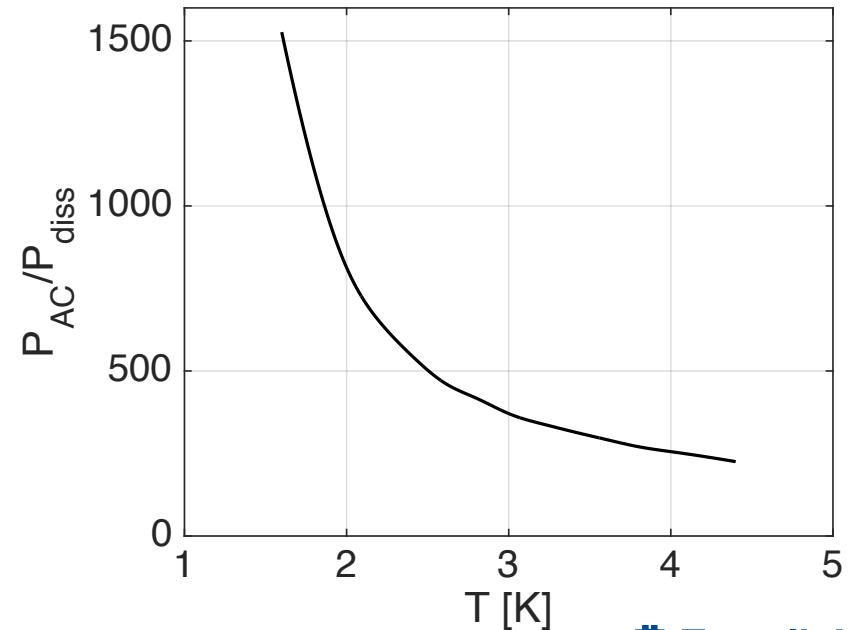
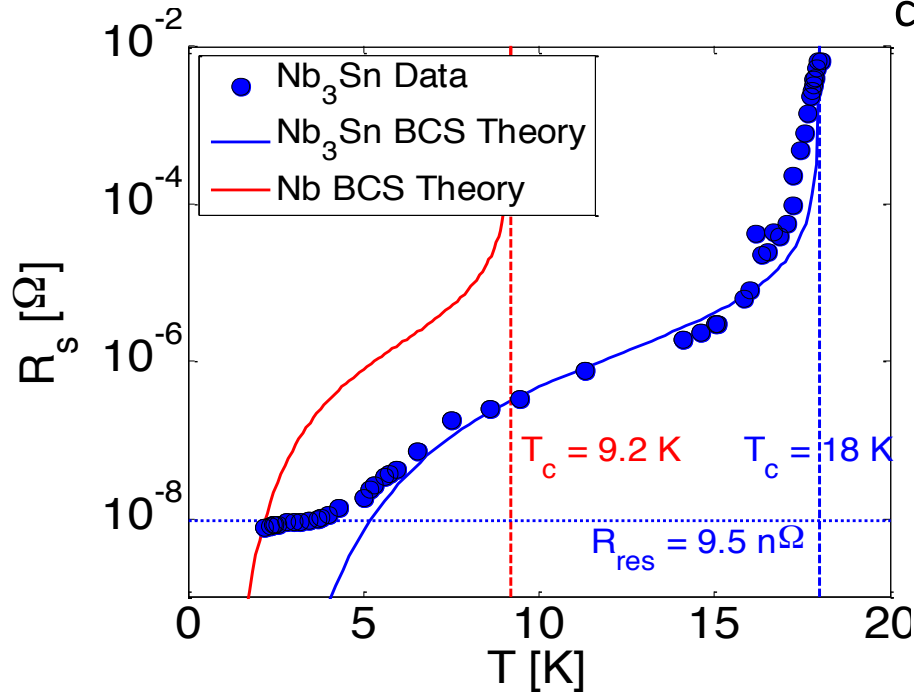


Nb₃Sn Q₀(T)

- Large T_c ~ 18 K
- Very small R_{BCS}(T) – $R_{BCS}(T) \sim e^{-1.76T_c/T}$
- High Q₀ even at relatively high T
- Higher temperature operation
 - Simpler cryogenic plant
 - Higher efficiency



Big effect! Cryoplants for large installations cost ~\$100 million and require MW of power

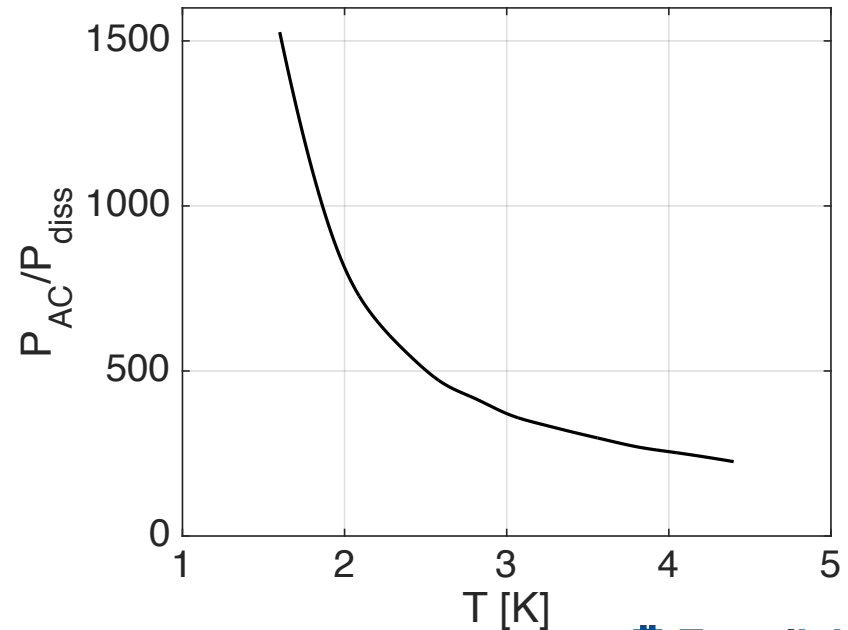
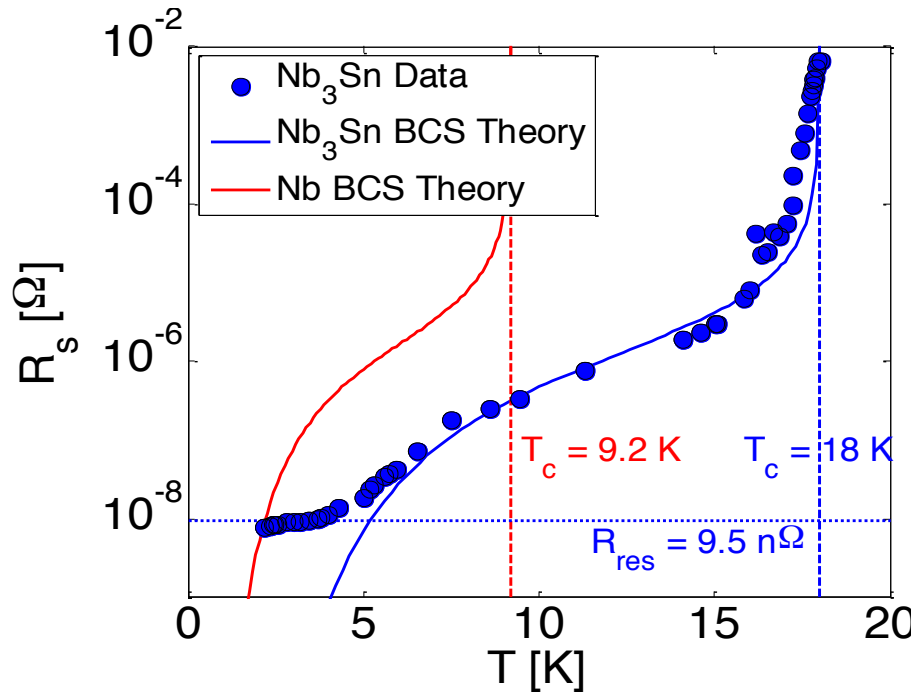


Higher $Q_0(T)$ with Nb_3Sn

- Large $T_c \sim 18$ K
- Very small $R_{BCS}(T)$ – $R_{BCS}(T) \sim e^{-1.76T_c/T}$
- High Q_0 even at relatively high T
- Higher temperature operation
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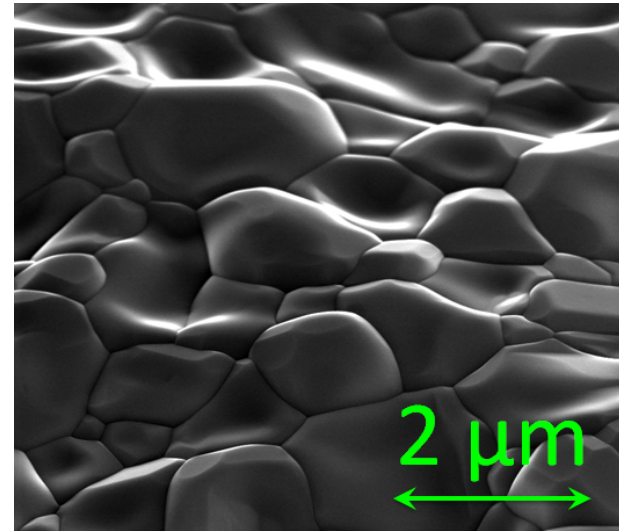
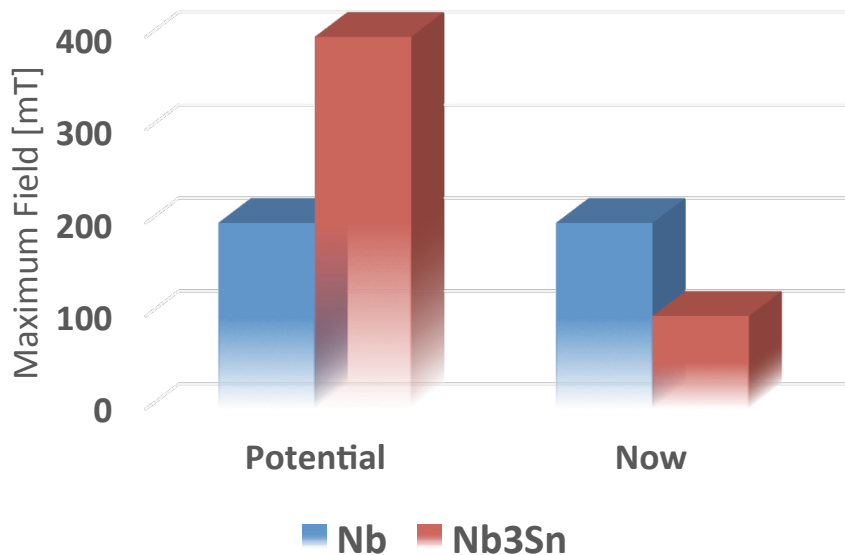


Possibility of cryocooler operation! Industrial accelerators for treatment of wastewater & flue gas, border security...



High H_{sh} with Nb_3Sn

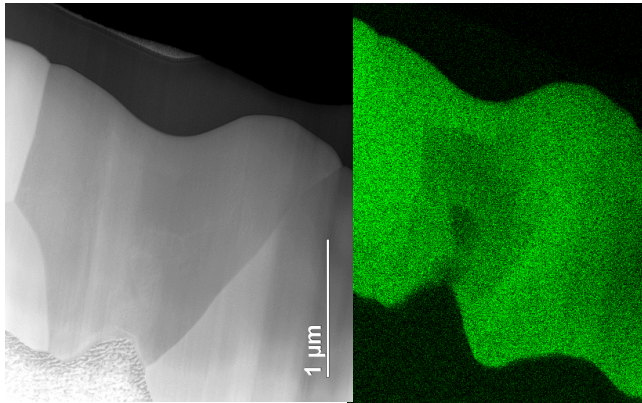
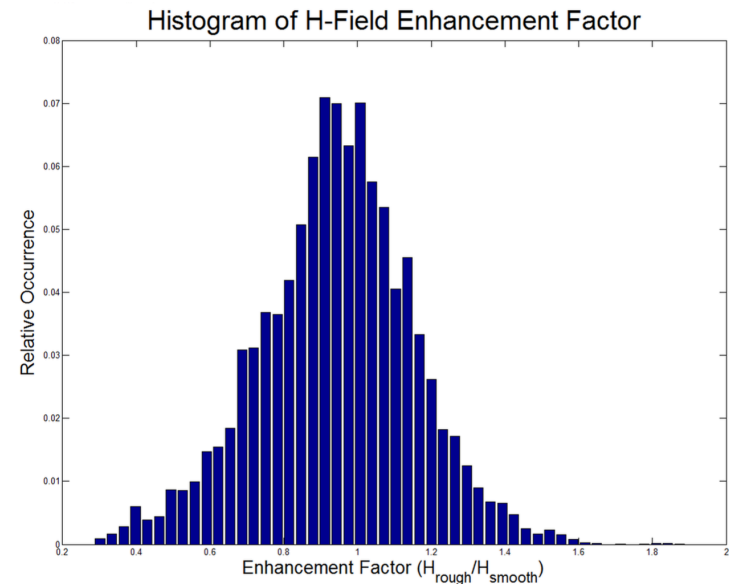
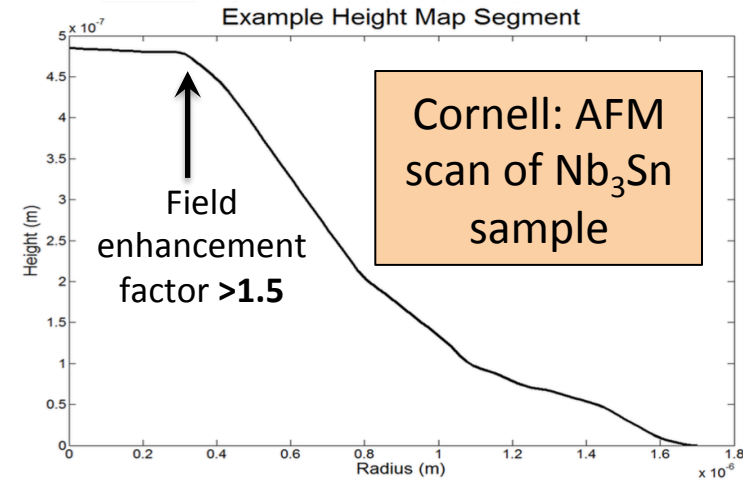
- Nb_3Sn is predicted to have **2x** the fundamental metastable limit of niobium



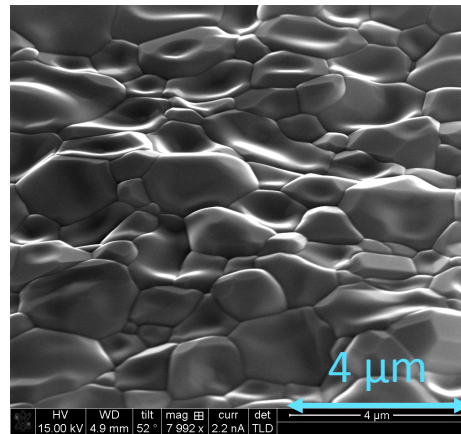
- Twice the energy gain per cavity?
- Not there yet—additional R&D required

Increasing E_{acc} in Nb_3Sn Cavities

- Substantial recent improvement in maximum E_{acc} with high Q_0
- Several promising paths forward for continued progress including:
 - Smooth sharp-edged surfaced
 - Reduce low tin content regions
- 20-40% increase: 22-25 MV/m



Y. Trenikhina, S. Posen, D. Hall, and M. Liepe, Proc. SRF Conference 2015, TUPB056, 2015

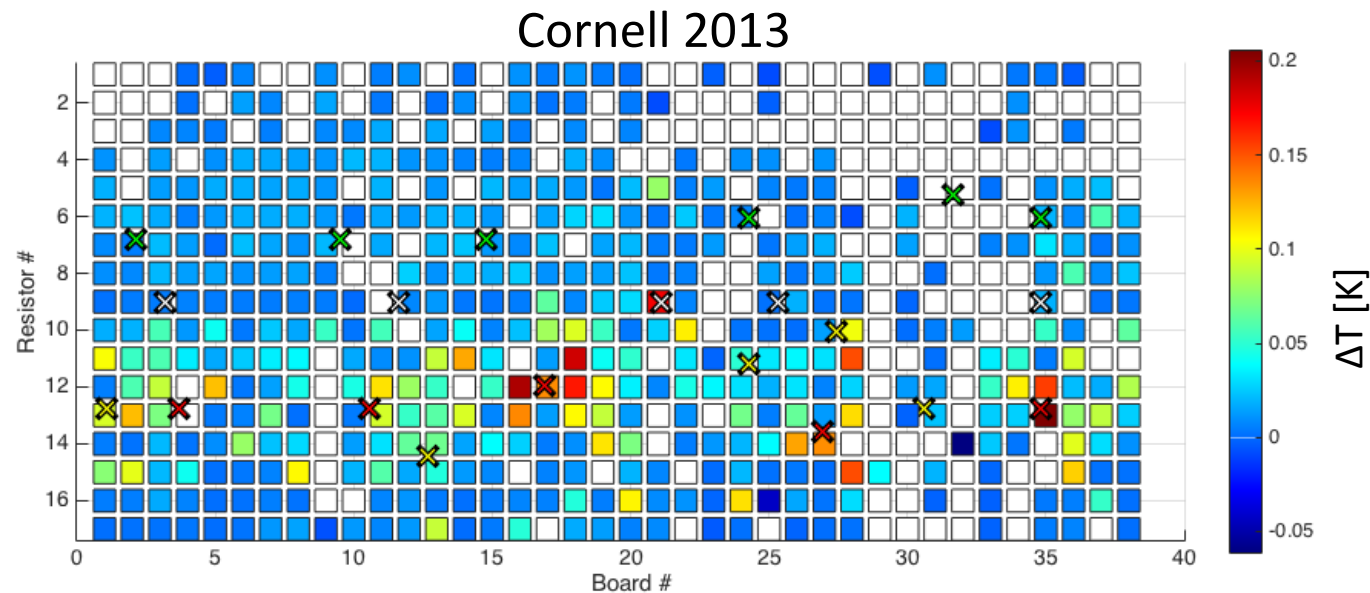
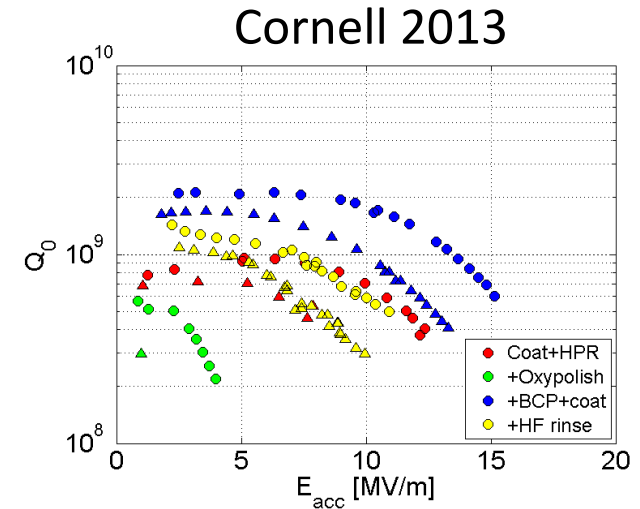


S. Posen, Ph.D. Thesis, Cornell University (2015).

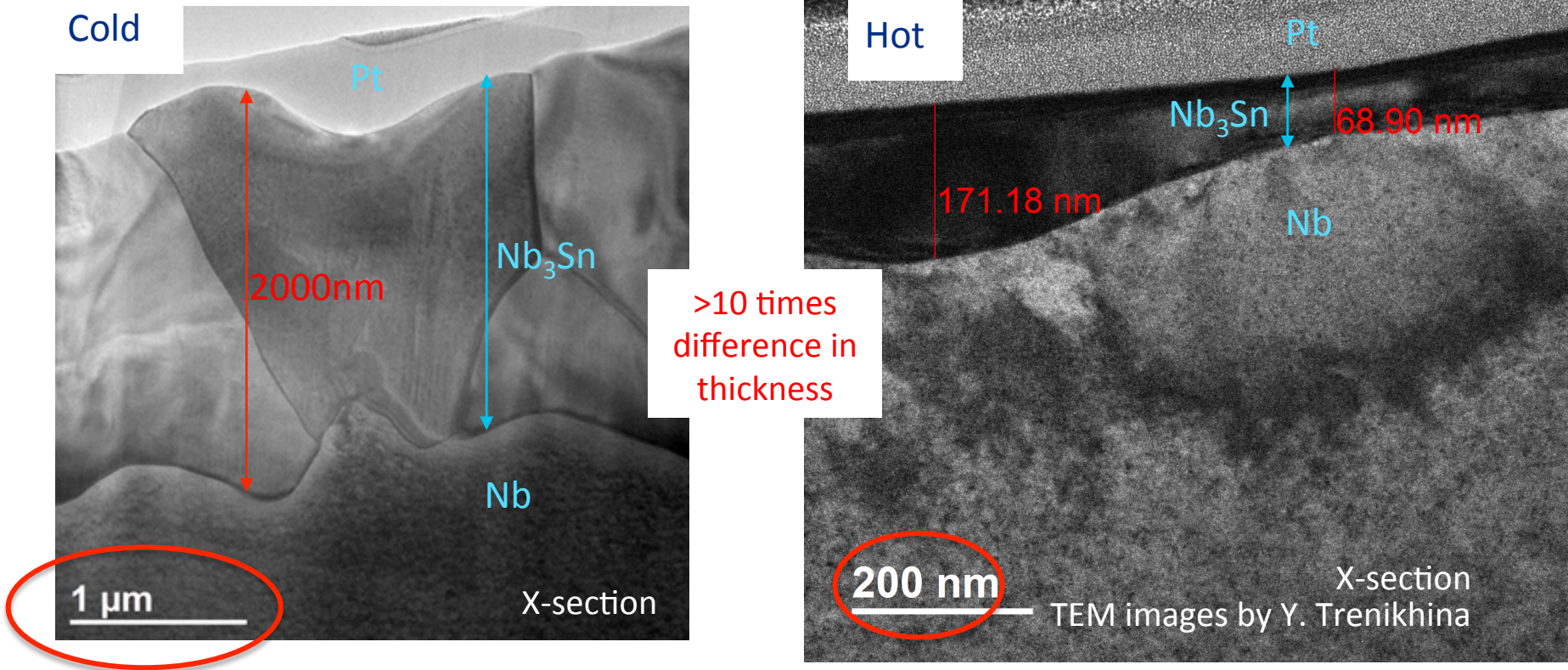
R. Porter, D. L. Hall, M. Liepe, J. T. Maniscalco, Proc. Linac Conference 2016, MOPRC027 (2016).

Coupons from Poorly Performing Cavity

- Poorly performing cavity coated in 2013
- In collaboration with Cornell, both labs retested it and cut out coupons from regions that showed high R_s and low R_s (hot spots and cold spots)



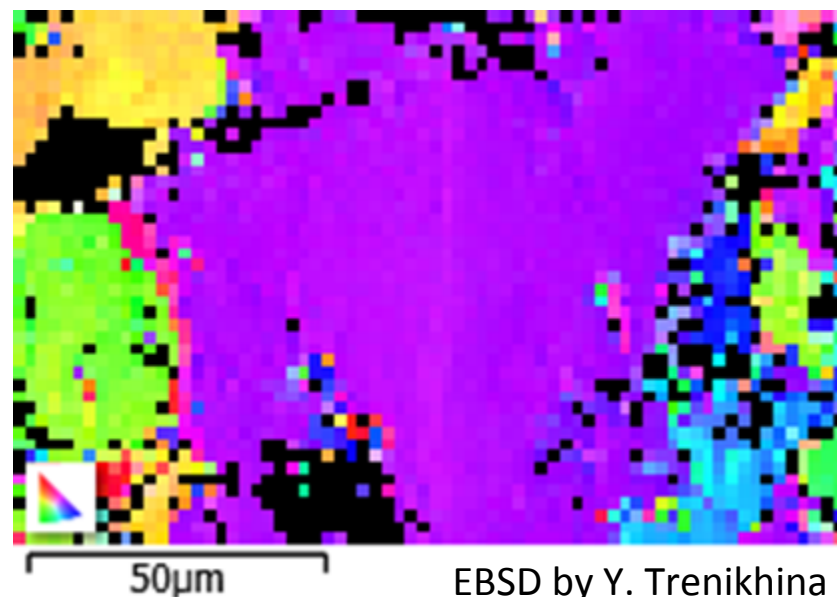
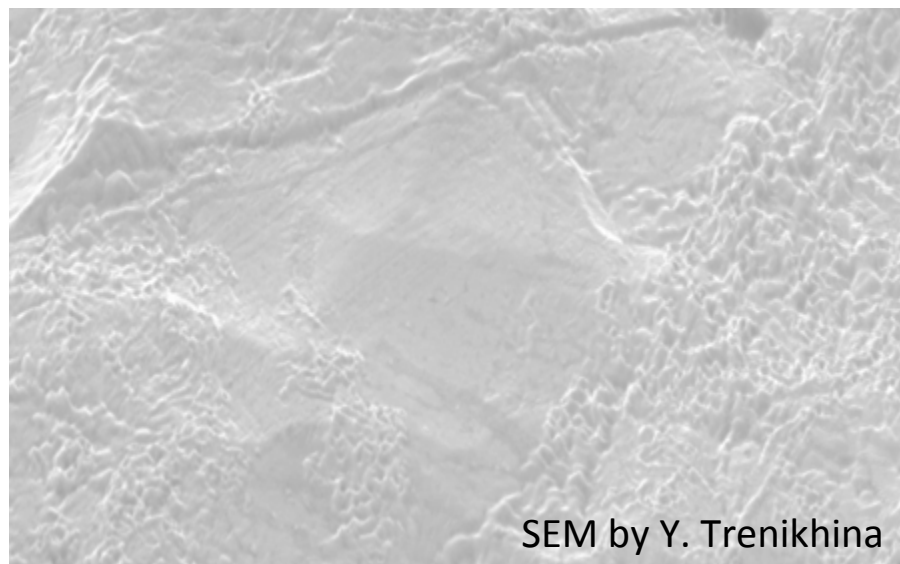
Thin Coatings in High Dissipation Hot Spots



- Regions with highest dissipation show very thin coatings, not thick enough to fully screen RF currents from Nb and intermediate Nb-Sn phases below Nb₃Sn layer

Thin Regions are Unusually Large Grains

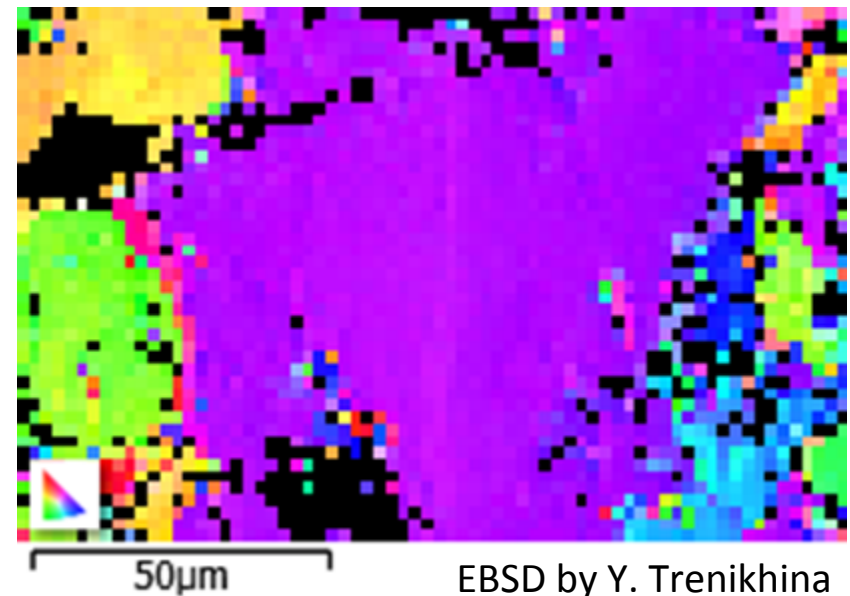
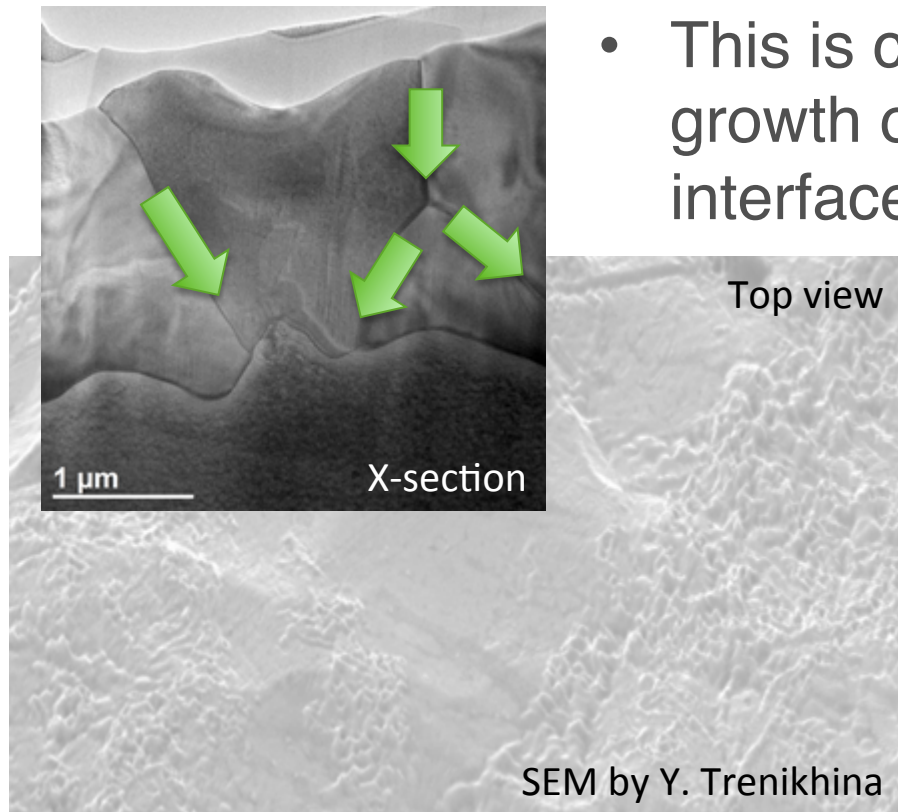
- EBSD analysis of grain orientation reveals that the thin regions in the hot spots are in fact large grains, with diameter ~ 100 microns vs ~ 1 micron for standard Nb_3Sn grains



Thin Regions are Unusually Large Grains

- EBSD analysis of grain orientation reveals that the thin regions in the hot spots are in fact large grains, with diameter ~ 100 microns vs ~ 1 micron for standard Nb_3Sn grains

- This is consistent with mechanism for growth of grains: diffusion of tin to interface via grain boundaries

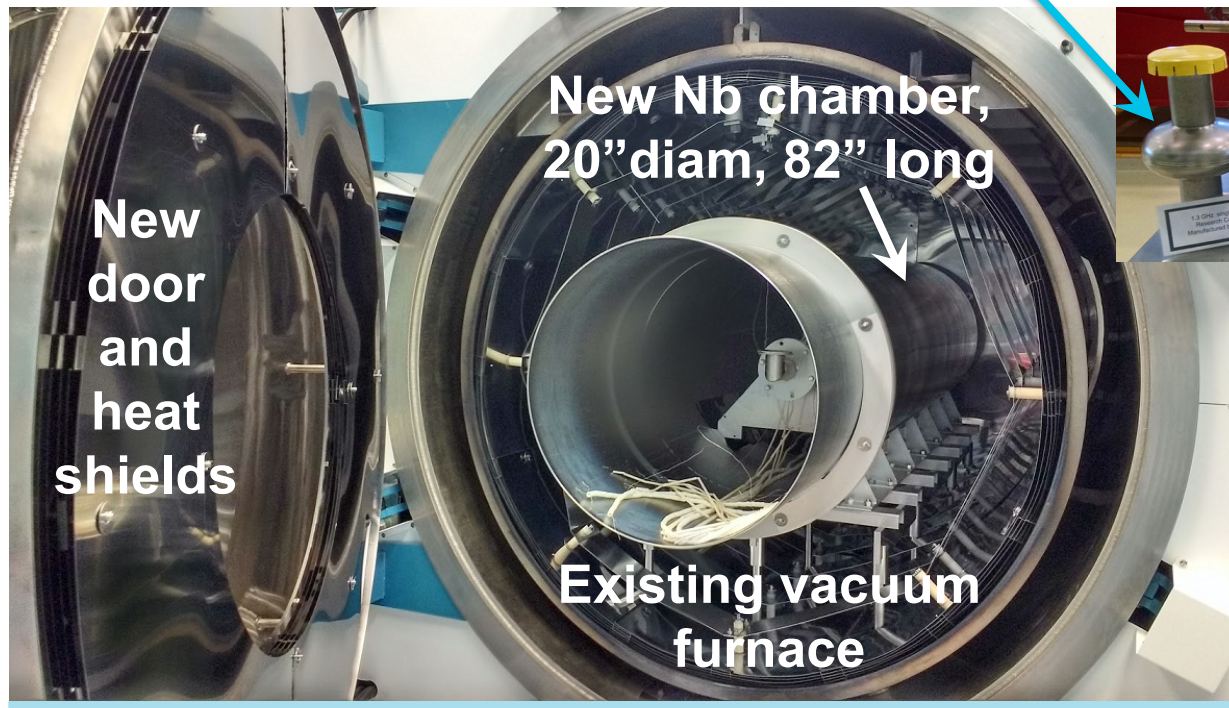


Nb₃Sn SRF Experimental Program at Fermilab

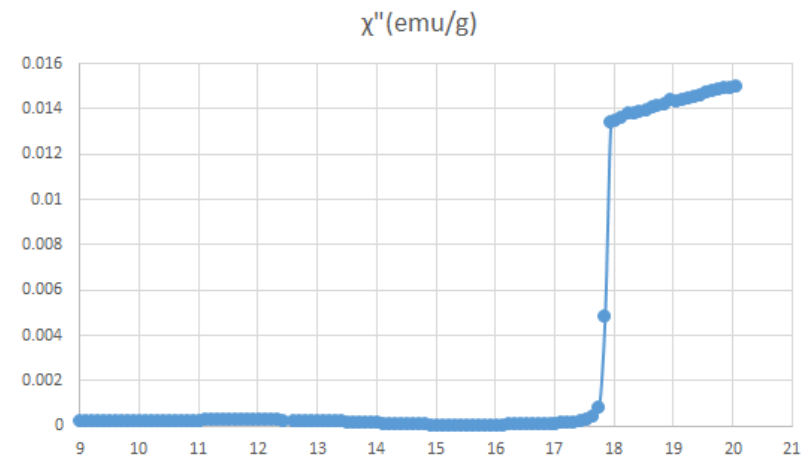
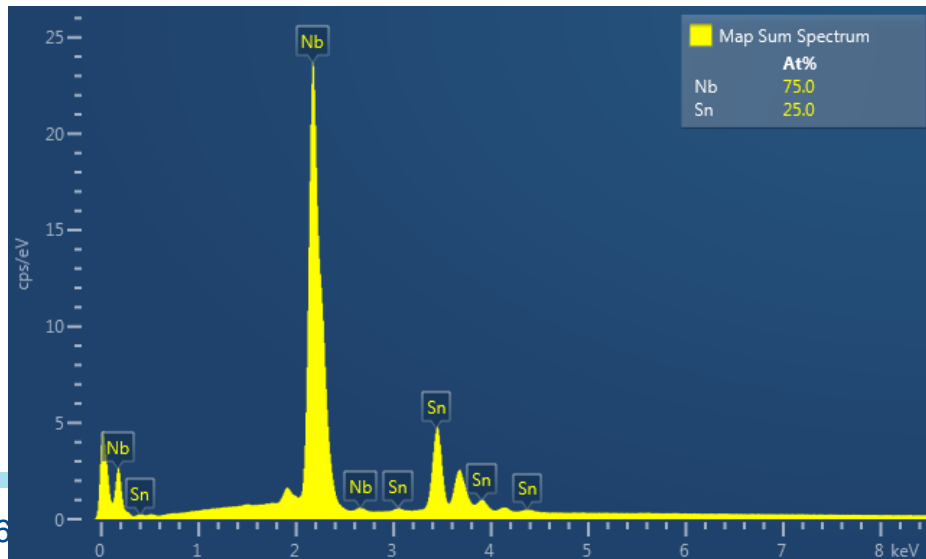
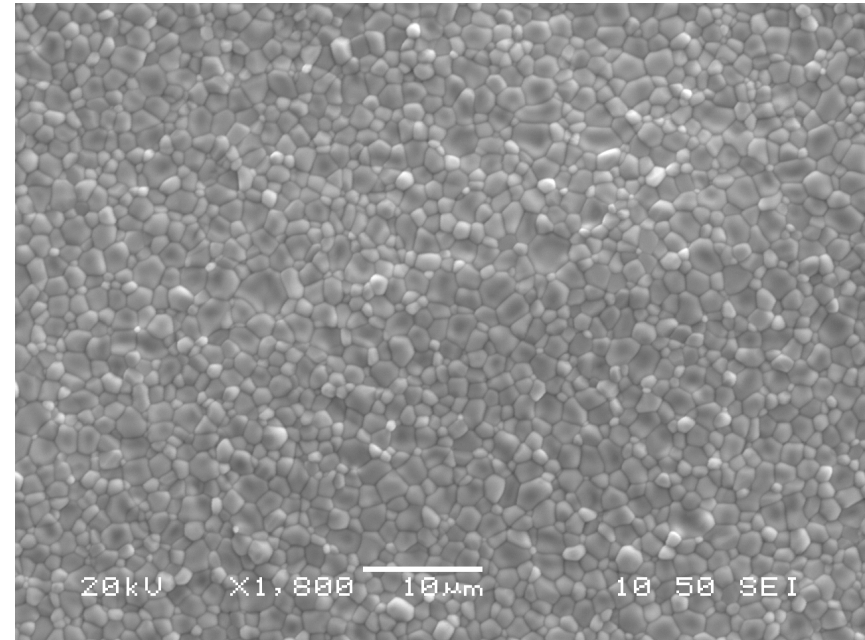
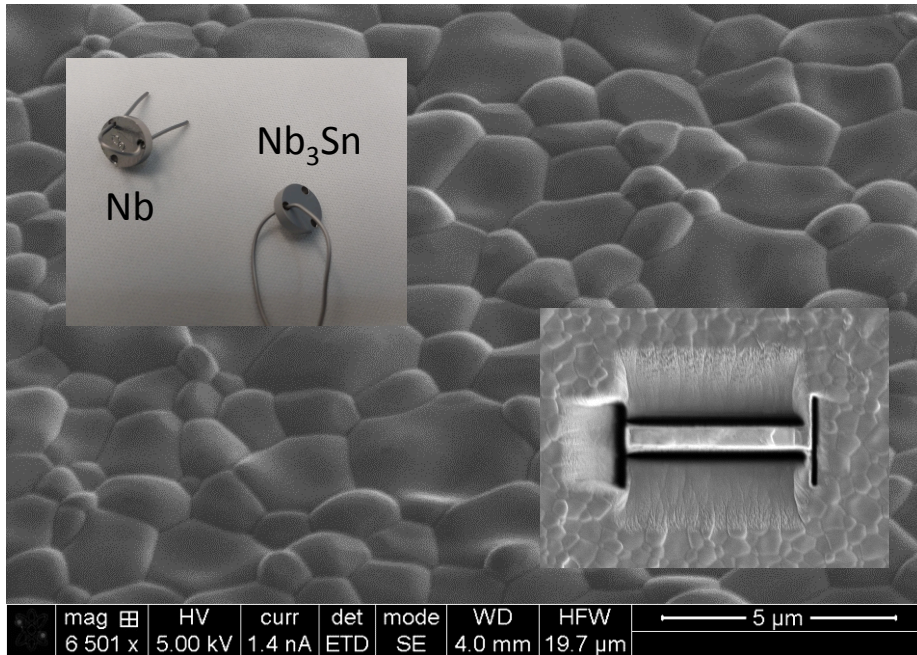
- Goals: 1) further improvement to Q_0 and E_{acc} 2) scale up to production-style cavities
- Large Nb₃Sn SRF coating apparatus recently commissioned
- First samples coated Jan 30, 2017

1.3 GHz 1-cell (current state of Nb₃Sn R&D)

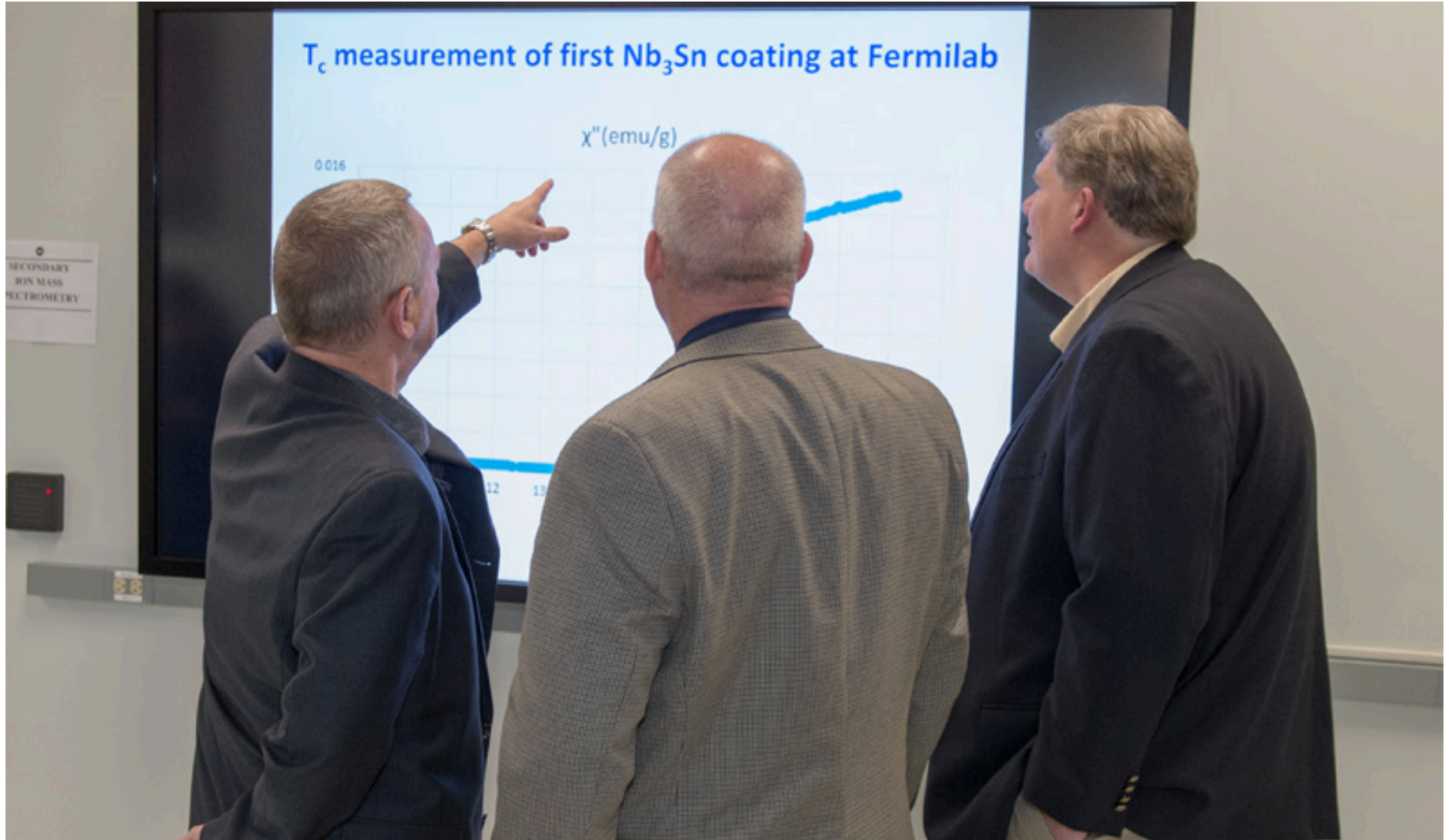
650 MHz 5-cell (future)



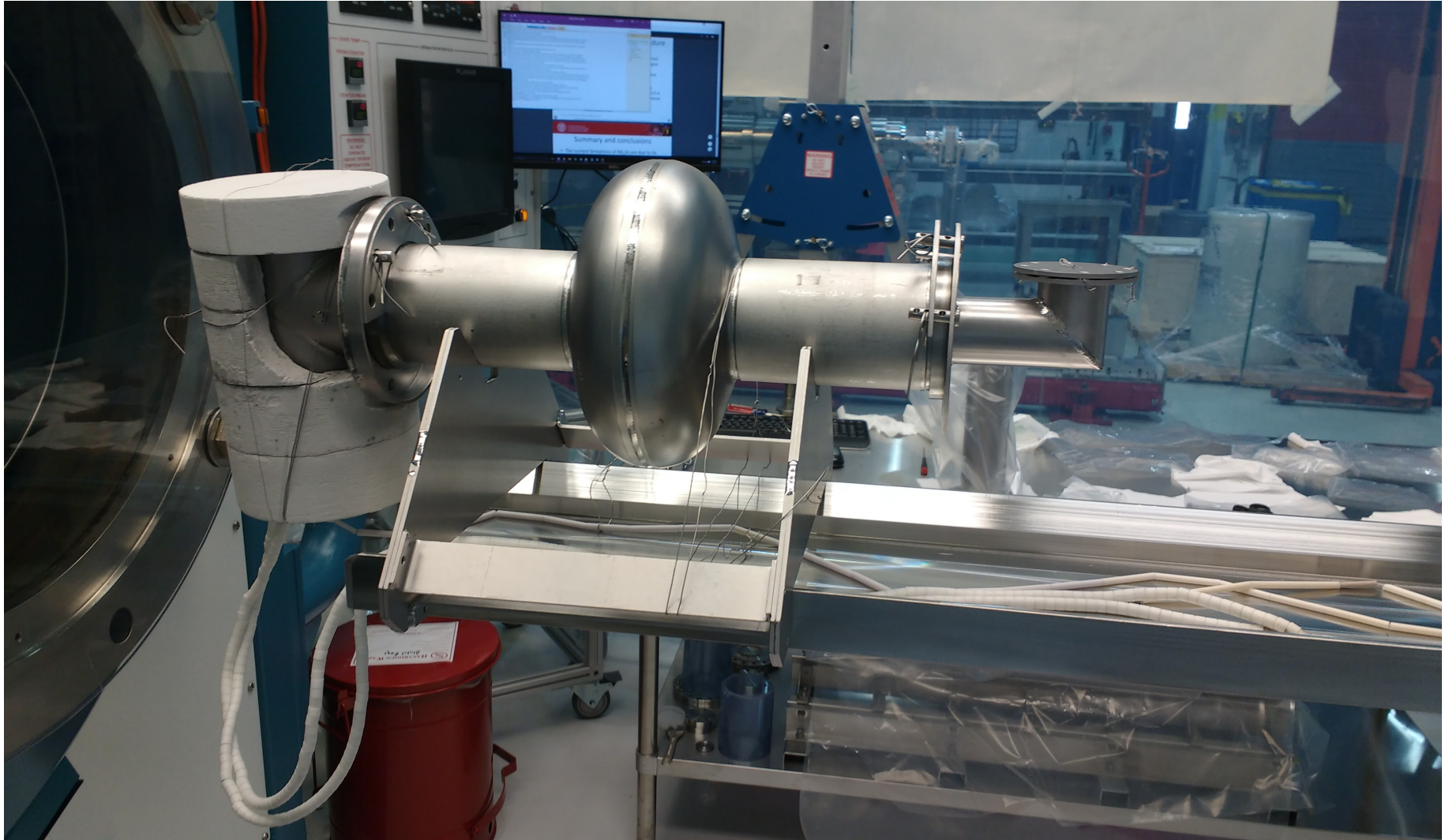
First Nb₃Sn Samples via Vapor Diffusion at FNAL



Excitement Brewing for Nb₃Sn SRF Coatings



First Fermilab Nb₃Sn Cavity Coated – RF Test Soon

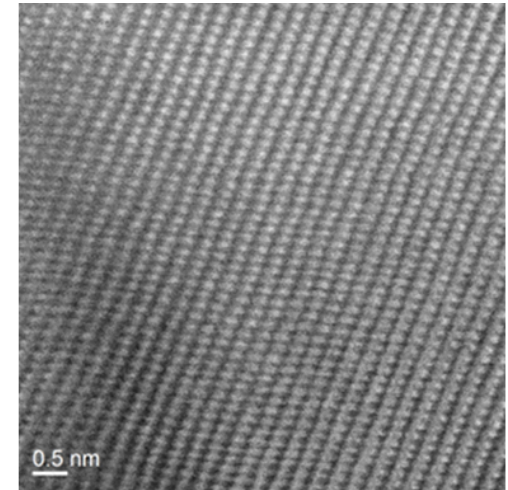


Outline

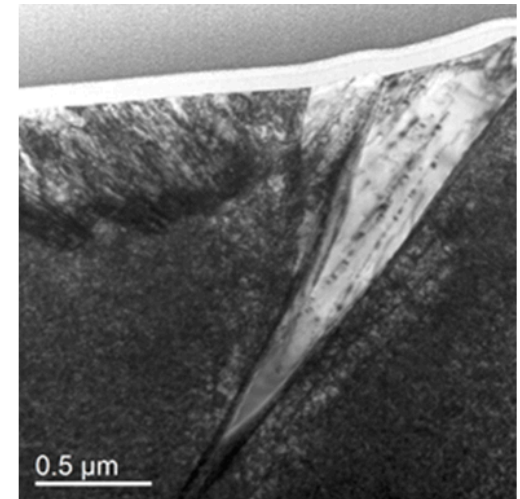
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 - Plasma cleaning



ILC cryomodule with record average gradient for FAST facility



Aberration-corrected STEM image of the near-surface of Nb cavity - Y. Trenikhina

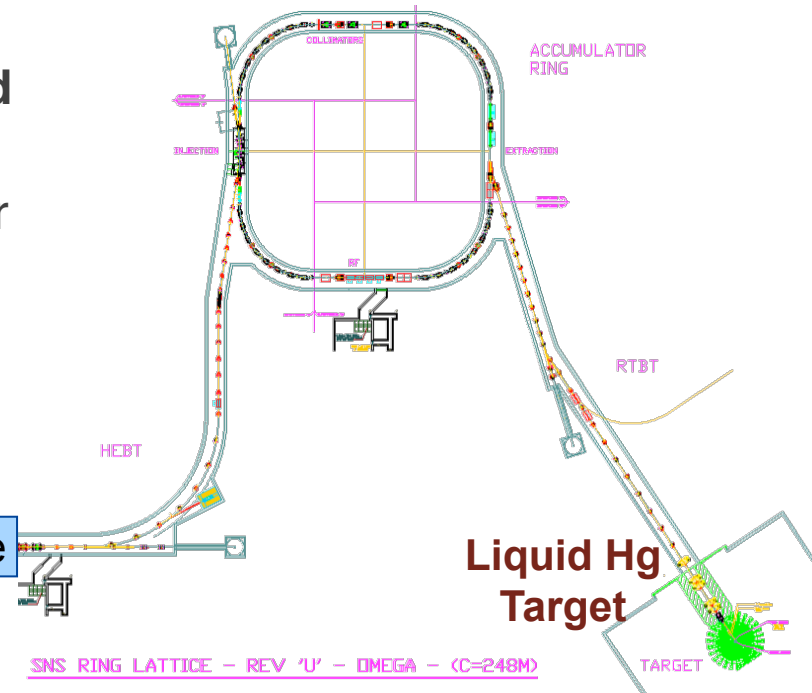


Niobium near-surface after the first step of nitrogen doping - Y. Trenikhina

In-situ plasma processing at SNS

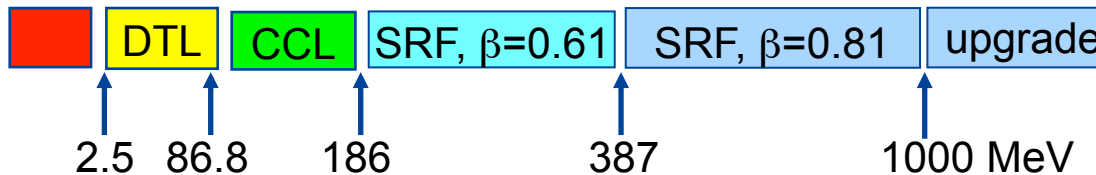
- Most cavities at SNS are limited by field emission (FE) leading to thermal instability in end-groups
 - Avg. Eacc are ~12 and 13 MV/m for the two cavity geometries
- In-situ plasma processing to reduce FE and increase accelerating gradients
 - Higher linac output energy adds margin for reliable operation at 1.4 MW
 - Aim is to increase high-beta SRF cavities gradient by 15%

**Accumulator Ring:
Compress 1 msec long pulse to 700 ns**



Liquid Hg Target

TARGET



Front-End:

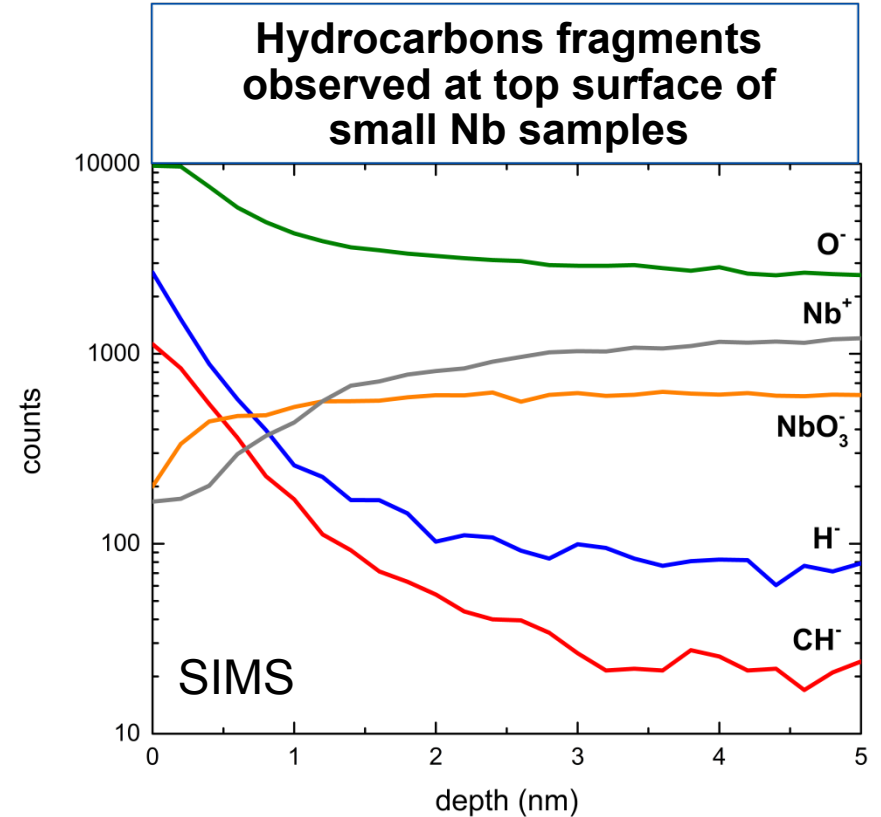
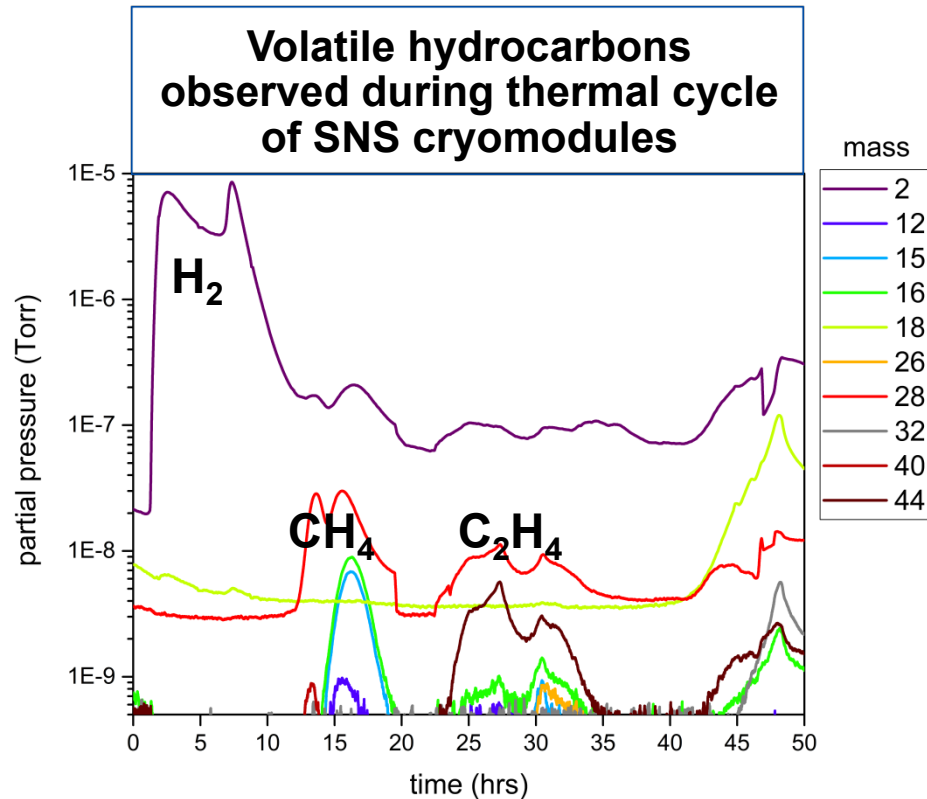
1 msec long chopped H-beam at 60 Hz

Linac:

Accelerates beam to 1 GeV
SRF linac: 23 cryomodules

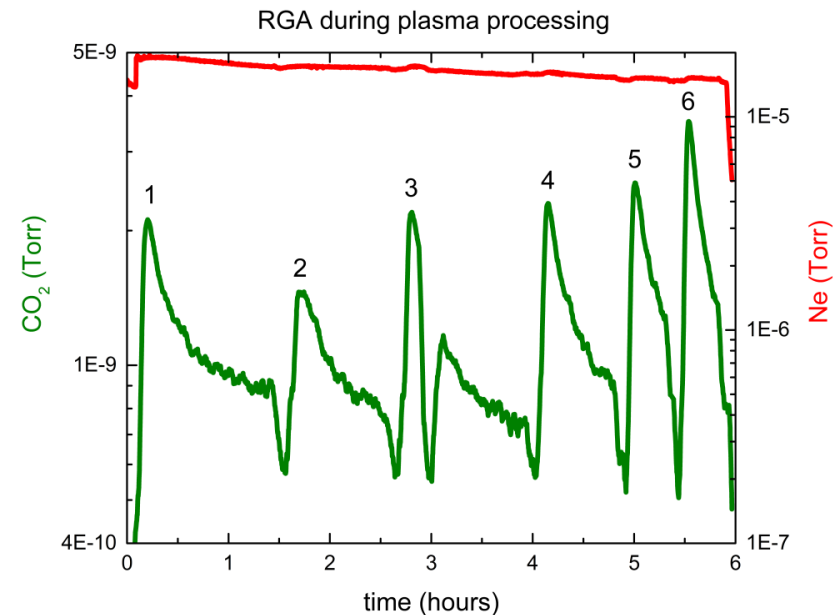
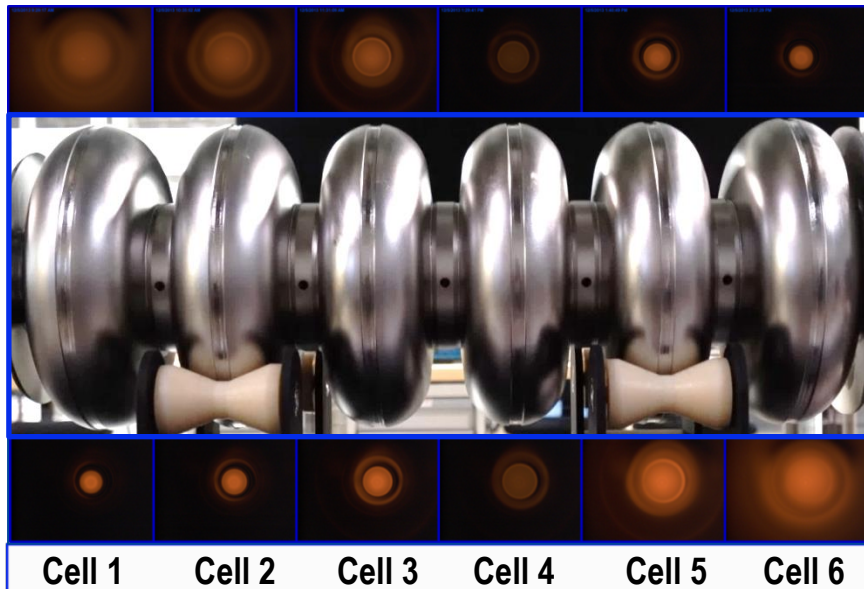
Hydrocarbon contaminants on Nb surfaces

- Hydrocarbon contamination observed on various Nb surfaces
- Hydrocarbons lowers the work function of Nb surface
 - Lower work function aggravates field emission
 - In-situ plasma processing at SNS removes hydrocarbons from cavity RF surface

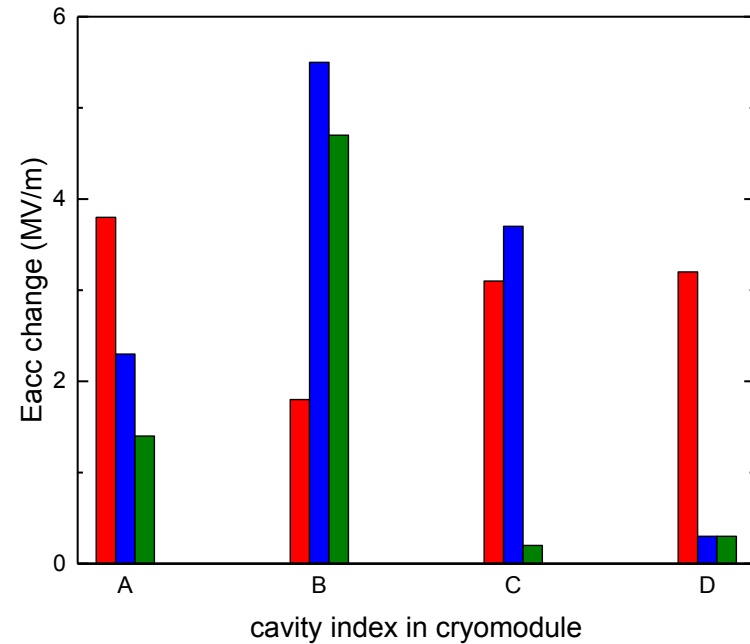
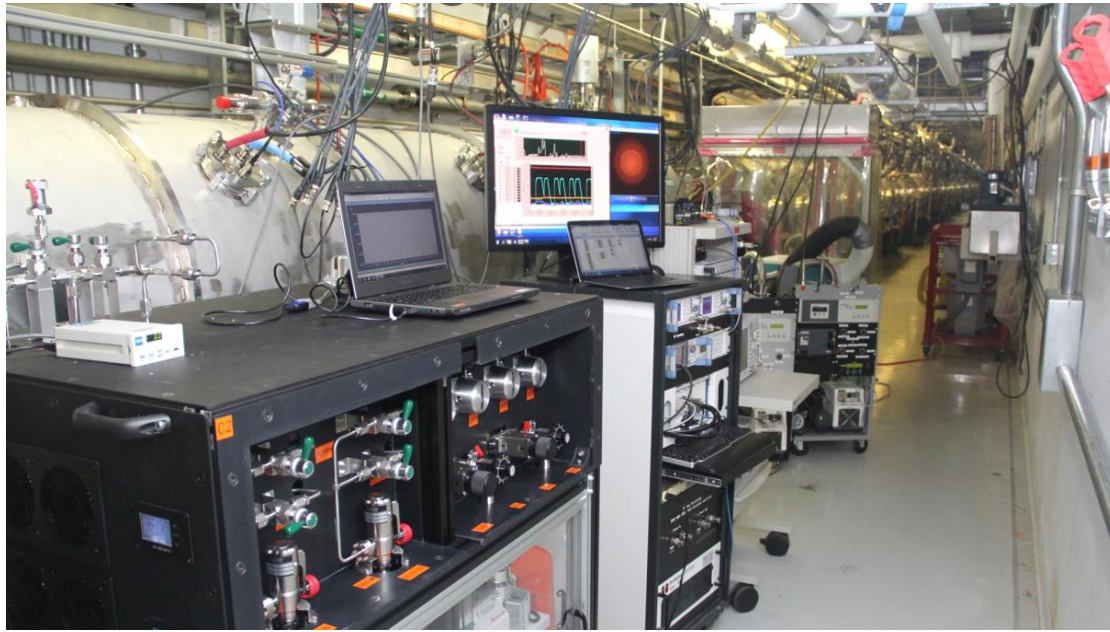


Neon-oxygen cleaning applied to SNS HB cavities

- **Reactive oxygen plasma at room-temperature**
 - 150 mTorr neon gas with ~1% oxygen
- **Hydrocarbons removed from top surface through oxidation and formation of volatile by-products such as CO₂**
- **Residual gas analysis used to monitor plasma cleaning**
 - Observed removal of ~monolayers equivalent of hydrocarbons
 - Six cells of SNS cavities cleaned sequentially



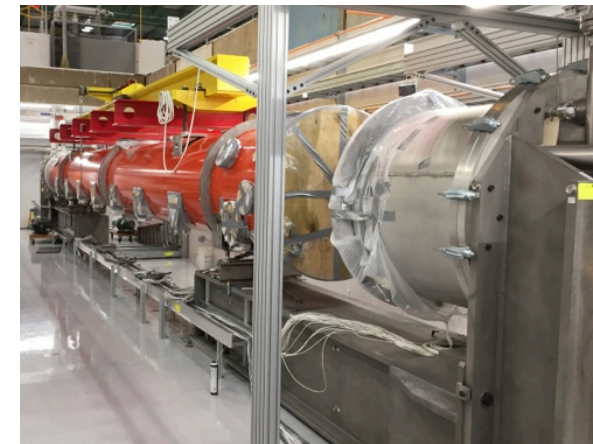
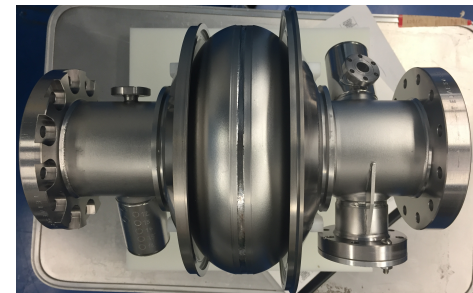
Plasma processing in SNS tunnel started



- **Improvement of Eacc after plasma processing of 3 cryomodules**
 - Ranges from 0.2 MV/m to 5.5 MV/m
 - 2.5 MV/m increase on average so far (21%)
- **SNS linac**
 - Currently operating at 972 MeV
 - Highest energy on production target at 60 Hz to date
 - Plan to reach 1 GeV in FY17

Mid-Term Goals: Plasma processing for LCLS-II

2017	2018	2019	2020
Present	Goals	Goals	Goals
Applicability of SNS plasma processing to LCLS-II	Plasma ignition in 9-cells cavity	Plasma processing in-situ in a cryomodule-like environment (HTS)	Plasma processing in-situ in LCLS-II cryomodules
Design of RF and vacuum system	Plasma processing and RF test of 9-cell cavities in VTS	Monitor of plasma ignition based on resonance shift	Improve maximum E_{acc} of $\sim 15-20\%$
Goals	Improve maximum E_{acc} of field emitting cavities		
Plasma processing of 1.3 GHz single-cell cavity			
RF test in VTS			



State-of-the-Art SRF Technology

- Exploring the limits of Q_0 and E_{acc} for accelerators
 - Impurity doping, Nb₃Sn, flux expulsion, plasma cleaning
- Science of RF Superconductivity
 - Manipulation of mean free path and energy gap with impurities
 - *Decrease* of surface resistance with increasing field
 - Flux trapping at grain boundaries and dislocations
 - Ultra-fast pulses to outpace vortex dissipation
 - Q_0 at *tiny* gradients (quantum computing)
- Auxiliaries: tuners, couplers, resonance control...
- Accelerators
 - LCLS-II
 - SRF for intensity frontier experiments at Fermilab
 - Future collider (circular or linear)

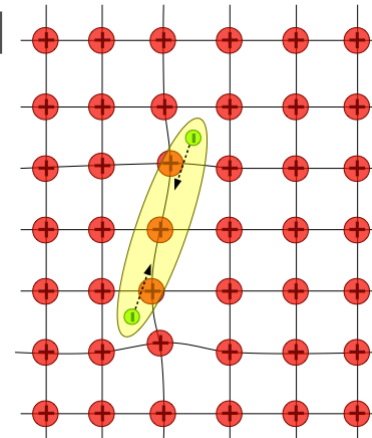
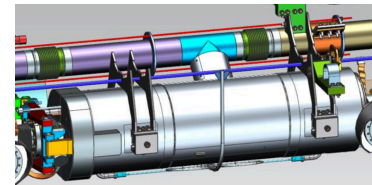


Image - arXiv:1208.5025



Video

Huge thanks to VMS for this incredible video: Jim, Reidar, Al, Lauren (Narrator)



*SRF
Technology*

SRF Technology Video

